

## Research Article

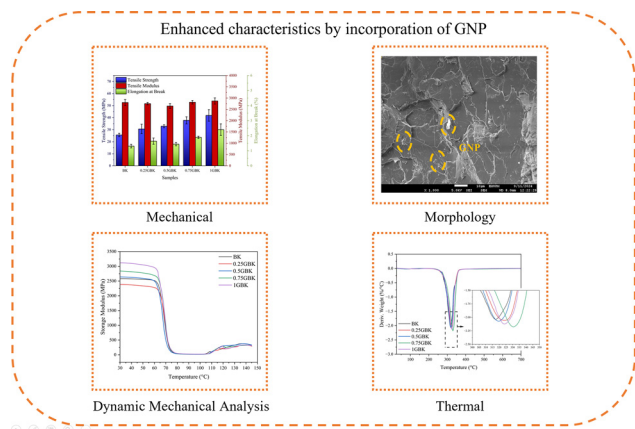
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# Enhancing natural fiber-reinforced biopolymer composites with graphene nanoplatelets: Mechanical, morphological, and thermal properties

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**Abstract:** Many researchers have focused on developing eco-sustainable materials in response to the environmental concerns associated with the use of synthetic polymers, such as natural fiber-reinforced polylactic acid (PLA) composites. In this study, short bamboo and kenaf fibers were employed to reinforce PLA composites. However, using these short natural fibers resulted in composites with inadequate characteristics. Therefore, this research examined the effects of incorporating a small quantity of graphene nanoplatelets into bamboo/kenaf fibers-reinforced PLA hybrid composites enhanced their poor characteristics. The research revealed that incorporating a small quantity of graphene nanoplatelets into composites enhanced their strength, with tensile strength of 41.86 MPa (1 GBK), flexural strength of 71.6712 MPa (0.25 GBK), impact strength of 3.63 kJ/m<sup>2</sup> (1 GBK), and hardness of 86.80 HD. Furthermore, the morphology demonstrates that graphene nanoplatelets are successfully dispersed in composites. Also, the dynamic mechanical investigation of the composites demonstrates that the use of



## Graphical abstract

graphene nanoplatelets resulted in a significant rise in the composites' storage modulus. Additionally, incorporating a small quantity of graphene nanoplatelets to the composites improves thermal stability, as seen by the highest degradation temperature of 330°C for 0.75GBK. The study aims to discover simple, low-cost ways to overcome the composite's inadequate properties, enabling them to be used in a wider variety of applications.

**Keywords:** graphene nanoplatelets, mechanical properties, morphological, natural fibers polylactic acid, thermal properties

## 1 Introduction

The development of eco-sustainable materials has become the focus of many researchers due to the environmental problems associated with the use of synthetic polymers [1]. In 2015, over 6,300 million metric tons of plastic trash were produced around the world. Approximately 9% was recycled, 12% was burned, and the rest was disposed of in landfills or spread across the environment. One of the

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eco-sustainable materials is biocomposites, which are composed of bioreinforcements such as natural fibers and biopolymers. According to Shebaz Ahmed *et al.* [2], the global biocomposites market is expected to develop at 9.59% per year, reaching USD 41 billion by 2025. Among biodegradable polymers, polylactic acid (PLA) has been known as the most promising biopolymer with the potential to change the common plastics in different applications and has remarkable qualities compared to synthetic polymers such as polystyrene and poly(ethylene terephthalate) [3–6]. PLA's use is promising because of its numerous advantages, including its biocompatibility, transparency, biodegradability, high elastic modulus, and strength [7]. However, PLA has a major disadvantage, namely its price, which can be overcome by reinforcing it with natural fiber [8]. These fibers could be utilized to lower costs due to their low price [9].

Natural fiber offers several benefits, including being abundant, low density, high performance, recyclable, strong mechanical qualities, renewable, adaptability, low specific mass, non-toxic, and biodegradability [10–14]. However, employing natural fiber has a significant disadvantage: the hydrophilic behavior of natural fibers leads to fibers to expand and form voids at the matrix-fiber interface, resulting in poor mechanical properties of composites [15]. These fibers are made of lignocellulose, which includes hydroxyl groups, causing them to be hydrophilic and unsuitable for usage in a hydrophobic polymer [16]. Consequently, this problem must be overcome with chemical treatments that enhance the compatibility of natural fiber and matrix [17]. Alkali treatment is an easy and cost-effective process for fiber modification. Alkali treatment roughens the fiber surface by removing lignin, contaminants, and hemicellulose, which improves fiber and matrix adherence and mechanical characteristics [18]. Hence, this research employs alkaline treatment on the fibers.

Short natural fibers are chosen in this study, owing to the mixing process in the internal mixer between PLA and fiber. The use of short fibers enables a more uniform distribution within the polymer, but their mechanical characteristics tend to be lower compared to long fiber composites [19]. Furthermore, bamboo and kenaf fibers were used in this study; these two fibers were used because of their plentiful availability in Malaysia [20,21]. Also, bamboo and kenaf fibers represent another eco-friendly and biodegradable fiber source, showing promise in composites and wide industry applications [22,23]. However, the previous study by Khan *et al.* [24] found that using bamboo and kenaf fibers in reinforced PLA composites made them less strong, with a tensile strength of 17.82 MPa and a flexural strength of 33.18 MPa. This meant that the composites were not as strong as plain PLA. As a result of these

concerns, the utilization of nanofiller material is being investigated. Nanofillers such as nanoplatelets, nanotubes, nanorods, and nanospheres are increasingly being added to polymers to enhance their mechanical characteristics and thermal stability [25].

In this study, graphene nanoplatelets are used. Graphene nanoplatelets represent a category of graphene materials, which consist of multiple layers of graphene that exhibit a platelet-like form and generally range from 5 to 50 nm in thickness [26]. Graphene nanoplatelets are chosen as nanofillers in this study due to their remarkable properties, which are high aspect ratio, light weight, thermal conductivity and electrical, enhanced mechanical performance, and low cost [27,28]. According to Gao *et al.* [29], using graphene nanoplatelets as fillers in PLA can rise the mechanical characteristics of the composites; however, a higher amount of graphene nanoplatelets can lead to a fall in the composites' mechanical strength. Furthermore, Husain *et al.* [30] also observed the same phenomenon: the use of graphene nanoplatelets in PLA improves tensile strength, but a higher amount results in a decline in mechanical strength. However, there are fewer studies conducted to analyze the influence of the incorporation of graphene nanoplatelets in the natural fiber-reinforced PLA composites.

Therefore, this research proposes a cost-effective and simple solution to the inadequate characteristics of bamboo/kenaf-reinforced PLA hybrid composites (BK composites) for industrial applications. Hence, this work comprehensively investigates the utilization of small amounts of graphene nanoplatelets as nanofillers in BK composites, a subject that has not received much attention in earlier studies. This research investigates several features of composites, including mechanical characteristics (tensile, flexural, impact, and hardness tests), dynamic mechanical analysis (DMA), morphology, and thermal properties (thermogravimetric analysis [TGA] and differential scanning calorimeter [DSC]). The study aims to provide solutions to overcome the inadequate properties of the composites, thereby expanding their application in many areas, specifically for industrial applications.

## 2 Materials and methods

### 2.1 Materials

PLA, Nature Works USA, Ingeo 2003D, with a density of 1.24 g/cm<sup>3</sup>, a 190°C of melting point, and 6 g/10 min of melt flow rate, was procured from Mecha Solve Engineering, Malaysia.

The graphene nanoplatelets, supplied by GO Advanced Solutions Sdn. Bhd., Selangor, Malaysia, had a diameter of 2–7  $\mu\text{m}$ , an 0.06–0.09 g/mL of apparent density, a thickness of 2–10 nm, a 107–102 S/m of electrical conductivity, a carbon content of >99%, a 120–150  $\text{m}^2/\text{g}$  of surface area, and less than 2% water content. The analytical reagent NaOH (R&M Chemicals, Malaysia) was provided by Evergreen Engineering & Resources, Selangor, Malaysia. Bamboo culms (*Schizostachyum brachycladum*) were sourced from Kelantan, Malaysia, while kenaf fibers (*Hibiscus cannabinus*) were procured from the National Kenaf and Tobacco Board in Kelantan, Malaysia.

## 2.2 Methods

### 2.2.1 Bamboo and kenaf fibers preparation

The preparation of bamboo and kenaf fibers, along with the alkali treatment, was conducted in accordance with established methodologies from prior research [31]. The bamboo culms were processed by being cut into strips using a knife, subsequently cleaned, and then subjected to sun drying. Following the drying process, the bamboo strips were cut into tiny pieces utilizing an electric bench saw (LB1200F, Makita Corp, Japan). Then, the tiny pieces underwent crushing utilizing a crusher (9FQ-350, Zhengzhou Shenwu New Mechanical and Electrical Co., Ltd., China). Initially, the kenaf fiber was processed by cutting it into shorter lengths with scissors to facilitate the subsequent pulverization process. Next, both of the fibers underwent a crushing process with a pulverizer fitted with a sieve cassette of 0.5 mm (Pulverisette 19 universal cutting mill, FRITSCH GmbH, Germany). Both fibers underwent an alkaline treatment by being soaked in a 5% w/w sodium hydroxide (NaOH) solution for a duration of 48 h. After the treatment, the fibers underwent a washing process with distilled water until a pH of 7 was reached, followed by oven drying at 50°C for a duration of 30 h. The particle size of both fibers was recorded using a particle size analyzer (Metasizer 2000, Malvern Instruments Ltd., UK) to ascertain their precise particle size. Figure 1 illustrates the dimensions of fiber particles.

### 2.2.2 Fabrication of composite samples

PLA matrix, bamboo and kenaf fibers, and graphene nanoplatelets filler were mixed with an internal mixer (50 EHT, Brabender GmbH, Germany) for 10 min at 190°C and 50 rpm. The materials were blended with dissimilar ratios

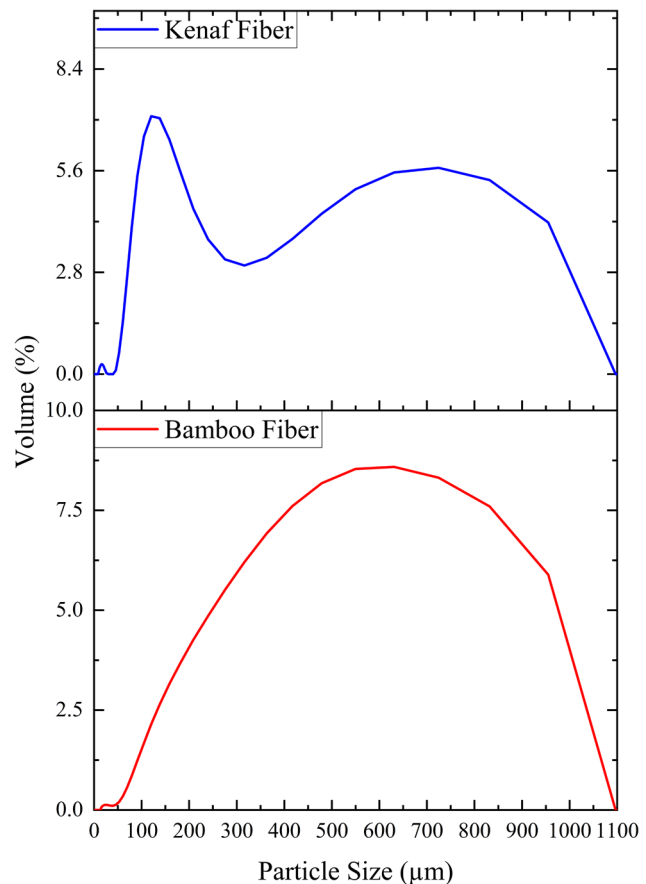


Figure 1: Particle size of bamboo and kenaf fibers.

of graphene nanoplatelet filler, specifically 0, 0.25, 0.5, 0.75, and 1 phr (per hundred resin) and were labeled BK, 0.25 GBK, 0.5 GBK, 0.75 GBK, and 1 GBK, respectively. The composite sheets were created by compression molding the blended materials using a Technopress-40HC-B from Technovation, Malaysia. The procedure used a 150 mm  $\times$  150 mm  $\times$  3 mm mold, 170°C temperature, 60 bar pressure, a 10 min preheating, a 3-vent cycle, 13 min of full press, and 10 min of cooling.

## 3 Characterizations

### 3.1 Tensile test

The composite samples underwent tensile testing utilizing the Universal Testing Machine (UTM) INSTRON 3366 to determine tensile behavior, in accordance with ASTM D638-14 [32]. The testing procedure involved applying a cell load weight of 10 kN, conducted at room temperature. The crosshead speed was set to 2 mm/min, and each sample

underwent five repetitions. Prior to testing, the dimensions of all composites were determined with digital calipers (CD-8 C, Mitutoyo Corporation, Japan). The samples were subsequently oven-dried for a duration of 12 h at 50°C.

### 3.2 Flexural test

The samples underwent a three-point flexural test utilizing the UTM INSTRON 3366 to determine flexural characteristics, in accordance with ASTM D790-17 [33]. The testing procedure involved applying a load weight of 10 kN at room temperature, with a 48 mm span length (maintaining a 16:1 ratio relative to the thickness of the sample), utilizing a 1 mm/min speed of crosshead, and conducting five duplications for each sample. Prior to testing, the dimensions of all composites were recorded with digital calipers (CD-8 C, Mitutoyo Corporation, Japan). Subsequently, the samples were subjected to drying at 50°C for a duration of 12 h in an oven.

### 3.3 Impact test

The composite samples underwent an Izod Pendulum Impact test using the INSTRON CEAST 9050 in order to evaluate their impact properties, as per ASTM D256 [34]. The experiment was conducted using a 0.5 J hammer under ambient temperature conditions, with five repetitions performed for each sample. Prior to conducting the tests, the samples were notched using an INSTRON CEAST manual notching machine with a depth and angle according to the standard. The thickness and width of all composites were recorded with a digital micrometer. Also, the composites were kept at 50°C in an oven for 12 h before test.

### 3.4 Hardness test

The hardness of the composites was measured with a digital shore type D durometer with a measuring range of 0–100 HD, according to ASTM D2240-15 [35]. The samples were placed on a flat surface before testing, then the instrument was pressed onto the samples for 1 s, and the instrument reading was recorded. The sample measurements were repeated at five points on the surface. The sample hardness is calculated as the average of the instrument readings.

## 3.5 Morphological of the composites

The sample's morphology was observed using a JSM-7600F FESEM (JEOL Ltd., Japan) at 5 kV at magnifications of 200×, 500×, and 1,000×. Before being examined, the composites were coated to lower the electron charge with platinum for 2 min with a coater (JEC-3000FC, JEOL Ltd., Japan).

## 3.6 DMA

DMA was utilized using the DMA Q800 V20.24 Build 43 (TA Instruments Co., USA). The test was carried out with a sample measuring 17.5 mm × 13 mm × 2.5 mm with a single cantilever clamp at 1 Hz frequency, a temperature range of 30–145°C, and a 5°C/min heating rate.

## 3.7 TGA and derivative thermogravimetric (DTG)

TGA and DTG were utilized using TGA Q500 V20.13 Build 39 (TA Instruments Co., USA). The measurement was carried out with about 6 mg composite samples and heated from 25 to 700°C with 50 mL/min of nitrogen flow rate and 10°C/min of heating rate.

## 3.8 DSC

DSC was carried out using DSC Q20 V24.11 Build 124 (TA Instruments Co., USA). About 11 mg of composite samples was heated using a temperature range of 25–200°C, a 50 mL/min nitrogen flow rate, and a heating rate of 10°C/min.

## 3.9 Statistical analysis

The data from experimental results were subjected to an analysis of variance with SPSS software. The Tukey test was implemented to compare means at a significance level of 0.05 ( $p \leq 0.05$ ). Statistically significant differences are represented by the use of different letters in the figures.

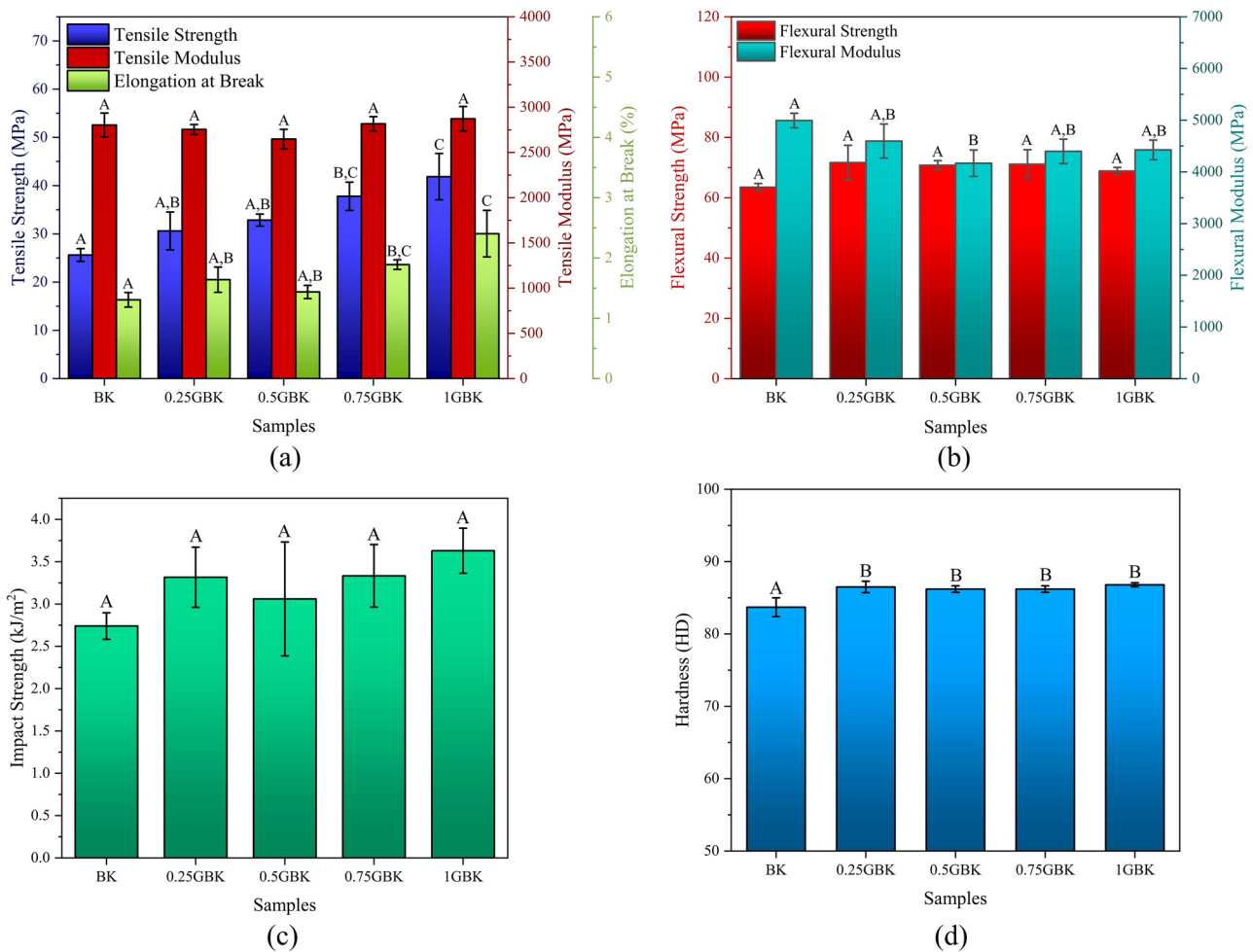
# 4 Results and discussion

## 4.1 Tensile properties

The tensile strength of BK composites improved as the graphene nanoplatelets loading increased, with 1 GBK

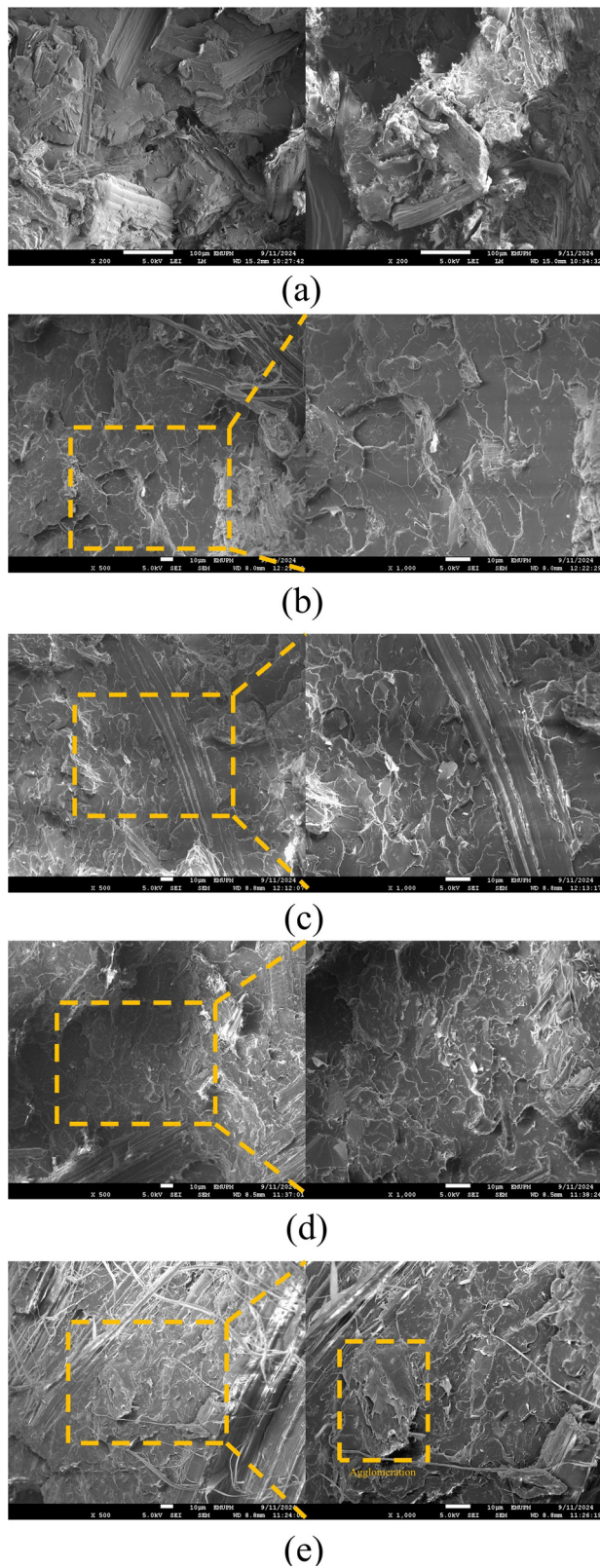
exhibiting the highest tensile strength at 41.86 MPa, a significant increase of 63.39% compared to BK (25.62 MPa), as depicted in Figure 2(a). From Figure 3(a), it can be noticed that, in BK samples, the bamboo and kenaf fibers were agglomerated, making the lower mechanical strength compared to the neat PLA, in which, according to Nazrin *et al.* [36], the tensile strength of PLA is 49.08 MPa. Furthermore, Figure 3(a) shows not only agglomerated fiber, but the poor interfacial bonding between fibers and PLA also found resulting small gap between fibers and matrix, which also the reason of the low tensile characteristics of the composites because of the hydrophilic characteristic of the fibers and hydrophobic PLA matrix [37,38]. However, the increase in tensile strength of the graphene nanoplatelet-filled composites was due to the very high intrinsic mechanical behavior and high aspect ratio of graphene nanoplatelets [39]. The increases in tensile characteristics were also confirmed by the Scaffaro *et al.* [40] study, which

stated that the use of graphene nanoplatelet filler rises the tensile strength of the Posidonia flour-reinforced PLA composites. The graphene nanoplatelets' well dispersion in the composites also becomes the reason for the increase in the mechanical characteristics, as shown in Figure 3. Additionally, the rise in samples' elongation at break as the graphene nanoplatelets' amount rose, from 1.31% (BK) to 2.40% (1 GBK), can be explained by the rougher surface of the composites along with the rise in the graphene nanoplatelet filler, as shown in Figure 3(b)–(e). The rougher surface occurred due to the crack growth barriers of the composites since the strong adhesion between graphene nanoplatelets and PLA matrix led to a tortuous crack, which is evaluated in Section 4.5, and led to the increase in elongation at break. On the other hand, incorporating a small quantity of graphene nanoplatelet filler into the composites had no significant impact on the samples' tensile modulus, resulting in a slight difference in the composite sample stiffness.



**Figure 2:** (a) Tensile characteristics; (b) flexural strength and modulus; (c) impact strength; and (d) hardness of the composite's samples.





**Figure 3:** Tensile fracture image of (a) BK, (b) 0.25 GBK, (c) 0.5 GBK, (d) 0.75 GBK, and (e) 1 GBK.

## 4.2 Flexural properties

Flexural testing is a destructive test that evaluates the force required to bend a beam under three-point loading settings [41]. The incorporation of graphene nanoplatelets into the composites results in a slight rise in flexural strength, with values 71.67 MPa (0.25 GBK), 70.77 MPa (0.5 GBK), 71.12 MPa (0.75 GBK), and 68.84 MPa (1 GBK) compared to 63.48 MPa (BK), as depicted in Figure 2(b). The increased flexural strength is due to the high stiffness of graphene nanoplatelets, which have a high aspect ratio, as well as improved bending between components in the composites [42]. However, the flexural strength of the 1 GBK sample is slightly lower than that of other graphene nanoplatelet-filled composites. This was expected because the graphene nanoplatelets in the composite sample are agglomerated. This discussion is confirmed by Giner-Grau *et al.* [43], who stated that carbon particles agglomerated due to van der Waals interactions, which contact with other carbon-based entities rather than polymer chains and reduce reinforcing potential in the composites. On the other hand, the flexural modulus of the composites slightly declines with the incorporation of graphene nanoplatelets in the composites, which shows that the stiffness of the material reduces. This phenomenon was expected due to the fact that after rotary relaxation, graphene nanoplatelets can interact by direct contact or by bridging through polymer chains [44].

## 4.3 Impact properties

Impact testing was conducted on the composite samples to evaluate their capacity for absorbing immediate force [41]. The effect of graphene nanoplatelets addition on the BK composites' impact strength is depicted in Figure 2(c). It was noticed that after filling the composites with graphene nanoplatelets, the impact strength was increased compared to BK samples, representing  $3.32 \text{ kJ/m}^2$  (0.25 GBK),  $3.06 \text{ kJ/m}^2$  (0.5 GBK),  $3.33 \text{ kJ/m}^2$  (0.75 GBK), and  $3.63 \text{ kJ/m}^2$  (1 GBK). As previously discussed in the section on tensile properties (see Section 4.1) and explained in Section 4.5, the tortuous crack that occurred in the graphene nanoplatelets-filled composites was due to the toughening effect of the graphene nanoplatelets present in the composites [45]. This phenomenon led to an increase in impact strength for the composites. Furthermore, according to Al-Maqdasi *et al.* [46], the increases in impact strength are due to the relatively good dispersion of the graphene nanoplatelet filler.

#### 4.4 Hardness properties

The impact of the addition of graphene nanoplatelets to BK composites on their hardness is depicted in Figure 2(d). By testing the hardness of the composites, it was found that adding graphene nanoplatelet filler to the BK composites made them harder. The hardest composites had a value of 86.80 HD, which is 3.70% higher than the BK samples. However, it is worth noting that these increments in graphene nanoplatelets loading have no significant effect on the composites' hardness values, which are 86.50 HD (0.25 GBK), 86.20 HD (0.5 GBK), 86.20 HD (0.75 GBK), and 86.80 HD (1 GBK). The increases in the hardness of the composites are due to the enhanced interfacial bonding on the graphene nanoplatelets-filled composites, which also increases other mechanical properties, including flexural, tensile, and impact strength. It is supported by the Batakiev *et al.* [47] study, who reported that using graphene nanoplatelets on PLA composites improves their hardness owing to the greater degree of interfacial bonding, resulting in improved load transmission between graphene nanoplatelets and PLA. Furthermore, the toughening effects of graphene nanoplatelets in composites, which are explained in the impact properties, also account for the increased hardness.

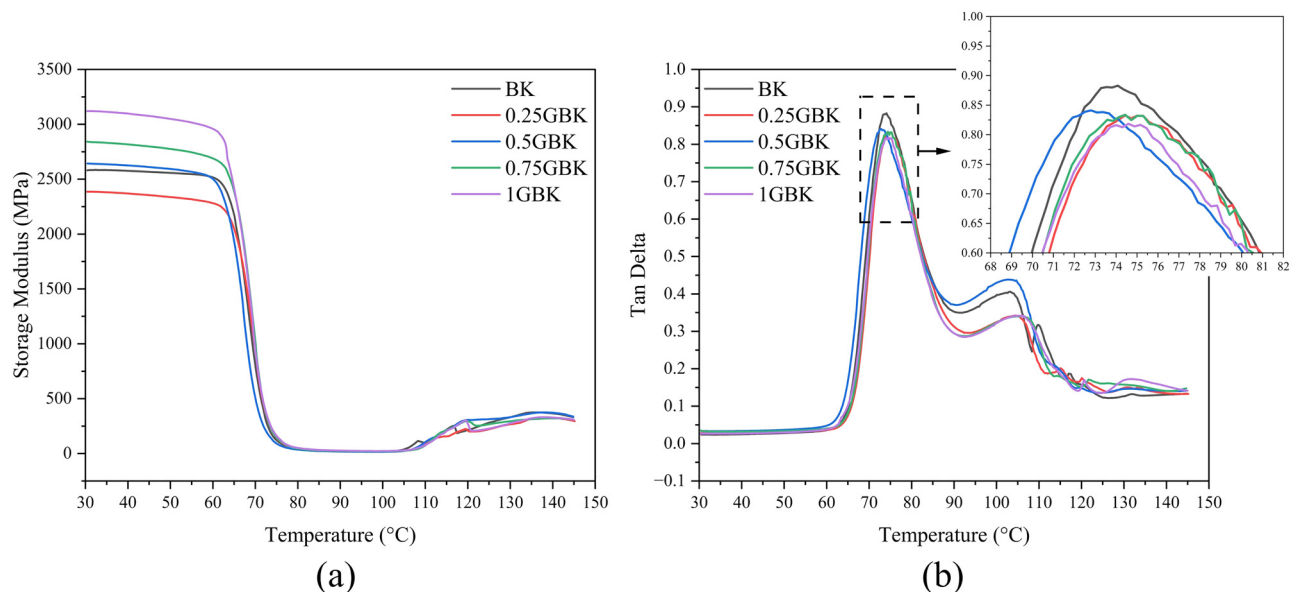
#### 4.5 Dispersion of graphene nanoplatelet filler in bamboo/kenaf fiber PLA hybrid composites

The dispersion analyses of graphene nanoplatelet-filled BK composites were analyzed using field emission scanning

electron microscope, using the tensile fracture of the composites, as depicted in Figure 3(a)–(e). The dispersion of the graphene nanoplatelets and bamboo/kenaf fiber in the composites is important due to the dispersion of this filler can influence the properties of the composites. In the Figures 3(b)–(e), it can be noticed clearly that the graphene nanoplatelets presence in the composites, which due to the graphene nanoplatelets filler in the composites have survived the high shearing during melt mixing, as confirmed by Kashi *et al.* [48] study. The graphene nanoplatelet filler is also well dispersed in all the samples, which can occur due to the hydrophobic graphene nanoplatelets having good adhesion with the hydrophobic PLA matrix. The increases in the graphene nanoplatelets filler also make the fracture surface rougher, which confirms that the graphene nanoplatelets and PLA matrix have strong bonding. This statement is confirmed by the Ab Ghani *et al.* [44] study, which stated that because of the good graphene nanoplatelet dispersion, the PLA matrix surface condition became rougher with the rise in graphene nanoplatelet loading. However, in the 1GBK sample, graphene nanoplatelet agglomeration begins to develop, as predicted, because of the van der Waals contact between the graphene nanoplatelets, which results in graphene nanoplatelet agglomeration in the composites, as explained in the flexural section.

#### 4.6 DMA

The effect of the addition of graphene nanoplatelets to BK composites on their storage modulus and tan delta is depicted in Figure 4. The DMA provides reliable



**Figure 4:** (a) Storage modulus and (b) Tan delta of the samples.

**Table 1:** DMA data of the samples

Sample code	Storage modulus at 30°C (MPa)	$T_g$ (°C)/storage modulus	$T_g$ (°C)/tan delta
BK	2580.42	65.54	74.08
0.25 GBK	2385.89	66.36	74.42
0.5 GBK	2642.87	65.18	72.80
0.75 GBK	2842.58	66.21	74.43
1 GBK	3119.76	65.23	74.59

information on the relaxation behavior of materials under varied temperature and frequency limitations [49].

Table 1 shows that the storage modulus among the graphene nanoplatelets-filled composites rises along with the increase in the graphene nanoplatelets loading, showing 2385.89 MPa (0.25 GBK), 2642.87 MPa (0.5 GBK), 2842.58 MPa (0.75 GBK), and 3119.76 MPa (1 GBK). This rise in storage modulus was attributed to the improved interfacial bonding, which resulted in better adhesion strength of the composites [50]. Additionally, these graphene nanoplatelet layers allow efficient stress transfer in the composites [49]. This phenomenon supported the increases in mechanical strength, which was explained in the previous section. Furthermore, according to a study by Liu *et al.* [51], the increases in storage modulus also occurred due to the higher modulus of graphene in the composites. On the other hand, in terms of glass transition temperature ( $T_g$ ), the temperature showed very few differences between all samples, which is in line with the Gao *et al.* [29] study, who stated that the use of graphene nanoplatelets did not significantly affect the mobility of the PLA matrix. Furthermore, this phenomenon was also confirmed by the Valapa

*et al.* [52] study, who stated that the reinforcement of graphene did not lead to the formation of short-chain PLA molecules. However, in the lowest loading of GNP (0.25 GBK), samples exhibit a decrease in storage modulus compared to the BK samples. This decrease is anticipated as a result of the weak chemical bond between graphene and the polymer matrix, along with the stress concentration within the composites [53,54].

## 4.7 TGA and derivative (DTG)

The thermal stability studies through TGA and DTG are represented in Figure 5. It is evident that the composites underwent two stages of degradation. The first stage of degradation happens at 25–200°C and is linked to the bamboo/kenaf fiber in the composites. This is because water evaporates along with OH groups contained in the cellulose structure [55]. However, the moisture content in the first stage of degradation is reduced along with increases in graphene nanoplatelets loading in the composites, as can be seen in Table 2. This phenomenon occurs because the presence of graphene nanoplatelets creates a tortuous path that allows water to absorb in the composites [42]. The second-stage thermal degradation process began at a temperature slightly above 225°C. Table 2 represents the temperatures corresponding to 10, 20, and 50% weight reductions ( $T_{10\%}$ ,  $T_{20\%}$ , and  $T_{50\%}$ ), as well as the onset temperature ( $T_{onset}$ ) and the peak temperature of the maximum weight loss ( $T_{max}$ ) for the composites. It is noticeable that the temperature rose with the incorporation of graphene

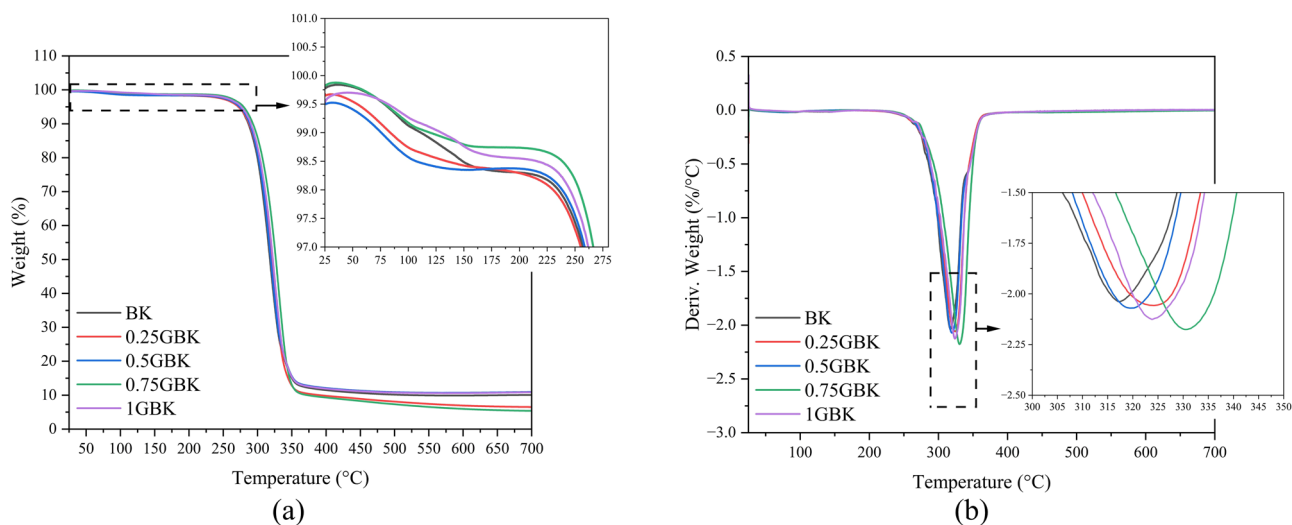
**Figure 5:** (a) TGA and (b) DTG curve of the samples.



Table 2: TGA and DSC data of the samples

Sample code	TGA					DTG	DSC		
	$T_{10}$ (°C)	$T_{20}$ (°C)	$T_{50}$ (°C)	$T_{onset}$ (°C)	Moisture content (%)		$T_g$ (°C)	$T_c$ (°C)	$T_m$ (°C)
BK	286.62	299.87	318.11	294.16	1.474	317.39	60.64	100.64	150.85
0.25 GBK	287.71	301.11	320.56	296.87	1.294	324.23	60.83	102.91	151.59
0.5 GBK	288.69	301.22	319.46	295.68	1.156	319.46	61.33	105.45	150.03
0.75 GBK	293.27	307.05	326.69	303.98	1.093	330.54	61.22	106.97	151.69
1 GBK	290.38	303.30	322.44	299.10	0.983	323.91	60.89	104.59	151.17

nanoplatelets compared to BK samples, which led to a rise in the thermal stability of the composites. The shape of graphene nanoplatelets creates a shielding effect that stops volatile decomposition products within the composites, which makes them more thermally stable [39]. Additionally, Cai *et al.* [50]

stated that the added graphene nanoplatelets could act as weight transfer barriers against the volatile pyrolyzed products in the PLA matrix, thereby delaying the thermal degradation of the composites. However, in the 1 GBK samples, a slight decrease in the temperature is noticed. This

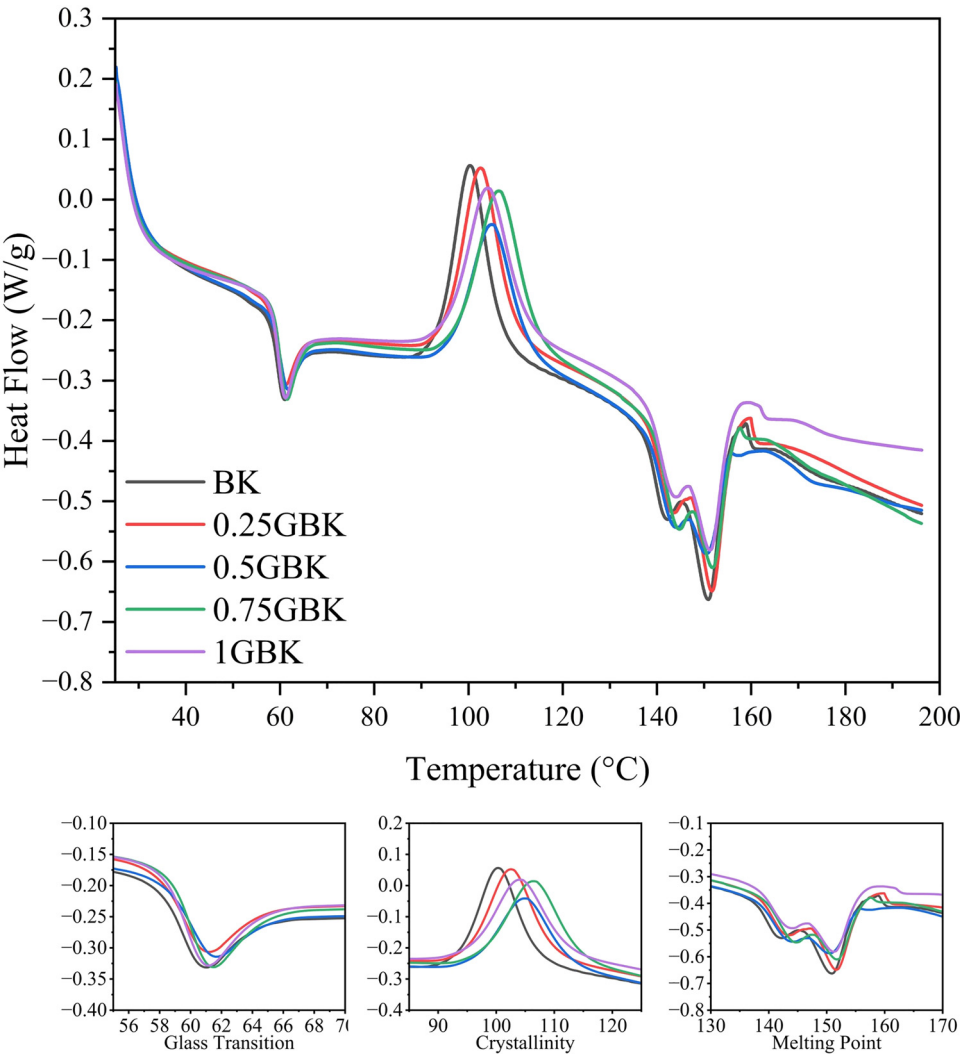


Figure 6: DSC curve of the samples.

phenomenon occurred as expected due to the agglomeration formation of graphene nanoplatelets in 1 GBK samples, which is shown in Figure 3(e).

## 4.8 DSC

The utilization of a low quantity of graphene nanoplatelets in the BK composites changed their thermal characteristic, including their glass transition temperature ( $T_g$ ), crystallization temperature ( $T_c$ ), and melting temperature ( $T_m$ ). These properties were found using DSC analyses, as shown in Figure 6.

Table 2 shows that adding graphene nanoplatelets did not significantly affect the  $T_m$  and  $T_g$  for all the composite samples. These temperatures stayed the same at about 151 and 61°C, respectively. This was also confirmed by Gao *et al.* [29], who stated that adding graphene nanoplatelets did not have a big effect on the  $T_m$  and  $T_g$ . Furthermore, the phenomenon of a small difference in  $T_g$  between all the samples in the DSC analysis is consistent with the  $T_g$  obtained by DMA, which exhibits the same behavior. On the other hand, it is noticeable that in the melting region there are two peaks, which can be attributed to the lamellar thickness and crystal morphology of PLA [56]. However, the addition of graphene nanoplatelets to the composites leads to a rise in  $T_c$ , but in the 1 GBK sample, it decreased, which is attributed to the nucleating effect of graphene nanoplatelets [57]. In addition, the  $T_c$  of 1 GBK samples behaved similarly, with the slight loss of thermal stability, which was expected because of the graphene nanoplatelet agglomeration on the composites.

## 5 Conclusions

This study looked into the utilization of a small quantity of graphene nanoplatelets in BK composites to improve their poor characteristics. This study found that incorporating a low quantity of graphene nanoplatelets into the composites enhanced their strength. The composites had a tensile strength of 41.86 MPa (1 GBK), flexural strength of 71.67 MPa (0.25 GBK), impact strength of 3.63 kJ/m<sup>2</sup> (1 GBK), and hardness of 86.80 HD (1 GBK). Furthermore, the morphology indicates that graphene nanoplatelets are well dispersed in the composites. Additionally, the DMA of the composites reveals that the addition of graphene nanoplatelets solely influences the storage modulus, leading to a significant rise in the composites' storage modulus, while

the change in  $T_g$  is negligible. The utilization of a small quantity of graphene nanoplatelets did not affect the composites'  $T_g$  or  $T_m$ . However, incorporating a small quantity of graphene nanoplatelets into the composites improves their thermal stability, as evidenced by the highest degradation temperature of 330°C for 1 GBK. Additionally, the incorporation of graphene nanoplatelets raises the  $T_c$  in the composite materials. The utilization of a small quantity of graphene nanoplatelets can enhance composites' insufficient qualities, allowing them to be utilized in a wide range of industrial applications.

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