

DFT study on electronic structures and optical absorption properties of C, S cation- doped SrTiO₃

Research Article

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Abstract:

The effects of C cation and S cation doping on the electronic structures and optical properties of SrTiO₃ are investigated by density function theory (DFT) calculations. The calculated results reveal that the top of the valence band is predominately made up of the O 2p states for the pure SrTiO₃. When SrTiO₃ was doped with C cation and S cation, the top of the valence bands consists mainly of O 2p+C 2s hybrid orbitals and O 2p+S 3s hybrid orbitals, respectively. The band gap of SrTiO₃ is narrowed by the doping with C cation and S cation, especially for the C and S-codoped SrTiO₃. Moreover, the red shifts of the absorption edge are found by the calculated optical properties, which is consistent with reported experiment results. It is the explanation for their visible light responsivity by the presence of C 2s and S 3s states on the upper edge of the valence band. All of these results can explain the good photocatalytic properties of C, S cation-codoped SrTiO₃ under visible light irradiation.

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1. Introduction

Photocatalysis is a technique that has great potential for the conversion of photon energy into chemical energy and for the decomposition of pollutants in air or solution. There is a need for effective catalysts for degradation of pollutants and water splitting. Because of its effective photocatalysis and photostability [1, 2], titanium dioxide (TiO₂) has been used for various industrial ap-

plications, such as water or air purification [3], antibacterial agents [4], and self-cleaning surfaces [5]. In addition to TiO₂, other metal oxides such as ZnO, Ta₂O₅, and SrTiO₃ have been reported as efficient photocatalysts [6–8]. Among these photocatalysts, SrTiO₃ has been used for water splitting and mineralization of organic pollutants under UV light irradiation [9–11]. However, their wide band gap (3.0–3.2 eV) requires UV light irradiation ($\lambda < 387$ nm), that only utilizes about 3% of the solar energy available. Many studies have been conducted to improve the TiO₂ photocatalytic efficiency and extend the photosensitivity toward the visible light region to use natural solar energy [12–14] by means of the catalyst doping [15–17].

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Recently, nonmetal-doped TiO_2 has become a rapidly growing field of interest [18–22]. N, S, or C anion-doped TiO_2 photocatalysts having an anatase form that show a relatively high level of activity under visible light irradiation have been reported [23–27]. Asahi *et al.* predicted by first-principle calculations and demonstrated in experiments that nitrogen-doped TiO_2 can lower its band gap and exhibit enhanced photoactivity under visible-light irradiation [23]. Recently, Wang *et al.* also reported an N-doped SrTiO_3 photocatalyst [28]. Teruhisa *et al.* also reported S, C cation-codoped SrTiO_3 and its photocatalytic activity under visible light [29]. After C and S ions were doped into SrTiO_3 , the absorption edge of SrTiO_3 powder shifted greatly from 400 to 700 nm. Moreover, the photocatalytic activity of the doped SrTiO_3 was improved. Electronic band structures, density of states and optical properties have been reported for the large series of Strontium titanate modified crystals [30]. However, we have not found any report on electronic structures of nonmetal-doped SrTiO_3 semiconductor photocatalyst. It is necessary to understand the effects of different non-metal doping on the electronic structures and photocatalytic properties of SrTiO_3 .

In this paper, we calculated and compared the electronic structures of undoped, C cation - and S cation -doped SrTiO_3 by using DFT calculations. From these results, the photocatalytic properties of C, S cation -doped SrTiO_3 can be explained.

2. Computational details

The crystal structure of SrTiO_3 is cubic perovskite with the space group $Pm\bar{3}m$, and the lattice constant is 0.3095 nm at room temperature. The Wyckoff positions of the atoms are: Sr 1a (0,0,0), Ti 1b (0.5,0.5,0.5) and O 3c (0,0,0.5,0.5). In our calculation, the electronic band-structure calculation was based on the CASTEP program [31]. The generalized gradient approximation (GGA) was adopted, an improvement of the local density approximation (LDA) within density functional theory [32–34] that is known to be an efficient and accurate scheme for solving the many-electron problem of a crystal. In the calculations, a $2 \times 2 \times 2$ super-cell was used in constructing the C, S cation-doped structures. To set the doped models, one titanium atom was replaced with a carbon atom and a sulfur atom for the C cation -doped and S cation-doped SrTiO_3 models, respectively. Two titanium atoms were replaced with a carbon atom and a sulfur atom for the C, S cation-codoped SrTiO_3 model, respectively. We used a $3 \times 3 \times 3$ Monkhorst–Pack k -point mesh for the Brillouin Zone integration. A scissors operator of 1.32 eV was introduced to shift the conduction

levels to be consistent with the measured value of the band gap [23, 35].

The absorption curves can be obtained from the imaginary part of the dielectric constant from the DFT calculation. The imaginary part of the dielectric constant is described as

$$\epsilon_2(\hbar\omega) = \frac{2e^2}{\Omega\epsilon_0} \sum_{c,v} \sum_k \left| \langle \Psi_k^c | \hat{u} \cdot \mathbf{r} | \Psi_k^v \rangle \right|^2 \delta[E_k^c - E_k^v - \hbar\omega].$$

Where Ω is the volume of the elementary cell, k represents v the k point, ω is the frequency of the incident light, c and v represent the conduction and valence bands, respectively. Ψ_k^c and Ψ_k^v are the eigenstates, \mathbf{r} is the momentum operator, and \hat{u} is the external field vector.

3. Results and discussion

3.1. Electronic structures

3.1.1. Density of states

In order to show the S, C doping effects on the electronic structure of SrTiO_3 , the total density of states (TDOS) of each system were calculated and the results are shown in Fig. 1. From Fig. 1 we can see that the widths of the valence bands of C, S cation - doped SrTiO_3 are a little larger than that of undoped SrTiO_3 because of the C 2p and S 3p bands located in the nether part of the valence band. In addition, the VB and CB positions of C- and S-doped SrTiO_3 slightly shift compared with that of undoped SrTiO_3 . For the C -doped SrTiO_3 , the top of valence band and the bottom of conduction band slightly shift towards high energies. However, for the S -doped and C, S -codoped SrTiO_3 , the band potentials slightly shift downwards. This indicates that VB of C, S-codoped SrTiO_3 should have a stronger oxidative power than that of undoped SrTiO_3 . So it is reasonable that C, S cation-codoped SrTiO_3 shows a higher photocatalytic activity than undoped SrTiO_3 [36].

3.1.2. Partial density of states

To further analyze the S, C doping effects on the electronic structure of SrTiO_3 , the partial density of states (PDOS) have been calculated, and the results are shown in Fig. 2a. The partial density of states of pure SrTiO_3 suggest that the top of valence band is made up predominately of the O 2p states, and the bottom of conduction band has Ti 3d character. There is an energy gap between the occupied O 2p states and the unoccupied Ti 3d states. For the C- and S-doped SrTiO_3 , the states in the valence

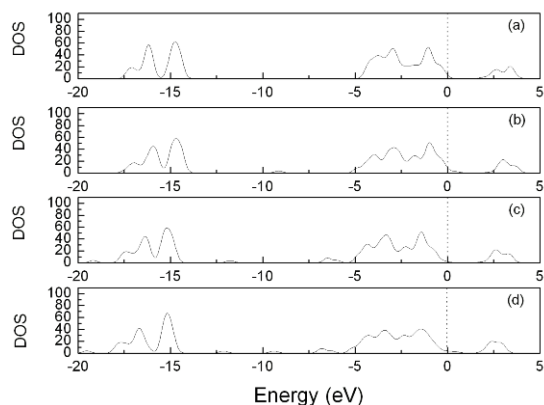


Figure 1. Calculated total density of states (DOS) of : (a) SrTiO₃, (b) C-doped SrTiO₃, (c) S-doped SrTiO₃, (d) C and S-codoped SrTiO₃.

band region mainly consist of the O 2p orbitals, but the contributions of C and S orbitals are not negligible. The top of valence band of the C-doped SrTiO₃ slightly shifts towards high energies by the presence of C 2s state localizing slightly above the upper edge of the valence band. In addition, for the C- and S-doped SrTiO₃, another small peak emerges above the lower VB at the energy of about -10 eV and -12.5 eV, in which C 2s and S 3s state make a great contribution, respectively. The conduction bands mainly consist of Ti 3d+C 2p orbitals for the C doping, and Ti 3d + S 3p orbitals for the S doping. For the C, S-codoped SrTiO₃, from the calculated results, we can find a similar variety. The conduction band mainly consists of Ti 3d+C 2p+S 3p orbitals.

There are three additional peaks between the upper VB and the lower VB. The three peaks are located at the bottom of the upper VB, above the lower VB and in the middle of the upper VB and the lower VB.

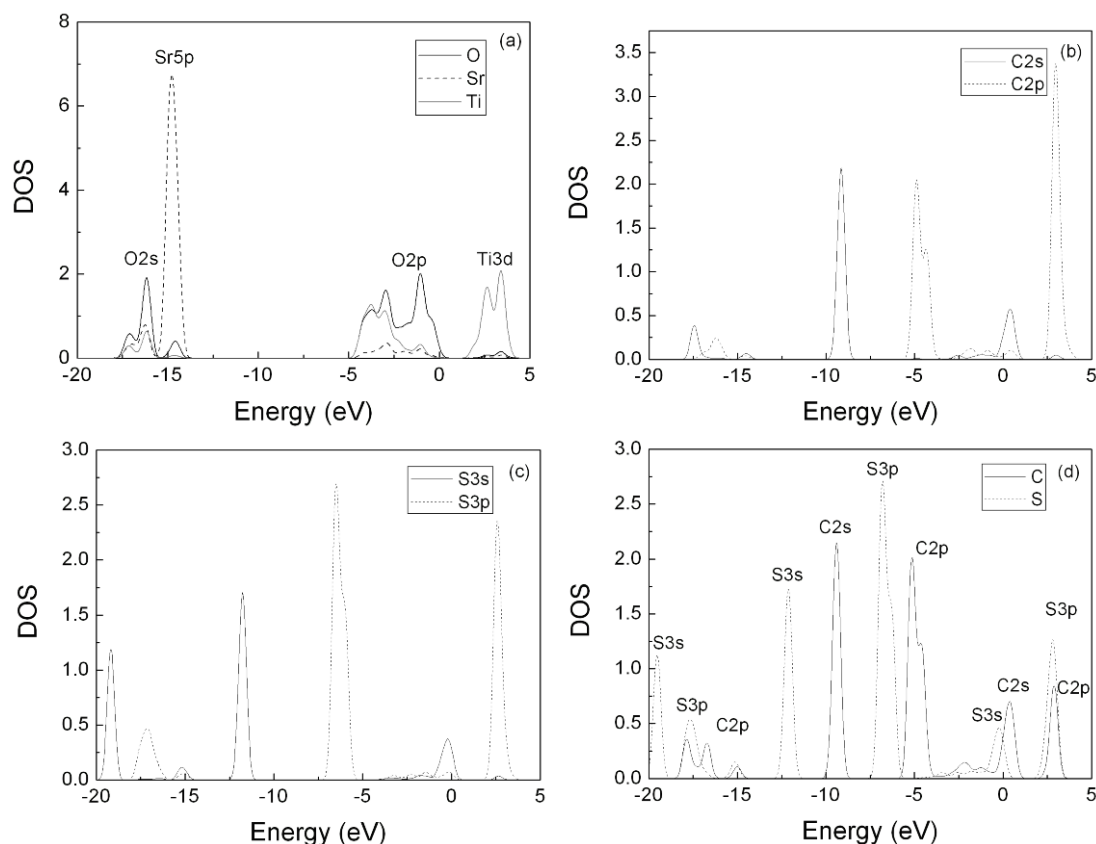


Figure 2. Partial density of states (PDOS) of undoped SrTiO₃ and doped SrTiO₃: (a) SrTiO₃, (b) PDOS of C in C-doped SrTiO₃, (c) PDOS of S in S-doped SrTiO₃, (d) PDOS of C,S in C and S-codoped SrTiO₃.

The contribution to the three peaks is S 3p+C 2p orbitals, S 3s orbitals and C 2s orbitals, respectively. The valence band mainly consists of the 2p orbitals of O atoms, mixed with some S 3s, C 2s and Ti 3d orbitals. There are some new states close to VB, and the contributions to the new states in the gap are derived from the mixture of S 3s, C 2s and O 2p orbitals. Excitations from the new states to a conduction band may be responsible for the red shift of the absorption edge observed in the C, S-codoped SrTiO₃ [36].

3.1.3. Band structures

The calculated energy band structure of each system is displayed in Fig. 3a. As shown in Fig. 3a, the band-gap energy is about 1.88 eV for the pure SrTiO₃, which is an overly underestimated value compared with the experimental value of 3.2 eV [37, 38]. This underestimation comes

from the inherent drawback of the GGA used for this calculation [39]. The calculated values however are acceptable for comparison. It is clear from this band structure that the band gap appears between the highest point of the valence band and the lowest point of the conduction band at the same G point. Therefore, a direct optical transition can occur. However, for the C-doped SrTiO₃ (Fig. 3b) and S-doped SrTiO₃ (Fig. 3c), the calculated band-gap energy is 1.52 eV and 1.72 eV, respectively. The band gaps show slight narrowing. Moreover, it indicates that C- and S-doped SrTiO₃ are all indirect band gap semiconductors with a minimum band gap between G and Q. Further analysis shows that the indirect band gap property is retained for the C, S-codoped SrTiO₃ and the band gap significantly becomes narrow due to C and S codoping. The band-gap energy is about 1.14 eV.

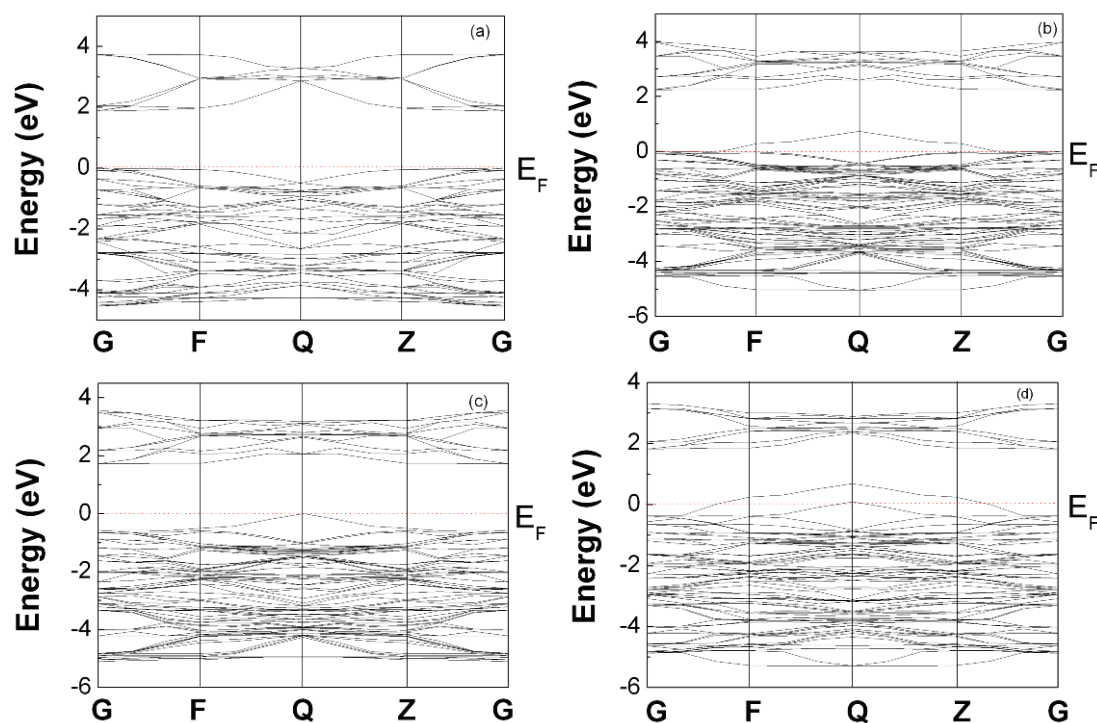


Figure 3. Calculated energy band structure of: (a) SrTiO₃, (b) C-doped SrTiO₃, (c) S-doped SrTiO₃, (d) C and S-codoped SrTiO₃.

Another pronounced feature shown in Fig. 3d is that the C 2s and S 3s states sufficiently overlap the original valence band of SrTiO₃. The formation of a continuum of states at valence band is crucial for the enhancement of the photocatalysis efficiency because it could reduce the

probability of the doping to trap the charge carriers. So it is reasonable that C, S cation-codoped SrTiO₃ shows a higher photocatalytic activity than undoped SrTiO₃, and the absorption edge is formed in the visible light region. These differences of photocatalytic activity under visible

light can be explained by correlation with their electronic structures [36]. This indicates to us that C and S play an important role in photocatalytic properties.

3.2. Optical properties

Further insights into the effects of C and S doping on the electronic structure of SrTiO₃ can be obtained from the calculation of optical properties. Fig. 4 shows the absorptions of undoped, C-doped, S-doped and C, S-codoped SrTiO₃. Red shifts of the absorption edges can be found for the C and S doping. This result is consistent with the large red-shift in UV-vis spectra reported by T. Ohno *et al.* [36]. These visible-light absorptions could be ascribed to the presence of the C 2s and S 3s state on the upper edge of the valence band reducing the electron transition distance for optical absorption. So a new absorption edge is formed in the visible light region due to doping of C and S atoms. The band-gap energies and the band-gap absorption edges calculated from the band-gap energies of these materials are summarized in Tab. 1. The absorption edge is 388, 437, 408, and 504 nm for the pure, C-doped, S-doped and C,S-codoped SrTiO₃ models, respectively. It can be speculated that the photocatalytic property of C-doped SrTiO₃ is better than that of S-doped SrTiO₃, and the photocatalytic property of the C, S-codoped SrTiO₃ is best.

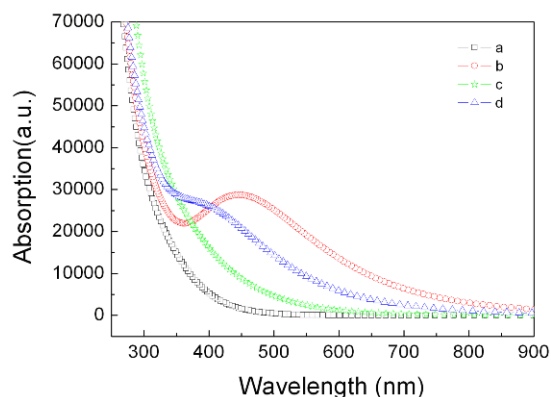


Figure 4. Calculated optical absorptions of: (a) SrTiO₃, (b) C-doped SrTiO₃, (c) S-doped SrTiO₃, (d) C and S-codoped SrTiO₃.

4. Conclusions

We have investigated the electronic structures including DOSs, PDOSs and band structures, as well as optical properties of the pure and C, S cation- doped SrTiO₃ by the calculations based on density function theory (DFT).

Table 1. The calculated band gap energy of the pure, C-doped, S-doped and C, S-codoped SrTiO₃, the value after correction of the scissor operator (+1.32 eV) compared with the experimental values in Ref. [36], the calculated absorption edge based on the band gap energy.

Model	Band gap (Calc.) (eV)	Scissor operator (+1.32 eV)	Band gap (Exp.) (eV)	Absorption edge (Calc.) (nm)
Pure SrTiO ₃	1.88	3.20	3.15	388
C-doped SrTiO ₃	1.52	2.84	—	437
S-doped SrTiO ₃	1.72	3.04	—	408
C,S-codoped SrTiO ₃	1.14	2.46	2.05~3.10	504

For the C and S cation-codoped SrTiO₃, O 2p orbitals mix with C 2s and S 3s orbitals, and Ti 3d orbitals mix with C 2p and S 3p orbitals, which contribute to the VB and CB, respectively. Moreover, the band potentials shift downwards and should have a stronger oxidation power. In addition, the calculated results indicate that owing to the C, S cation doping, the optical band gap of SrTiO₃ is narrowed and the red shift of the absorption edge is evidenced. Especially for the C and S cation-codoped SrTiO₃, the optical band gap is significantly lower. It might be the reason for the high photocatalytic activity under visible-light irradiation. So from these results, we can see that C, S cation doping is an efficient way to induce visible light activity for SrTiO₃ by providing C 2s and S 3s state on the upper edge of the valence band.

Acknowledgements

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