

Research Article

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Preparation, optical properties, and thermal stability of polyvinyl butyral composite films containing core (lanthanum hexaboride)–shell (titanium dioxide)-structured nanoparticles

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Abstract: Nano-sized lanthanum hexaboride (LaB_6)@titanium dioxide (TiO_2) particles with a core–shell structure has been successfully synthesized via a simple sol–gel method. LaB_6 @ TiO_2 particles were used as filler in polyvinyl butyral (PVB) matrix and performance of the TiO_2 shell was evaluated. The core–shell nanoparticles were characterized for morphology and structure properties. X-ray diffraction and transmission electron microscope testing results confirm the formation of LaB_6 – TiO_2 core–shell structure. In composite film, LaB_6 improved the thermo-decomposing temperature of PVB matrix from 369.2 to 372.8°C, while the same amount of LaB_6 @ TiO_2 further increased the temperature to 381.0°C. In addition, TiO_2 shell redshifted the maximum transmittance of the film from 605 to 669 nm in the visible region. In the near infrared region, its absorption peak shifted from 1,466 to 1,476 nm. This

result will be helpful for the development of transparent and thermal insulating materials.

Keywords: LaB_6 @ TiO_2 particles, optical properties, heat insulating materials, thermal stability, core–shell structure

1 Introduction

As a thermionic electron emitter material, LaB_6 is known for its excellent thermal stability, creep resistance, low work function, and so on. Up to now LaB_6 is found to possess higher electronic emissivity than other materials and therefore, it has been extensively applied in various areas, such as electron lithography, high resolution optical system, and coating film for resistor [1–4].

Surface plasmon resonance refers to collective oscillations of metallic electrons at the metal–dielectric interface. The metal-like plasmonic material lanthanum hexaboride (LaB_6) exhibits strong surface plasmon resonance absorption in the near infrared (NIR) region after being reduced to the nanoscale [5,6]. It indicates that LaB_6 absorbs the NIR light, while maintaining high transparency in visible region. Hence, LaB_6 acts as effective component in functional materials to impair heat transmission [7]. And now, different forms of the nanoparticles (NPs) like filler or coating of polymer films have found applications in glass curtain wall and automotive glass [8,9].

During the early 1990s, researchers prepared concentric multilayer semiconductor Nps in order to improve materials' properties. Since then, materials with a core–shell structure have attracted much attention [10–12]. This kind of materials could be synthesized by combining many components to form various cores and shells. These nanoscaled particles have many advantages over single components, such as wider applicability, better tunability, and higher sensitivity. Indeed, extensive research has been conducted on these materials. Kim et al. found

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that Ag/TiO₂ core-shell nanowires imparted exquisite resistance switching behavior to flexible, composite, and the ultra-thin shell in NPs and enhanced thermal stability of Ag core, while effectively blocking recombination of excitons [13]. Thus, this material was efficient in the disinfection of water under solar light irradiation due to distinctive surface plasmon resonance and electron charge transfer [14]. Controllable rutile titanium dioxide (TiO₂) shell in anatase@rutile core@shell TiO₂ nanosheets improved ethanol sensing performances [15]. Obviously, TiO₂ shell has attracted a great deal of interest in many fields due to its excellent chemical stability, photochemical activity, antibacterial activity, and so on [16–19].

More importantly, TiO₂ helps LaB₆ NPs distribute uniformly in polyvinyl butyral (PVB) due to electrostatic attraction formed at the interfaces between TiO₂ and LaB₆ [20]. So we expected to integrate LaB₆ core and TiO₂ shell to investigate how the shell would act in the core-shell structure and relevant properties. In the present study, PVB-based hybrid nanocomposite films were prepared by using core-shell structured LaB₆@TiO₂ NPs as fillers. The effect of the core-shell NPs on optical and thermal properties of the nanocomposite films will be introduced in this article.

2 Experimental methods

2.1 Materials

LaB₆ particles with diameter of 20–50 nm were provided by Ronghua Technology Co. (China). Tetrabutyl orthotitanate, silane coupling agent KH570, ammonium hydroxide (28–30 wt%), PVB (average molecular weight = 35,000–45,000 g/mol), dodecylbenzenesulfonic acid, ethanol, and ammonia were selected from Aladdin Chemical Co. (China). Deionized water was used throughout.

2.2 Preparation of core-shell LaB₆@TiO₂ NPs

According to Chen *et al.* research [21], LaB₆ NPs need to be pretreated with an anionic surfactant dodecylbenzenesulfonic acid to achieve a stable dispersion of LaB₆ in ethanol. In brief, LaB₆ particles were mixed with ethanol solution of dodecylbenzenesulfonic acid (0.05 wt%), and the content of LaB₆ NPs was fixed at 1 wt%. Then, the mixture was stirred for a few minutes until a navy blue

suspension was obtained. In order to achieve better homogenization, it was subsequently milled in a planetary ball mill for 24 h.

A sol-gel-assisted hydrothermal method elaborated elsewhere [22] was used for preparation of LaB₆@TiO₂ NPs. At first, 20 mL of the navy blue suspension, 2 mL of NH₃·H₂O, and 2 mL of tetrabutyl orthotitanate were added to 100 mL of ethanol-water (with volume ratio of ethanol/water = 9/1). Then, the obtained mixture was kept in a water bath at 40°C under vigorous stirring for 6 h. In this process, amorphous TiO₂ shells were formed on surfaces of LaB₆ NPs. Subsequently, a hydrothermal process was used to prepare LaB₆@TiO₂ NPs with anatase TiO₂ shells. The suspension was transferred into a Teflon-sealed autoclave and then placed in an oven at 160°C for 6 h. The products were collected by centrifugation and washed with distilled water and absolute ethanol several times.

2.3 Preparation of LaB₆@TiO₂-PVB and LaB₆-PVB nanocomposite films

Preparation of LaB₆@TiO₂-PVB composite films consisted of three steps. First, PVB powders were dissolved in ethanol via continuous stirring at 65°C for almost 1 h, and volume of ethanol was fixed at 50 mL throughout. Thus, ethanol solution of PVB was obtained. Second, LaB₆@TiO₂ particles and silane coupling agent KH570 were mixed in ethanol under vigorous stirring in an ultrasonic bath. KH570 aimed to promote the compatibility and dispersion of LaB₆@TiO₂ in PVB matrix. After complete dispersion, a stable navy blue suspension appeared. Finally, the suspension and an ethanol solution of PVB (10 wt%) were vigorously stirred at 65°C for 6 h (including 30 min of ultrasonic treatment). The as-obtained mixture was then cast onto slides by dip-coating method, and subsequently dried at 40°C for 24 h. After drying, films were removed from slides. Thickness of the films was fixed between 0.15 and 0.3 mm. A schematic diagram for fabrication of LaB₆@TiO₂ NPs and LaB₆@TiO₂-PVB nanocomposite films was presented in Figure 1. LaB₆-PVB films were produced when the fillers were LaB₆ NPs.

2.4 Measurements

Morphology of LaB₆@TiO₂ NPs was determined by Transmission electron microscope (FEI-TEM; Tecnai G2 F20

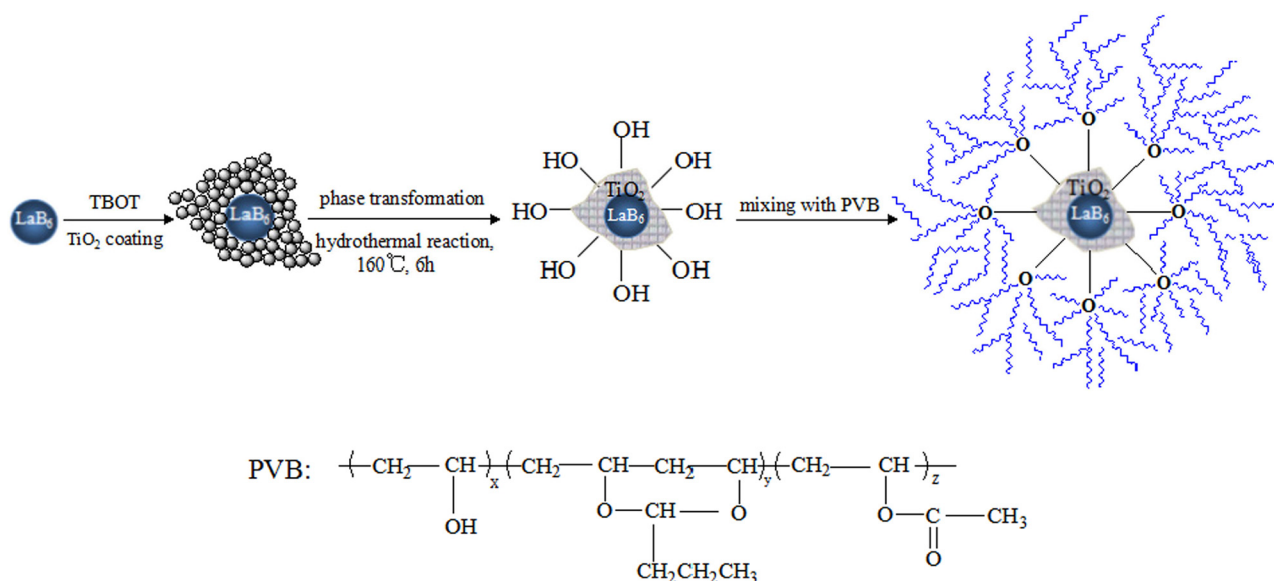


Figure 1: The schematic diagram of fabrication of $\text{LaB}_6@ \text{TiO}_2$ NPs and $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films.

S-TWIN) equipped with energy dispersive X-ray spectroscopy. For the TEM measurement, $\text{LaB}_6@ \text{TiO}_2$ NPs were dispersed in ethanol and then transferred onto a copper grid covered with carbon film. XRD patterns of LaB_6 , TiO_2 , and $\text{LaB}_6@ \text{TiO}_2$ NPs were recorded by D-MAX2500 (Rigaku) with $\text{Cu K}\alpha$ radiation ($\lambda = 0.1542 \text{ nm}$ and step size = 0.02°) at a scanning rate of $8^\circ/\text{min}$ in the 2θ range of $15\text{--}80^\circ$. Nicolet-6700 spectrophotometer (USA) with $4/\text{cm}$ resolution was used to measure the FT-IR spectra of PVB and nanocomposite films in the range of $4,000\text{--}400/\text{cm}$.

The UV-Vis-NIR transmission spectra of 0.30 wt% LaB_6 -PVB film and 0.30 wt% $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films were detected by a UV-vis-NIR spectrophotometer (Japan, Shimadzu UV-3600) in a wavelength range of $380\text{--}2,000 \text{ nm}$. The neat PVB films were set as a blank background. Thermogravimetric findings (TGA) and differential scanning calorimeter (DSC) curves of neat PVB and nanocomposite films were recorded by a STA449C simultaneous thermal analyzer at a heating rate of $10^\circ\text{C}/\text{min}$ from 25 to 700°C under N_2 atmosphere.

3 Results and discussion

3.1 XRD patterns

Figure 2 presents the XRD patterns of LaB_6 , TiO_2 , and $\text{LaB}_6@ \text{TiO}_2$ NPs. The diffraction planes at 21.4° , 30.4° (strongest), 37.4° , 43.5° , and 49.0° were assigned to the (100), (110), (111) (200), and (210) planes of cubic LaB_6

(JCPDS No. 34-0427), respectively. TiO_2 prepared through the sol-gel-assisted hydrothermal method exhibited diffraction peaks at 25.4° (strongest), 37.0° , 37.8° , 38.6° , 48.1° , and 54.0° which were attributed to (101), (103), (004), (112), (200), and (105) crystal faces of anatase phase (JCPD No. 89-4921), respectively. Since LaB_6 NP was enveloped with TiO_2 shell by a sol-gel hydrothermal process, diffraction peaks of the core-shell structured particles were compatible with cubic LaB_6 and anatase TiO_2 . In addition, no other diffractive peaks appear in the pattern. Neetu [23] and Liu et al. [24] also reported hydrothermal synthesis of anatase TiO_2 shell, and their XRD patterns are consistent with ours. It is concluded that

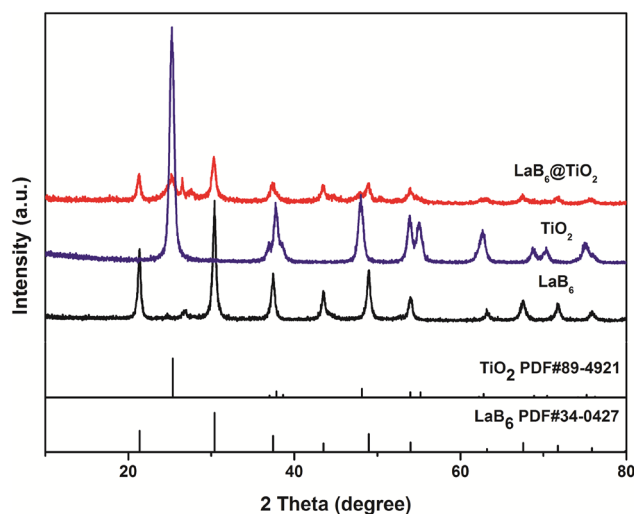


Figure 2: XRD patterns of LaB_6 , TiO_2 , and $\text{LaB}_6@ \text{TiO}_2$ NPs.

the sol–gel-assisted hydrothermal method to prepare the $\text{LaB}_6@\text{TiO}_2$ NPs with high purity is feasible.

3.2 Morphology analysis

The size and structure of the core–shell-structured $\text{LaB}_6@\text{TiO}_2$ NPs is determined using TEM characterization technique and

the images are shown in Figure 3. It is clearly visible in Figure 3a that many particles with an average diameter of 80 nm are aggregated and there is a shielding of core particles by the TiO_2 shell. Figure 3b presents an electron diffraction (ED) pattern of the $\text{LaB}_6@\text{TiO}_2$ NPs. The diffraction rings are attributed to (100) and (110) planes of LaB_6 (PDF#-34-0427) and (101) and (103) planes of TiO_2 (PDF#-89-4921), respectively. In order to detect the elements of the composite NPs, samples were analyzed by EDX, and the results are displayed

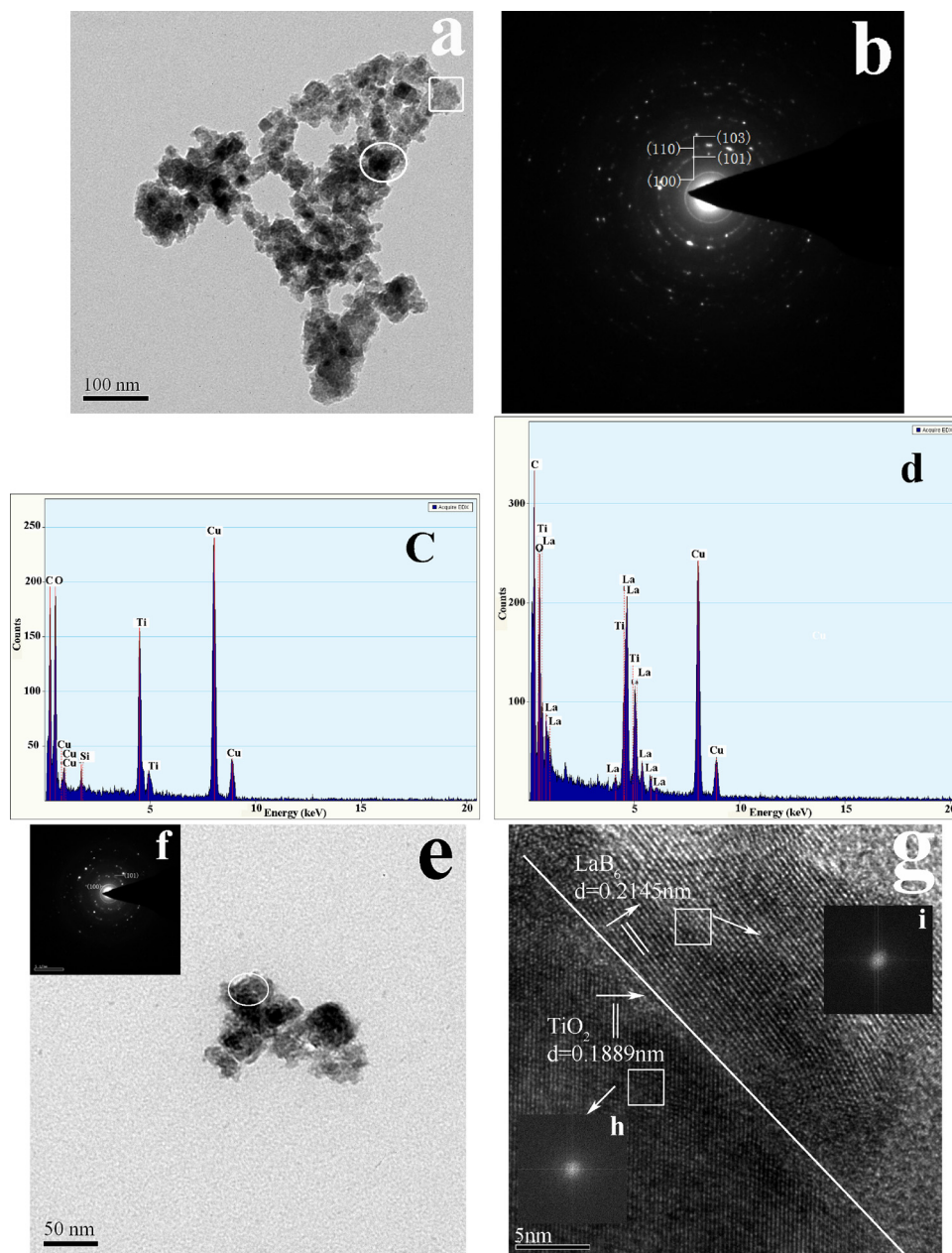


Figure 3: (a) TEM image and (b) electron diffractive spots of $\text{LaB}_6@\text{TiO}_2$ NPs. EDX results of (c) the square spot and (d) the circular spot in (a). (e) TEM image and (f) electron diffractive spots of a $\text{LaB}_6@\text{TiO}_2$ NP. (g) HRTEM image of $\text{LaB}_6@\text{TiO}_2$ NP, the insets (h) and (i) are FFT images of the core and the shell, respectively.

in Figure 3c and d, corresponding to spots marked by the square and the circle in Figure 3a, respectively. As shown in Figure 3c, Ti peaks which are attributable to TiO_2 shell formed in the sol-gel method are significantly distinguished from other peaks, and O peak is attributed to TiO_2 and residual dodecylbenzenesulfonic acid absorbed on surfaces of $\text{LaB}_6@ \text{TiO}_2$ core-shell NPs. Besides, no La peaks are found. As compared to Figure 3c, La peaks in Figure 3d are apparently observed due to high content of LaB_6 at this spot. However, the light element boron does not present any peaks here due to its property, or the peaks may be overlapped by carbon's. In addition, Ti, C, and O elements were also detected. Two peaks at 8–9 keV are originated from the Cu substrate used for TEM detection. Combined with the TEM image in Figure 3a, it can be obtained that the NPs are only consisted of LaB_6 and TiO_2 without any other elements, which is in good agreement with the XRD findings; and the dark LaB_6 cores are surrounded by TiO_2 shell.

A high-resolution TEM image of a single $\text{LaB}_6@ \text{TiO}_2$ NPs is presented in Figure 3e. The inner LaB_6 core is tightly surrounded by a loose TiO_2 layer, and the interface between the dark core and the translucent continuous shell is noticed. Most of the shells possess a thickness less than 5 nm, but a few of them are approximately 15 nm thick. Figure 3f (inset in Figure 3e) is an ED pattern of the section marked by a circle. To further demonstrate the detailed microstructure of the interface between the cores and shells, high-resolution TEM (HRTEM) and fast Fourier transmission (FFT) measurements were carried out, and data are displayed in Figure 3g. It is clearly displayed that the two different crystal growth directions which are perpendicular to their corresponding crystal planes exist, with a distinct boundary between them. The inter-planar spacing of around 0.2145 nm is attributed to the (200) plane of LaB_6 and the lattice spacing of approximately 0.1899 nm is ascribed to the (200) plane of TiO_2 . Moreover, Figure 3h and i illustrates diffractive spots of anatase TiO_2 shell and crystalline LaB_6 core, respectively. These results further confirm that there are two different crystal forms in the $\text{LaB}_6@ \text{TiO}_2$ NPs, coinciding with the results of the above XRD patterns.

3.3 FT-IR spectra

FT-IR spectra of the composites are exhibited in Figure 4. For the pure PVB films, the peaks at 1,700–1,600 and 3,700–3,100/ cm^{-1} are attributed to the stretching and bending vibrations of O–H in PVB matrix, ethanol, or H_2O left in the films. The peaks at 2,956 and 2,877/ cm^{-1} correspond to the

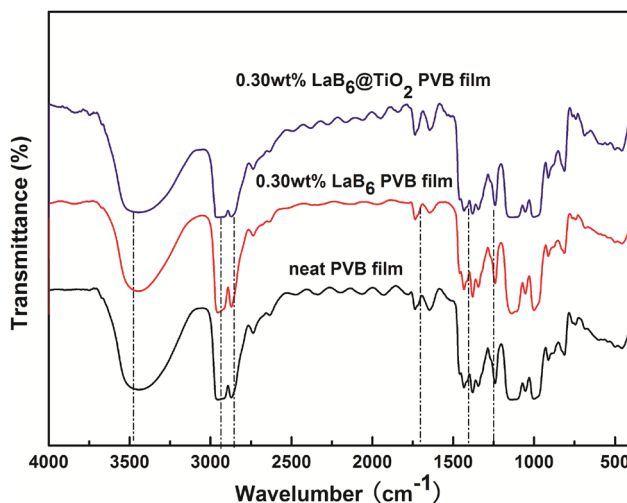


Figure 4: FT-IR spectra of the films.

stretching vibration of aliphatic C–H groups. The peaks at 1,435, 1,382, and 1,346/ cm^{-1} are attributed to the bending vibration of C–H groups. The peaks at 1,135 and 1,047/ cm^{-1} are originated from C–O–C–O–C stretching vibration of pentatomic and hexatomic cyclic acetal groups. The peaks at around 1,731/ cm^{-1} belong to the stretching vibration of C=O in acetate group. The peaks at 987 and 1,234/ cm^{-1} are assigned to –OH asymmetric stretching of polyvinyl alcohol [25]. For the $\text{LaB}_6@ \text{TiO}_2$ -PVB, the peaks at around 520/ cm^{-1} are characteristic peaks of TiO_2 crystals [20] and ascribable to the stretching vibration of Ti–O–Ti bonds in TiO_2 [26,27]. Tekin et al. [28] also observed Ti–O–Ti stretching vibration modes of anatase TiO_2 in a form of a broadband ranging from 647 to 830/ cm^{-1} .

These spectra are similar in appearance, and it may be attributed to low contents of solid particles in the composite films and their weak absorption in this region.

3.4 Thermal degradation characteristics

The thermodynamic stability of the films was investigated by means of DSC and TGA (Figure 5). As demonstrated by a TGA curve in Figure 5a, thermal degradation of neat PVB film included two steps. The first weight loss that occurred below 100°C may be resulted from water evaporation in the matrix, which was absorbed in the film due to Van der Waals force. The second weight loss that started at 250°C and finished at 400°C is probably attributed to the degradation of PVB matrix, and it exhibited a weight loss of 90.3%. However, it is obvious that PVB in LaB_6 -PVB and $\text{LaB}_6@ \text{TiO}_2$ -PVB films degraded at higher temperatures. Especially for $\text{LaB}_6@ \text{TiO}_2$ -PVB

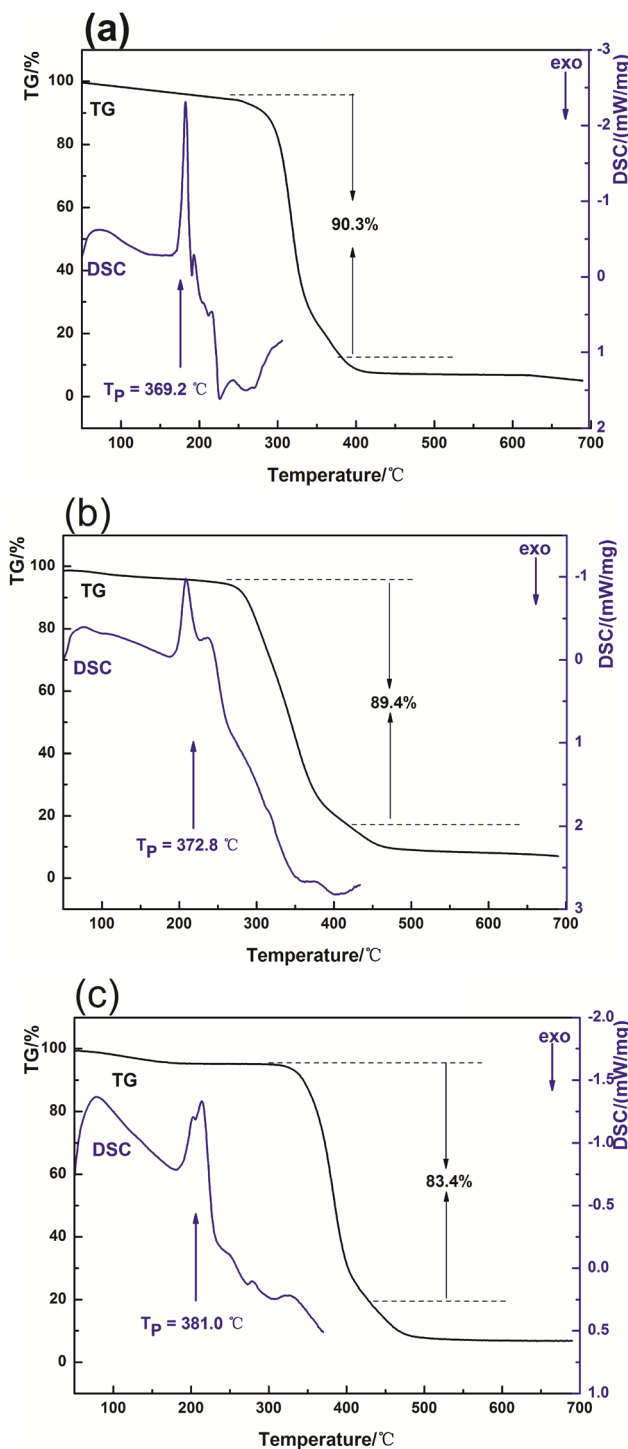


Figure 5: TGA and DSC thermograms of the films: (a) neat PVB, (b) LaB_6 -PVB, and (c) $\text{LaB}_6@ \text{TiO}_2$ -PVB.

nanocomposite films, it happened at around 320°C. Additionally, LaB_6 -PVB and $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films presented weight loss of 89.4% (Figure 5b) and 83.4% (Figure 5c), respectively. As a comparison, 0.3 wt% $\text{LaB}_6@ \text{TiO}_2$ -PVB film is not vulnerable to heat.

Endothermic peaks at 369.2°C (Figures 5a), 372.8°C (Figure 5b), and 381.0°C (Figure 5c) are apparently noticed in the DSC curves of neat PVB, LaB_6 -PVB, and $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films, respectively, and they might be originated from degradation of PVB matrix. In addition, when $\text{LaB}_6@ \text{TiO}_2$ NPs were added into neat PVB, the decomposition temperature of the matrix almost improved from 369.2 to 381.0°C, indicating an enhancement of thermal properties of the nanocomposite films. This phenomenon should be ascribed to the strong hydrogen bonding effect between the polar hydroxyl groups of PVB and the oxygen atoms of $\text{LaB}_6@ \text{TiO}_2$ NPs. Similar reports have been presented by Hemdana *et al.* [12] and Mallakpour and Dinari [26].

3.5 UV-Vis-NIR spectra

Transmission spectra of 0.30 wt% LaB_6 -PVB and 0.30 wt% $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films in Figure 6 were measured in a wavelength region between 380 and 2,000 nm. Both samples show high transmittance in visible region (380–780 nm), and strong absorption peaks in NIR region (780–1,500 nm), and their transmittance curves in the spectra are similar in appearance. Additionally, transmittance spectra of LaB_6 -PVB and $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films are almost the same with a maximum absorption in the region of 1,000–1,500 nm. In contrast, $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films illustrated better transmittance in visible region.

The peak transmittance of LaB_6 -PVB and $\text{LaB}_6@ \text{TiO}_2$ -PVB film were located at 605 and 669 nm, respectively.

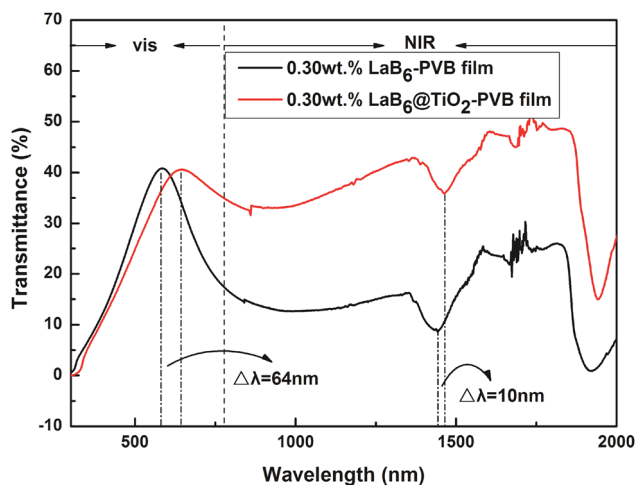


Figure 6: Transmittance spectra of composite films.

It means that an anatase TiO_2 shell enables the peak transmittance to redshift for around 64 nm in visible region, and the absorption peak in NIR region to shift slightly from 1,466 to 1,476 nm. It might be attributed to enlarged diameter of $\text{LaB}_6@ \text{TiO}_2$ NPs by TiO_2 shell. As has been reported in a previous research, enlarged diameter of a core-shell structure enabled absorption peak and transmission peak to redshift [12]. Thereby, it provides us a promising method to tune the wavelength range of the visible light transmission without any effect on the absorption of the NIR light.

To the best of our knowledge, films with high transmittance in the visible region and low thermal insulating properties have great application potential in automobile glasses. Based on the above results, the $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films could fulfill these two requirements and therefore, would be a potential candidate for transparent heat-insulated materials in some applications.

4 Conclusion

$\text{LaB}_6@ \text{TiO}_2$ NPs with a core-shell structure have been synthesized via a sol-gel-assisted hydrothermal method. XRD characterization shows that the NPs are composed of cubic LaB_6 and anatase TiO_2 . TEM results confirm that the NP is composed of a cubic LaB_6 core and an anatase TiO_2 shell. FT-IR and TGA analyses illustrate that the thermal stability of the composite was enhanced by the TiO_2 shell due to the interaction between the NPs and PVB matrix. In $\text{LaB}_6@ \text{TiO}_2$ -PVB nanocomposite films, an anatase TiO_2 shell apparently expands the wavelength range of the visible light transmission of the composite films. This work provides us a comparatively simple method to synthesis $\text{LaB}_6@ \text{TiO}_2$ NPs and a feasible way to produce composite films with high light transmittance and high thermal insulation.

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Author contributions: Hongbo Tang and Jian Zhou conducted experiments and analyzed data and then prepared

all the figures and wrote the main manuscript. Lihua Xiao and Yuchang Su oversaw the project and guided the writing of the manuscript. The manuscript was reviewed by all the authors.

Conflict of interest: Authors state no conflict of interest.

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