

## Review Article

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# A review of application, modification, and prospect of melamine foam

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**Abstract:** Melamine foam (MF), a promising development in light materials, finds application in construction, agriculture, aviation, transportation, electronic message, and other fields due to its excellent thermal insulation performance, sound absorption and noise reduction capabilities, safety and health benefits, as well as easy processing. However, its own shortcomings such as hardness, poor toughness, fragility, and slag removal greatly limit its application scope. In this review, a survey of the literature from two aspects of toughening of melamine resin and regulation of MF pore structure are reviewed to explore the research progress of toughening modification of MF. The principle, merit, and demerit of different modification methods are analyzed. In addition, owing to the extensive literature available, this article also summarizes the representative achievements of the nanotechnology modification of MF derivatives (carbon foam and carbon aerogel). Eventually, based on an assessment of current application status for both MF and its derivatives while considering existing challenges in their modification processes using nanotechnology approaches, we discuss future prospects for their application.

**Keywords:** melamine foam, melamine resin, nanotechnology modification, carbon foam, carbon aerogel

## 1 Introduction

Melamine foam (MF) is an inherently flame-retardant foam derived from melamine formaldehyde resin by curing and foaming process [1]. In addition to its excellent flame-retardant property [2], it also exhibits good sound absorption [3], low density [4], high porosity [5], heat resistance, and preservation [6]. Consequently, MF finds applications in the development of oil–water separation materials [7,8], supercapacitors [9,10], sound absorption materials [11,12], vacuum insulated panels [13], *etc.* The performance characteristics and service range of MF are shown in Table 1. At present, domestic enterprises have made great progress in the development and application of MF, such as Puyang Greencos New Material Technology Co., Ltd, whose MF products have important application value in civil, industrial, construction, transportation, aviation, military, and other fields (Figure 1) [14–16].

### 1.1 Sound-absorbing application of MF

The MF possesses a fully open three-dimensional mesh structure (Figure 2), with a length to diameter ratio ( $L/D$ ) of the mesh ranging from approximately 10–20. Its high opening rate characteristics ( $\geq 99\%$ ) enable sound waves to effectively penetrate the deep layer of the foam body, where they are rapidly dissipated through vibration of the MF gridding. As a result, reflected sound waves are effectively eliminated, leading to enhanced sound absorption performance [12,29]. Due to its excellent sound-absorbing performance, MF has been successfully applied to the sound absorbing materials for military and civilian use [30].

Yang *et al.* [31] investigated the sound insulation of phenolic resin modified MF. The experimental results illustrated that the sound insulation property of MF modified by phenolic resin was remarkably improved compared to that without phenolic resin. Li *et al.* [32] developed a thin and lightweight sound absorber by utilizing MF's excellent sound absorption performance. The research results showed

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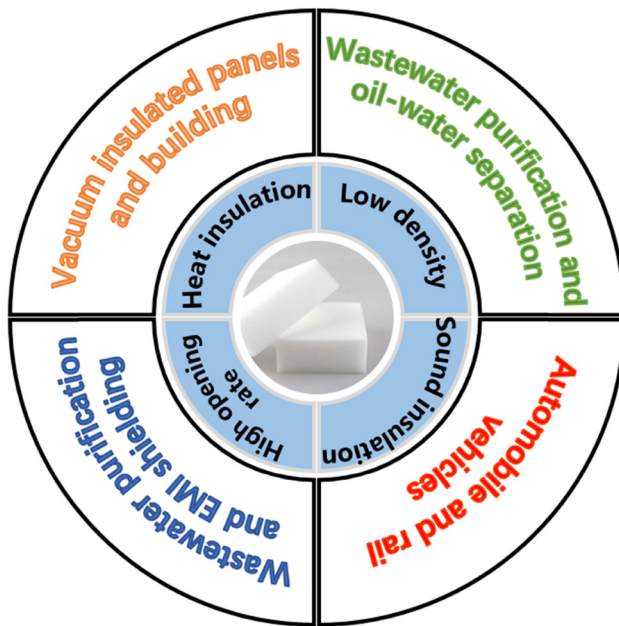
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**Table 1:** Performance characteristics and service range of MF

| Project          | Performance                 | Service scope                                                                         | Reference |
|------------------|-----------------------------|---------------------------------------------------------------------------------------|-----------|
| Density          | Low (<8 kg/m <sup>3</sup> ) | oil–water separation material and building, <i>etc.</i>                               | [17–20]   |
| Sound insulation | Excellent                   | Rail vehicles, aircrafts cabin, and automobile, <i>etc.</i>                           | [3,21–23] |
| Opening rate     | High (99%)                  | oil–water separation material and EMI shielding, wastewater purification, <i>etc.</i> | [24–26]   |
| Heat insulation  | Eminent                     | Building and vacuum insulated panels, <i>etc.</i>                                     | [27,28]   |

**Figure 1:** Typical applications of MF prepared by Puyang Greencos New Material Technology Co., Ltd.

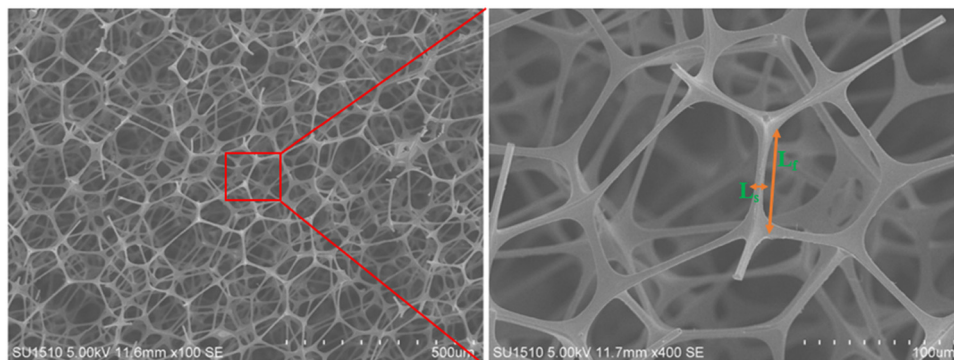
that incorporating a hollow perforated spherical structure with MF maintained effective sound absorption within the frequency range of 200–1,600 Hz, despite limited thickness constraints. These results also suggested that the proposed absorbers have great potential for use in noise control applications.

To investigate the sound absorption performance of MF and cotton, Li *et al.* [33] established a scaled down cylindrical section noise experimental device according to the objective conditions of the experiment and the application range of the simulation software. The experiment results showed that the noise reduction effect of MF with the same quality is about 1.42 times more than that of sound-absorbing cotton. In addition, MF can also be used as sound insulation panels of sports venues, lining of roll gate boxes, sound insulation materials of impellers of air conditioning and ventilation equipment, inner lining of fan hoods, *etc.* [34,35].

## 1.2 Oil–water separation application of MF

MF is characterized by high porosity, large surface area, low density, and low cost [4]. The inside of MF is composed of a large number of spheroid units, whose surface is a three-dimensional network skeleton ring of pentagon structure, which makes the surface water film tension of MF high after water absorption, so it has a strong water absorption ability [19,23]. Consequently, when hydrophobically modified, the high surface area enables fast oil absorption by MF and makes it an ideal material for oil–water separation applications [16,36].

Duan *et al.* [5] reported a simple method to prepare high-performance oil absorbents by loading ball-mill

**Figure 2:** Microscopic characterization of MF.

biochar (BMBC) and octadecylamine onto MF skeleton. The experimental results showed that BMBC successfully transformed the hydrophilic surface of MF into a hydrophobic surface, and endowed MF with higher oil absorption capacity (43–155 times dead weight) and excellent recyclability. Mu *et al.* [37] presented a superhydrophobic foam *via* coating the hydrolyzed alkylsilane on the skeleton of commercial MF. The study results (Figure 3) showed that the wettability of MF changed to superhydrophobic due to the presence of alkylsilane, which exhibited excellent oil absorption capacity and exceptional recyclability. Moreover, the modified MF could be readily used in industrial oil spill clean-up.

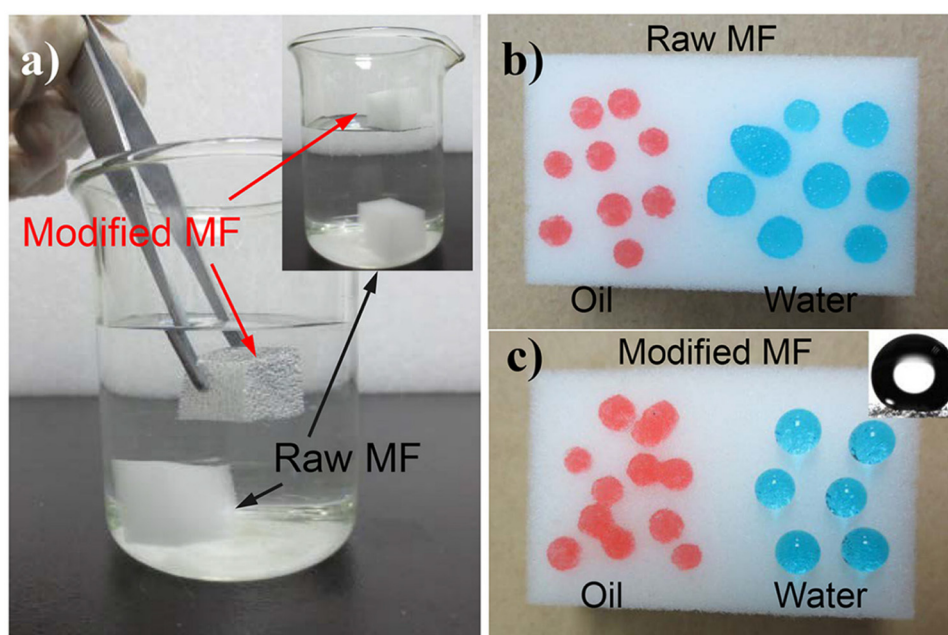
oil–water separation modification generally requires polymers with low surface energy (such as polysiloxane, fluoropolymer, or long alkyl) to cover the surface of melamine sponge through a series of reactions [38,39], or high temperature pyrolysis to remove the hydrophilic groups on the surface of melamine sponge to improve its hydrophobicity [36]. After hydrophobic modification, MF can absorb leaked oil or organic solvent.

### 1.3 Heat insulation application of MF

There are a lot of studies, which showed that MF has a high cross-linked three-dimensional network structure system to keep excellent thermal stability and aging resistance [40,41]. The research results show that MF starts to burn only when

the surface is in contact with an open flame. Once the combustion starts, the decompositions immediately produce a large amount of inert non-flammable gas, which slows down the combustion rate. At the same time, a dense coke layer is formed quickly on the surface of the burner, which effectively blocks the development of combustion to the deep layer, and automatically extinguishes after the open flame leaves [42,43]. MF can work at 200°C for a long time, without decomposition and deformation, and obvious decomposition only appears when the temperature increases above 350°C [13,43]. While for other forms, such as polyvinyl chloride, polyolefin, polystyrene, and polyurethane (PU), deformation and decomposition occur when the working temperature exceeds 120°C. The resistance to open flame ignition experiment of MF and the maximum temperature for long service time of some forms are shown in Figure 4.

Based on the exceptional flame retardant and heat insulation properties, MF exhibits significant potential for application in constructing energy conservation buildings and vacuum insulation panel (VIP) [13,47,48]. Li *et al.* [49] utilized a MF prepolymer derived from melamine and paraformaldehyde as the matrix material, cyclohexane as the foaming agent, dimethicone as the foam stabilizer, and hydrochloric acid as the catalyst. The MF rigid foam with high closed-cell content was prepared using an oven heating method. The prepared MF foams demonstrated great comprehensive performance with closed cell content of 83.5%, water absorption of 12.0%, compressive strength of 292 kPa, thermal conductivity of 0.033 W/(m K) and limiting oxygen index (LOI) of 36%. Compared to conventional organic foams, the as-prepared MF



**Figure 3:** Hydrophobic and oleophilic properties of (a) MF, (b) before treatment, and (c) after treatment [37].



| Project            | Maximum temperature for long service (°C) |
|--------------------|-------------------------------------------|
| MF                 | 200                                       |
| polyolefin         | 100                                       |
| polystyrene        | 60                                        |
| polyurethane       | 120                                       |
| polyvinyl chloride | 60                                        |

**Figure 4:** The resistance to open flame ignition experiment of MF and the maximum temperature for long service of some forms [13,44–46].

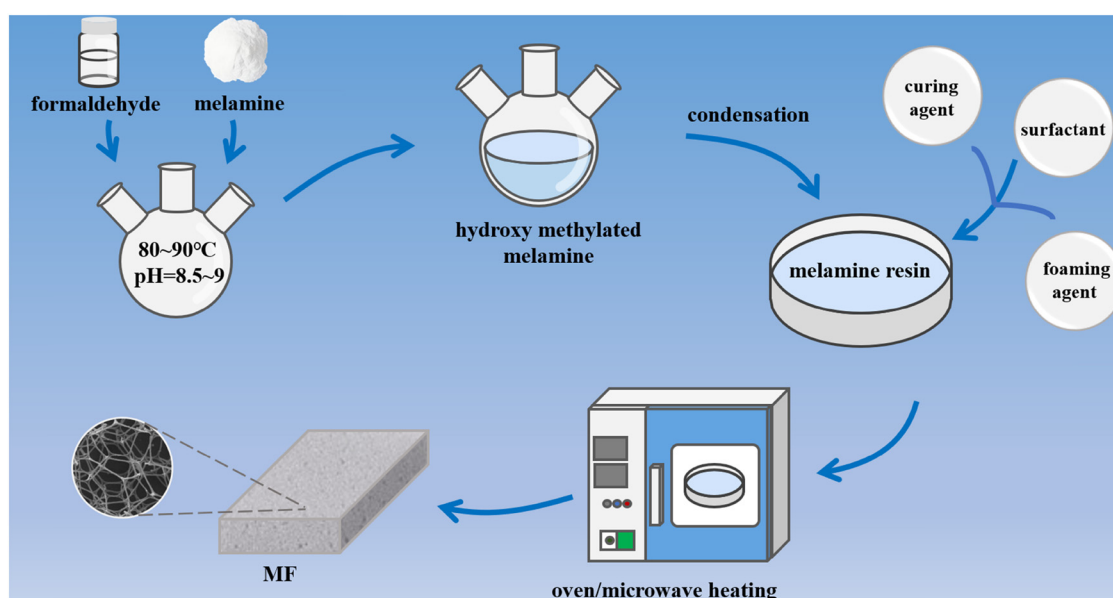
foams can be used as thermal insulation material for building exterior walls. Kavsek *et al.* [50] used pentane to blow melamine-formaldehyde resin emulsion and further catalyze and heat cure to prepare a new melamine-formaldehyde rigid foam material. Compared with other organic polymer foams, the produced MF foam has the characteristics of superior fire resistance, chemical stability, and low density, which made MF foam potentially useful in the manufacture of VIPs.

## 1.4 Preparation process and disadvantages of MF

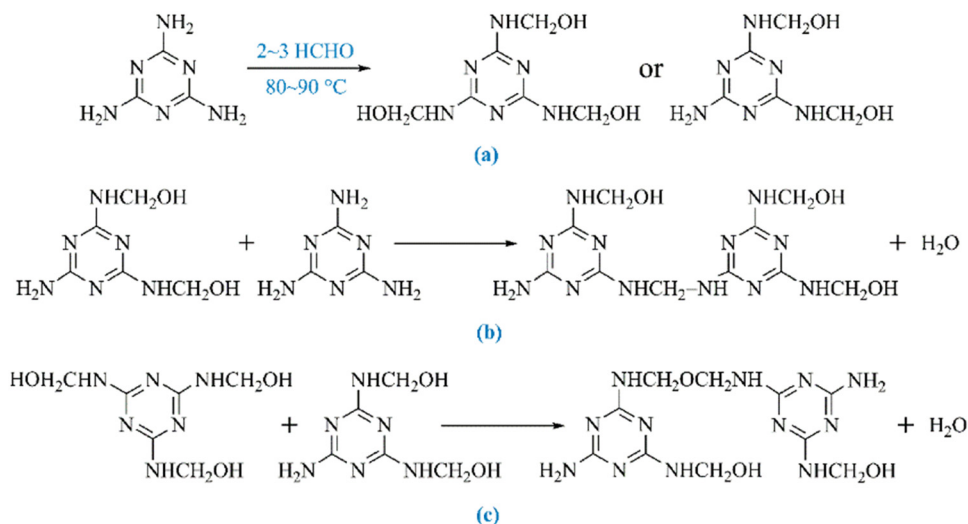
### 1.4.1 The preparation process of MF

The preparation of MF is usually divided into two processes: resin synthesis and curing foaming (Figure 5) [13,51]. The

synthesis stage of melamine resin (MR) can be divided into two steps. First, the melamine and formaldehyde are dissolved in a weakly alkaline medium ( $\text{pH} = 8.5\text{--}9.0$ ). Because melamine and formaldehyde will produce insoluble methylene melamine precipitation under acidic conditions, it is necessary to adjust the pH value of the solvent between 8.5 and 9.0 before the reaction, so as to ensure that the pH value of the reaction process is between 7.0 and 7.5. When the external temperature is in the range of  $80\text{--}90^\circ\text{C}$ , melamine and formaldehyde take place addition reaction to form hydroxymethyl melamine. Second, hydroxymethyl melamine is further poly-condensed in neutral or weakly alkaline media, resulting in low molecular polymer (MR) linked by dimethylene ether bond or methylene bond [52–54]. The reaction mechanism of MR is shown in Figure 6 [37]. The curing foaming process is to mix MR with emulsifier, curing agent, and foaming agent adequately and then the cross-linking



**Figure 5:** The production process of MFs.



**Figure 6:** The reaction mechanism of MR [13,51]. (a) Formation of methylol melamine. (b) Condensation of methylol of methylol melamine and amino of melamine. (c) Condensation of methylol of methylol melamine and one another.

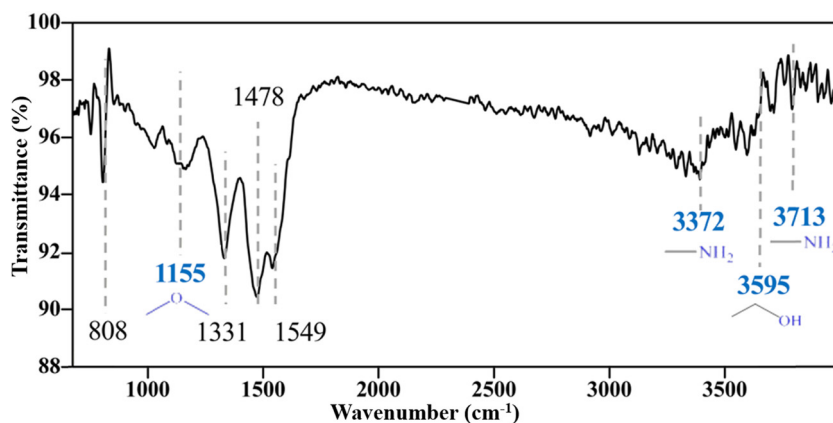
reaction occurs by microwave or oven heating to form polymer foam with three-dimensional network structure (MF) [55].

#### 1.4.2 Disadvantages and toughening modification prospect of MF

Although MF possesses high porosity, low density, high opening rate, and good flame retardancy, its inherent drawbacks such as hardness, poor toughness, fragility, and slag removal significantly limit its application scope [16]. The infrared spectrum of MF is shown in Figure 7. The wave numbers of 808, 1331, 1478, and 1549  $\text{cm}^{-1}$  correspond to the in-plane deformation of the triazine ring and the contraction and torsion of  $\text{C}=\text{N}$  in the triazine

ring [56]. It is well known that the reason for the high brittleness of MF is that its molecular structure contains a large number of rigid triazine rings and the flexible carbon chains between them are short [57]. Consequently, enhancing the toughness of MF has become a prominent research topic in recent years.

At present, most scholars have only explored the methods of toughening, strengthening, and improving the dimensional stability of MF; however, few experts have comprehensively summarized the causes of poor toughness and modified toughening methods. Generally speaking, the toughness of MF depends on both the properties of MR and foam microstructure. In this study, the methods of toughening modification of MF are described in detail from two aspects of toughening modification of MR and microstructure regulation of foam. Furthermore, the direction of



**Figure 7:** The infrared spectrum of MF [56].

toughening modification methods, and the application of MF products and their derivatives in the future are prospected.

## 2 Modification of MR

MR is a type of thermosetting resin wherein the properties are determined by the degree of curing [58]. The MR solution needs to be heated and cured in the process of use, so that the oligomers are gradually cross-linked to form a polymer with a three-dimensional network structure. If the MR is not cured completely, its adhesion, water resistance, chemical corrosion resistance, and other properties will be reduced [59]. The curing of MR is achieved by cross-linking methylene or dimethylene ether bonds, and its molecular structure is shown in Figure 8. At present, the toughening modification of MR mainly includes physical modification, flexible chain lengthening, and functional group end sealing [12,60,61]. The section will focus on these three approaches.

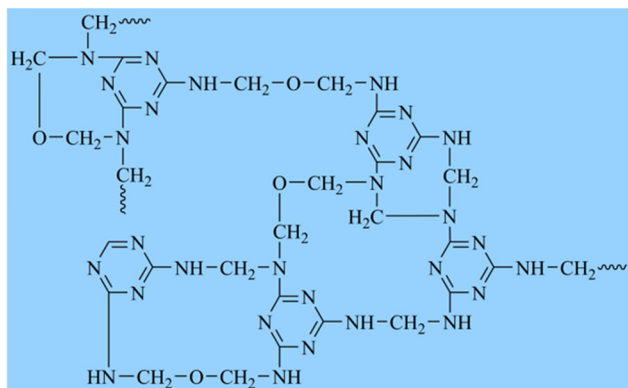


Figure 8: Molecular structure of MR [12,60,61].

## 2.1 Physical modification of MR modification

In the physical modification method, the modifier is combined with MR by blending method to prevent the triazine rings from getting close to each other, so as to realize the toughening of MR. The common physical modifiers mainly include nanoparticles, flexible polymers, and fiber materials [62–66].

### 2.1.1 Modification by adding nanoparticles

The addition of nanoparticles is a widely used approach to modify MR [67–70]. Nanoparticles have the characteristics of large specific surface area, a large number of unsaturated bonds on the surface, small size, and high surface activity [71,72]. When adding a small number of nanoparticles, the surface functional groups of MR react with these nanoparticles to generate an interfacial force that surpasses the van der Waals force. When the MR is subjected to external force, the nanoparticles can absorb part of the impact energy, so as to realize the toughening of the MR [63].

Based on the source, type, size, content, and shape, nanoparticles can reinforce various properties of the substrate MR (as summarized in Table 2). Therefore, nanomaterials can affect MR nanocomposites' final properties such as mechanical properties, curing behavior, formaldehyde emission, and flame retardancy even at low addition levels [15]. Commonly employed nanofillers include clay minerals, different oxides, carbon-based nanomaterials, graphite, cellulose nano-fiber, metals and metal alloys, *etc.* [61,73].

For instance, Li *et al.* [61] demonstrated that the incorporation of a dispersed peg-nano SiO<sub>2</sub> solution into the melamine/formaldehyde mixture could enhance the physical properties (such as curing time, free formaldehyde content, and elongation at break) of MR. The experimental

Table 2: Nanomaterials and their enhanced performance of MR

| Type of nanomaterials     |                        | Performance (enhanced)                                                                 | References |
|---------------------------|------------------------|----------------------------------------------------------------------------------------|------------|
| Mineral nanoparticles     | Montmorillonite        | Thermal stability, chemical resistance, processability                                 | [74,75]    |
|                           | Clay                   | Thermal stability, low cost, mechanical, <i>etc.</i>                                   | [76,77]    |
| Bio-based additive        | Cellulose nanocrystals | Mechanical strength, <i>etc.</i>                                                       | [78,79]    |
|                           | Carbon nanotubes       | Heat insulation performance, <i>etc.</i>                                               | [80,81]    |
| Metal-based nanoparticles | Fe                     | Fouling resistance, good wettability, <i>etc.</i>                                      | [82]       |
|                           | Ni                     | Specific capacitance value, <i>etc.</i>                                                | [83]       |
| Oxides nanoparticles      | SiO <sub>2</sub>       | Thermal stability, toughness, flame-retardant properties, impact strength, <i>etc.</i> | [84–86]    |
|                           | TiO <sub>2</sub>       | Thermal stability, smoke suppression, <i>etc.</i>                                      | [87,88]    |

findings suggested that the addition of dispersed nano-SiO<sub>2</sub> has the potential to improve toughness and other desirable characteristics in modified MR. However, excessive nanoparticle loading may result in uneven dispersion and agglomeration, thereby compromising its toughening effect.

### 2.1.2 Modification by flexible short chain polymers and fiber materials

Adding flexible short chain polymers and fiber material such as starch, carbon fiber (CF) [64], polyvinyl alcohol (PVA) [89,90], and PU [91] to MR is another common physical modification method. These modifiers contain numerous polar groups that can form hydrogen bonds or covalent bonds with MR to enhance its toughness. Lulu *et al.* [92] studied the effect of PVA on the properties of MR. The results showed that the blending fiber was endowed with effective flame-retardant property by the amino group and triazine ring group in melamine/PVA fibers, and the LOI reached 28–31. The best homogeneity of blended fibers was achieved when the solid content ratio between MR and PVA blend was 1:1. PVA could effectively improve the crystallinity, mechanical properties, and elasticity of MR.

Li *et al.* [64] prepared melamine-formaldehyde rigid closed-cell foams that were toughened by ethylene glycol (EG) and CF. Compared to unmodified MF rigid foams, these tough rigid closed-cell MF exhibited excellent pulverization rate, compressive strength, bending strength, cellular structure integrity, thermal insulation performance as well as flame resistance characteristics. It is worth mentioning that when the fiber is used as toughening agent, the viscosity of the resin will increase, and the mixing effect of the foaming agent will become worse in the foaming process, thus affecting the final performance of the foam. Fiber materials have been gradually phased out as toughening agents.

### 2.1.3 Summary of physical modification of MR

As a whole, when nanoparticles are added as modifiers, too much nanoparticles will result in agglomeration, and too little nanoparticles will limit the toughening modification effect, the solutions improve the dispersion and compatibility of nanoparticles by modifying nanoparticles in resin [93]. The modification of MR by adding flexible long chain polymer is another common physical modification method. The polar groups on the surface of flexible long chain polymer can form hydrogen bonds and covalent bonds with the surface of MR, so as to achieve the toughening

effect [94,95]. Moreover, the use of fiber material as toughening agent may affect the foaming effect of resin, which has been gradually eliminated [96]. The toughening by physical modification has the characteristics of simple, effective, and economical and practical, and has broad development space.

## 2.2 Functional group end-sealing method of MR modification

The functional group end-sealing method involves the reaction of modifying agents with melamine or hydroxymethyl melamine, resulting in the shielding of one amino group on the triazine ring to reduce cross-linking density. Small molecules with single functional groups, such as methanol, ethanol, aniline, and p-toluene sulfonamide, are often used as functional group shielding modifiers [65]. Fei *et al.* [97] prepared fragrance microcapsules by *in situ* polymerization of methanol-modified melamine-formaldehyde resin. FT-IR results showed that the essential oil was encapsulated by MR, and TG-DTG test showed that micro-encapsulation significantly improved the thermal stability of MR.

Furthermore, derivatives of melamine such as benzene melamine, aromatic melamine, and alkyl substituted melamine are also often used as modifiers. It is worth noting that the synthetic conditions of melamine derivatives are relatively complex and the yield is not high, so they cannot be commercialized on a large scale and are usually used in high-end markets such as aerospace. Therefore, the economical and suitable modifiers need to be further studied.

## 2.3 Flexible chain lengthening of MR modification

The high brittleness of MF is attributed to its molecular structure, which contains a large number of rigid triazine rings and short flexible carbon chains between them (as mentioned in Figure 5) [57]. Therefore, flexible chain lengthening is put forward to improve the flexibility of MR by increasing the length of the flexible chain between the triazine rings. Modifiers generally contain a variety of functional groups that can react with hydroxyl methyl groups or amino groups, and the distances between triazine rings are increased by molecular bonding. The flexible chain lengthening method is a common chemical modification idea. The flexible chain lengthening method makes the

molecular chains of MR have greater intramolecular deformability than that of methylene linkage, and reduces the crosslinking density of resin, so as to improve the flexibility of MF. PVA and allyl glycidyl ether are often used as multi-functional small molecule modifiers to increase the flexible chain length between the triazine rings [66].

Chen *et al.* [98] used PVA and benzoguanamine to modify MF resin, while investigating the enhanced tensile strength and flame retardant properties of the modified MR. Compared with the unmodified MR, the tensile strength and flame retardancy properties of the modified MR were improved. Yan *et al.* [12] employed allyl glycidyl ether to modify MR, and the modified resin was able to demonstrate a longer storage duration as compared to the conventional resins, and the storage time for the modified resin exceeded 90 days, and the free formaldehyde content was lower than 0.05 ppm. Additionally, the as-prepared foam exhibited excellent flame retardancy properties, low-volume water absorption, and lower thermal conductivity. Concurrently, the foam prepared based on the optimal conditions possessed a high compressive strength of 5.08 MPa, which was 17.5 times higher than that of the conventional MF. But, allyl glycidyl ether itself is toxic, so its application on MR is limited. The comparison of the performance of MF modified by allyl glycidyl ether and other previous studies are illustrated in Table 3.

In conclusion, both long-chain polymers and small-molecule compounds can improve the flexibility of MR by increasing the flexible chain length, and affect the heat resistance or flame retardancy of MR. However, common modifiers process toxicity, so this method is limited to incorporating modifications. Therefore, in order to effectively enhance the toughness of MF, molecular structure design should be prioritized.

## 2.4 A comprehensive analysis of MR modification methods

Overall, the three toughening modification methods of MR showed that physical modification can block the proximity

of triazine rings; flexible chain lengthening and functional group end-sealing method can change the molecular structure of the resin. All these modification techniques contribute to a reduction in crosslink density of MR; however, their individual toughening effects are limited. For instance, Li *et al.* [61] used polyethylene glycol (PEG) to modify nano-SiO<sub>2</sub> particles, and uniformly dispersed the modified solution into the melamine/formaldehyde mixture, thereby, the curing time and elongation at break of the MR were improved (Table 4). The addition of 0.42% PEG-modified SiO<sub>2</sub> reduced the glass transition temperature of the MR by 29.4°C, and thermogravimetric (TG) spectroscopy showed that the introduction of nano-SiO<sub>2</sub> into the MR hardly affected its thermal stability. These results also indicated that the addition of PEG-modified nano-SiO<sub>2</sub> can improve the elongation at break of MR. Therefore, further investigation is required to explore synergistic toughening approaches using a combination of different methods while ensuring flame retardancy and mechanical properties consistent with MF.

## 3 Regulation of MF pore structure

The pore structure directly affects the physical properties of MF [62]. During the foaming process, bubbles are generated and expanded, while the prepolymer gradually

**Table 4:** Effect of nano silica addition on the physical properties of MR [61]

| Amount of nano silica (%) | Cure time (s) | Elongation at break (%) |
|---------------------------|---------------|-------------------------|
| 0                         | 109 ± 3       | 5.35 ± 1.52             |
| 0.3                       | 94 ± 2        | 16.20 ± 2.61            |
| 0.36                      | 93 ± 4        | 13.00 ± 2.13            |
| 0.42                      | 95 ± 1        | 19.90 ± 2.86            |
| 0.48                      | 94 ± 2        | 9.45 ± 2.01             |

**Table 3:** Comparison of the performance of MF modified by allyl glycidyl ether and other studies [12]

| Project                               | Traditional MF | CF/EG | Dimethicone | MF modified by allyl glycidyl ether |
|---------------------------------------|----------------|-------|-------------|-------------------------------------|
| Apparent density (g/cm <sup>3</sup> ) | 0.040          | 0.078 | 0.063       | 0.250                               |
| Thermal conductivity (W/m K)          | 0.030          | 0.031 | 0.033       | 0.0256                              |
| Compressive strength (MPa)            | 0.23           | 0.44  | 0.292       | 5.08                                |
| Volumetric water absorption (%)       | 1.4            | 9.0   | 10          | 0.9                                 |
| LOI (%)                               | 39.4           | 39.6  | 39          | 38.9                                |

undergoes cross-linking to increase viscosity. In the later stage, the large foam bursts or even connects to form the open-cell foam. If it does not burst, it is the closed-cell foam [36]. The final pore structure of MF is controlled by two processes: foaming and curing. The curing rate of the resin primarily depends on the viscosity of the resin, solid content, curing agent dosage, *etc.*, whereas the foaming rate is mainly determined by foaming agent, foaming method, and foaming process [1,12,19]. Therefore, investigating how these factors affect both mechanical properties and microstructure of foam is crucial for enhancing the toughening effect of MF.

### 3.1 Physical properties of MR solution

The viscosity of MR solutions is a crucial factor that affects its foaming properties [49,99]. Zhang *et al.* [100] prepared open and closed cell thermosetting foams using MR and formaldehyde as the main raw materials, and investigated the impact of prepolymer viscosity on foam density and structure. The results showed that low initial viscosity resulted in open-cell foam formation while increasing viscosity led to incomplete bubble expansion resulting in closed-cell foam production. For open-cell foam (Figure 9), low viscosity solution resulted in bubble growth instability and collapse, and the bubble forming rate was low, which resulted in high density. With the increase in viscosity, the bubble forming rate increased and the density decreased. When the density increased again, the pores became smaller and denser, the density and the toughness increased again. For closed-cell foam, the higher the viscosity, the smaller the bubble hole and the higher the foam density (Figure 10). In both cases, there was an optimal viscosity that minimized the foam density formed, about 70 mPa s for the open-cell foams and 850 mPa s for the closed-cell foams. Moreover, only within this appropriate range did MF foam exhibit uniform bubble holes resulting in good toughness.

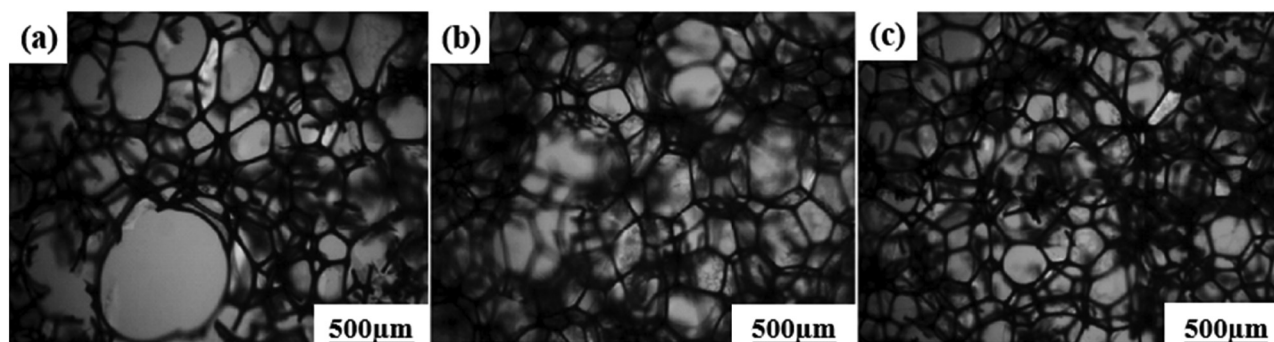
On the other hand, the solid content of MR is another one of the important factors affecting its foaming. Ruipeng *et al.* [101] analyzed the effect of solid content of MR on the physical properties and mechanical properties of MF. The results showed that when the solid content was too low, shrinkage bubble phenomenon occurred, but when the solid content increased, the reaction system and resin storage stability became worse, and more strict process control was needed. In addition, when the solid content of melamine formaldehyde resin was 69%, the flexural strength of MF reached the best, *i.e.*, 305 kPa.

### 3.2 Foaming method

The foaming methods of MF mainly include oven foaming and microwave foaming [49,64]. Xia and Wang [29,62] reported and compared the microstructure of MF prepared by these two methods. As shown in Figure 11a, the bubble hole formed during oven foaming exhibited a predominantly open-hole through structure resembling rigid spheres with approximately 14–16 sides. Due to the thickness of the foam wall, most of the skeletons were piled together and showed a sheet shape. The aspect ratio ( $H/D$ ) of the foam was between 1 and 2, so the flexibility was poor and the rigidity was strong. In contrast, as shown in Figure 11b, microwave-induced MF primarily consisted of interconnected ligamentous skeletons that were linked *via* three-tip inner swing tops, thereby increasing contact area and enhancing stability. Moreover, the length-to-diameter ratio ( $L/D$  value) of this mesh-like structure reached approximately 15, leading to improved toughness. This can be attributed mainly to the faster heating rate associated with microwave processing compared to oven foaming; uniform heating without temperature gradients allowed for instantaneous boiling and bubbling of the foaming agent, thus yielding a more pronounced three-dimensional network skeleton structure.

In fact, the length to diameter ratio and framework of MF formed by microwave and oven were very different, resulting in significant differences in the mechanical properties of MF. When applying pressure to the microwave-foamed MF frame by microwave, the frame can produce significant bending and torsion deformation, and the bending and torsion deformation back quickly when the external force removed. Macroscopically, the compression test showed the bending deformation between layers, and the local structural failure was inhibited by the contact generated by the bending deformation between skeletons. Hence, the damage spreading trend was avoided and showed good flexibility [102,103]. However, when the MF foamed by oven was stressed, the rigid skeleton did not undergo bending deformation when the load was transferred between adjacent layers, and the load continued to penetrate into the middle of the next layer of foam body, which resulted in cracks and bubble burst [29,62].

In addition, microwave penetration ensured even heating throughout the foam body; once microwaves ceased heating, the temperature inside returned to room temperature rapidly. Microwave heating also accelerated low molecular volatilization to form more foam pores and promoted cross-linking to stabilize foam growth [100]. Hence, microwave foaming has more obvious advantages than oven heating foaming as it can improve the toughness of MF by improving bubble structure.



**Figure 9:** Micrograph of open-cell foams with different viscosities ( $\eta$ ): (a)  $\eta = 40$  mPa s, (b)  $\eta = 100$  mPa s, and (c)  $\eta = 140$  mPa s [100].

### 3.3 Foaming process

Microwave radiation power and radiation time have great influence on foam performance [104]. In the microwave foaming process, it is crucial for the MR to rapidly reach a high temperature within a short period of time. First, the foaming agent is rapidly vaporized under the action of high temperature and emulsifier to produce a large amount of foam, and solidified rapidly under the action of curing agent at the same time [105]. The MF with good comprehensive performance can be prepared only when the two processes of curing and foaming of resin are matched [43,106].

Wan *et al.* [105] investigated the effects of microwave power and foaming time on the structure and properties of MF foam. Figure 12 illustrates micrographs of MF at different microwave powers when an initial temperature of 25°C was maintained during a 60 s microwave foaming process. At a microwave power level of 1 kW h, the curing rate exceeded the foaming rate resulting in relatively thick skeleton structures within foam holes and higher foam density. Conversely, at a microwave power level of 3 kW h where the foaming rate surpassed the curing rate excessively, larger merged foam holes were observed with relatively fine skeleton structures within them as well as lower

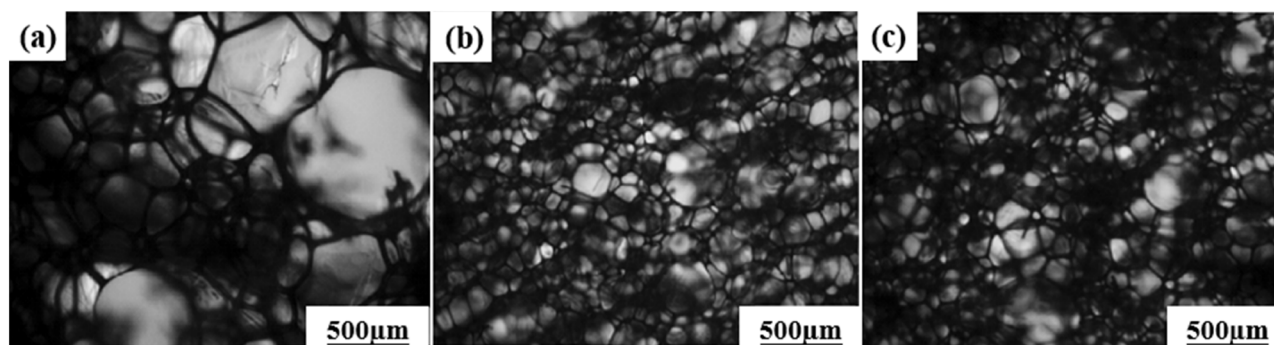
foam density. Only when using a microwave power level of 2 kW h did both foaming rate and curing rate match each other perfectly leading to highly uniform bubble holes in the foam.

Micrographs of the MF under different microwave times are presented in Figure 13, when the microwave power was set at 2 kW h. It can be observed from Figure 13 that excessive foaming time resulted in numerous cracks appearing in the foam structure. Conversely, if the foaming time was too short, incomplete curing and insufficient foam expansion occurred, potentially leading to severe shrinkage. The optimal foaming time was determined to be 60 s.

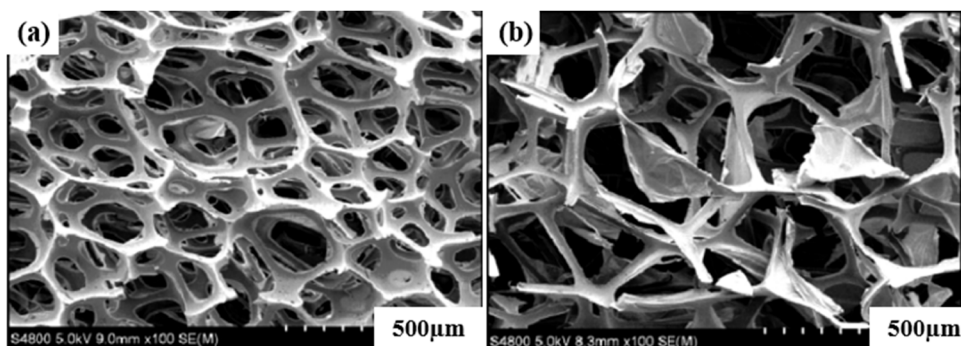
Therefore, in the process of MR foaming, the power and time of microwave radiation should be gradually regulated. In this way, the curing process and foaming process of resin can be matched, and MF with good comprehensive performance can be prepared.

### 3.4 Foaming agent and curing agent

Foaming agent and curing agent are two crucial additives in the foaming process of MR, which play a pivotal role in



**Figure 10:** Micrograph of closed-cell foams with different viscosities ( $\eta$ ): (a)  $\eta = 40$  mPa s, (b)  $\eta = 100$  mPa s, and (c)  $\eta = 140$  mPa s [100].



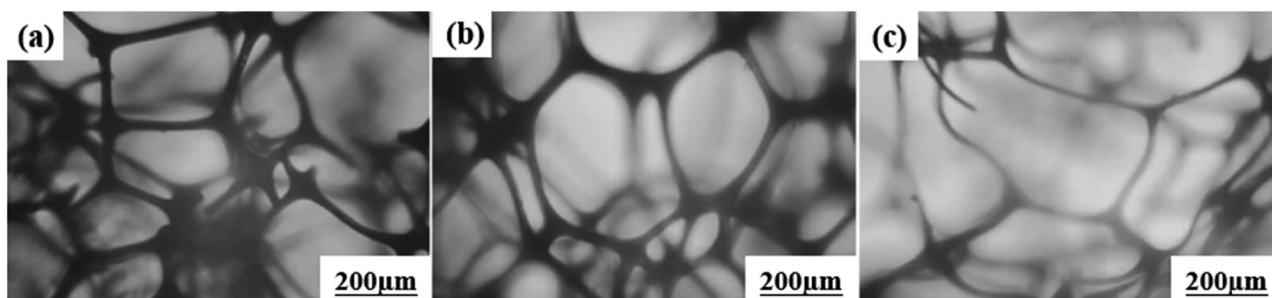
**Figure 11:** Morphology of MF produced by different foaming methods: (a) oven foaming and (b) microwave foaming [29,62].

maintaining foaming equilibrium [107,108]. The effect of foaming agent on the foam was mainly reflected on the foaming speed. Xia *et al.* [62] systematically investigated the effects of inorganic, organic, and compound foaming agents on the mechanical properties and flame retardancy of MF. The results showed that when  $\text{NaHCO}_3$  was used as foaming agent, the performance of the foaming agent was better than that of other inorganic foaming agents. In terms of flammability:  $\text{NaHCO}_3 > \text{Na}_2\text{CO}_3 > \text{organic foaming agent}$ ; in terms of mechanical properties, the mechanical properties of MF obtained by using chemical blowing agent diphenylmethane diisocyanate (MDI) were significantly better than other blowing agents, because MDI could be used not only as blowing agent but also as cross-linking curing agent, which greatly improved the mechanical properties of foam.

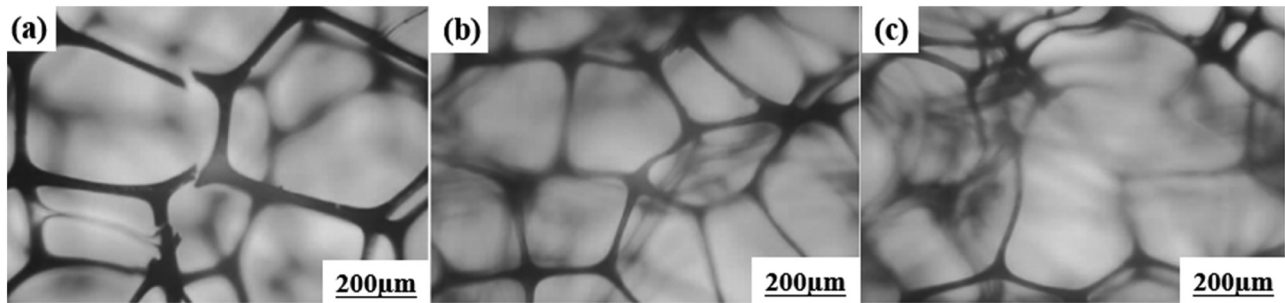
The curing agent's influence on the foam is mainly reflected in the curing speed. Generally speaking, when the curing rate surpasses the foaming rate, both the opening ratio and the length-diameter ratio will decrease and the bubble will collapse. Zhang *et al.* [108] reported the effect of the amount of formic acid (foaming agent) on the structures of the MF. It can be seen that excess formic acid decreased the opening rate (Figure 14a), and underuse increased the aspect ratio of the foam (Figure 14b).

Lei *et al.* [109] investigated the effects of the amount of foaming agent and curing agent on the apparent density and morphology of MF. The results demonstrated a direct influence of the foaming agent and curing agent quantities on the apparent density of MF (Figure 15). Excessive or insufficient amounts of foaming agent or curing agent led to an increase in foam's apparent density, as depicted in Figure 15. When there was an excess amount of foaming agent or a deficiency in curing agent, the foaming rate exceeded the curing rate, resulting in bubble fragmentation and consolidation, thereby increasing foam's apparent density. Conversely, when there was an excess amount of curing agent or a deficiency in foaming agent, the curing rate surpassed the foaming rate leading to inadequate foam expansion with more closed cells and increased apparent density. Optimal conditions for achieving light foam with uniform pore size and minimal apparent density were observed at 10% dosage for both foaming and curing agents relative to MR mass; this balanced ratio ensured similar rates between foaming and curing processes while enhancing MF toughness.

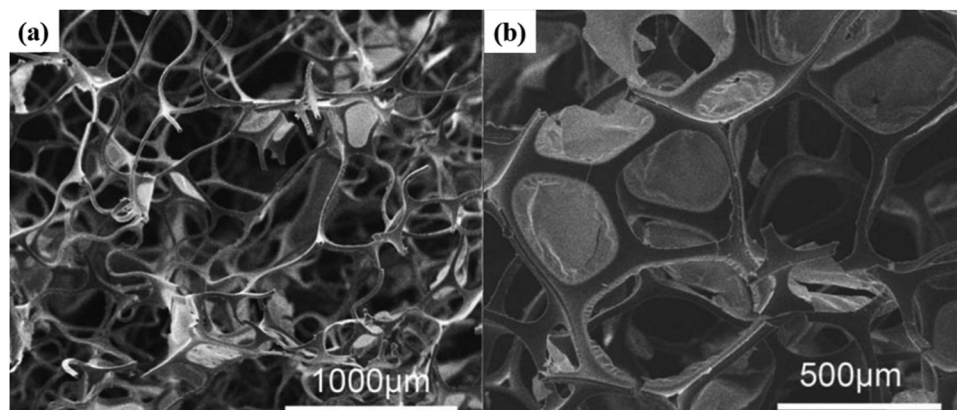
Hence, in the experimental design stage of MR foaming, the synergistic mechanism of foaming agent and curing agent should be carefully considered, so as to avoid the inconsistent phenomenon of foaming speed and curing speed, and then obtain MF with good toughness and low density.



**Figure 12:** Micrographs of MF under different microwave powers; (a) 1 kW h, (b) 2 kW h, and (c) 3 kW h [105].



**Figure 13:** Micrographs of MF under different microwave time; (a) 30 s, (b) 60 s, and (c) 90 s [105].

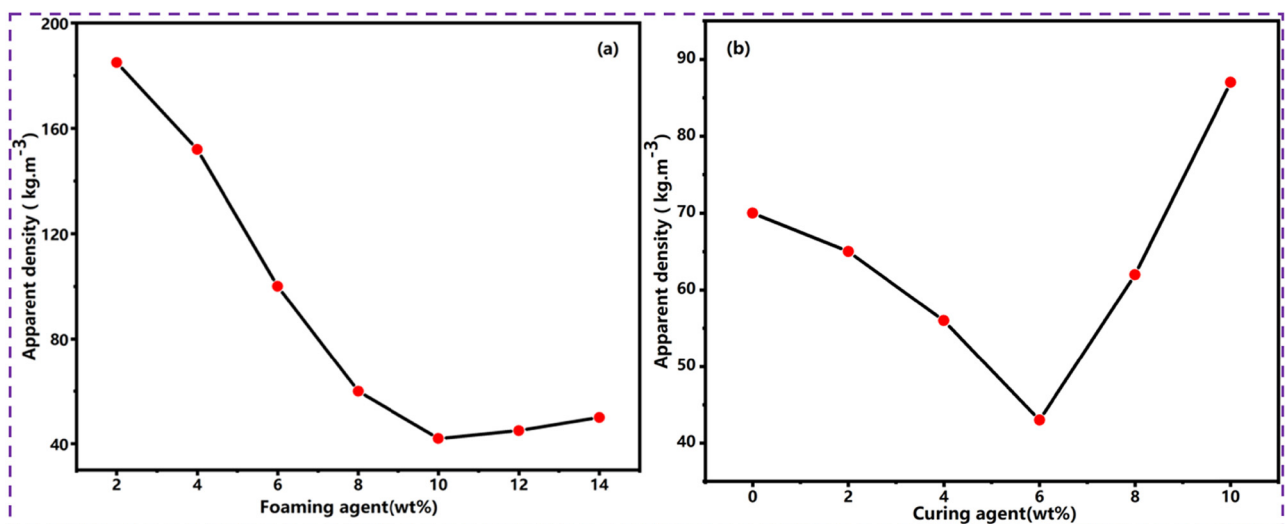


**Figure 14:** SEM micrographs of MFs. (a) The foam structure whose curing speed is slower than foaming speed and (b) the foam structure whose curing speed is faster than foaming speed [108].

### 3.5 Surfactant and other foaming AIDS

Surfactant is an important part of foam formation and stabilization until curing, and has a significant impact on the surface tension of the resin [104,106]. Too much surface

tension of resin is not conducive to the formation of foam, and too little surface tension is not conducive to the stability of foam. Therefore, the amount of surfactant must be controlled in a certain range [100]. Thongkham *et al.* [110] proposed a flexible and lightweight thermoelectric material



**Figure 15:** Effect of the mass fraction of foaming agent (a) and curing agent (b) on apparent density of MF [109].

developed on a MF using a simple dip-dry technique to self-assemble conductive nanofilms in the scaffold. Different amounts of poly(3,4-ethylenedioxythiophene):polystyrene-sulfonate conductive nanofilms were variedly fabricated in the foam due to altered amounts of sodium dodecyl sulfate (SDS) surfactant from 0 to 5 wt%. As a result, the highest nanofilm formation in the foam structure is achieved by adding 3 wt% SDS.

In addition to surfactant, some other additives also can be added to improve the foam performance during the foaming process of MR. Wang *et al.* [111] fabricated superhydrophobic MF by *in situ* coprecipitation of hydrophobic SA@Fe<sub>3</sub>O<sub>4</sub> (SA = stearic acid) nanoparticles on the surface of MF foam skeleton. Due to its low surface energy, high porosity, and large magnetic particles, the modified MF foam exhibits excellent super hydrophobicity, and oleophilic and magnetic responses.

Moreover, incorporating the MF into solvents during foaming completion represents an alternative approach to enhance the bubble structure and improve the toughness of MF. In order to improve the hydrophobic property of MF, Wang *et al.* [38] immersed the prepared MF in the aqueous solution of PVA. The results showed that PVA formed micro-nano structure on MF skeleton, which also improved the surface roughness and hydrophobic property. Compared with the complex methods and nanomaterials of traditional porous superhydrophobic adsorbents, the highly hydrophobic MF obtained by simple impregnation has better advantages in the treatment of oil spills and industrial organic pollution.

### 3.6 Comprehensive regulation analysis of MF pore structure

In summary, the effect of the bubble structure of MF properties is as important as that of the molecular structure of MR. In order to obtain the MF with uniform foam holes, small density, and good mechanical properties, the physical properties (viscosity and solid content), foaming method (oven foaming and microwave foaming), foaming time, and power of the resin solution are the key effect factors in the foaming process. In addition, the foaming speed (foaming agent), curing speed (curing agent), and surfactants are also the important effect elements of MF processing. By adjusting the content of foaming agent, curing agent, and surfactant, the MF pore structure and density can be well regulated and controlled. It is worth noting that the microwave foaming has more obvious advantages than oven heating foaming, which can improve the toughness of MF by improving the bubble structure and should be further studied. The influencing factors of MF pore structure and MR molecular structure are shown in Figure 16.

## 4 Extended application of MF foam

As previously mentioned, MF has the features of excellent flame retardance, good sound absorption property, low density, and high opening, these characteristics enable

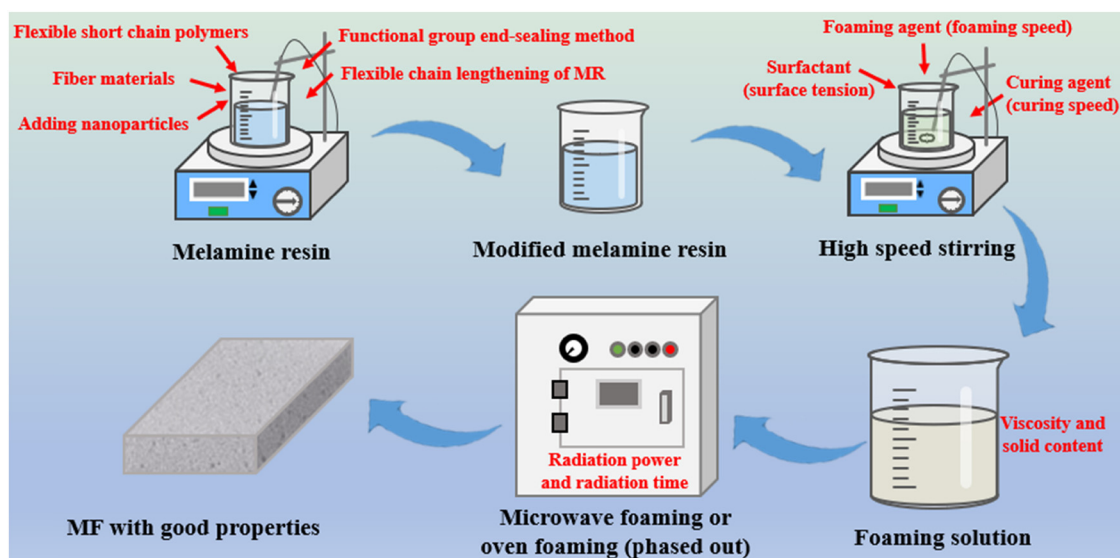
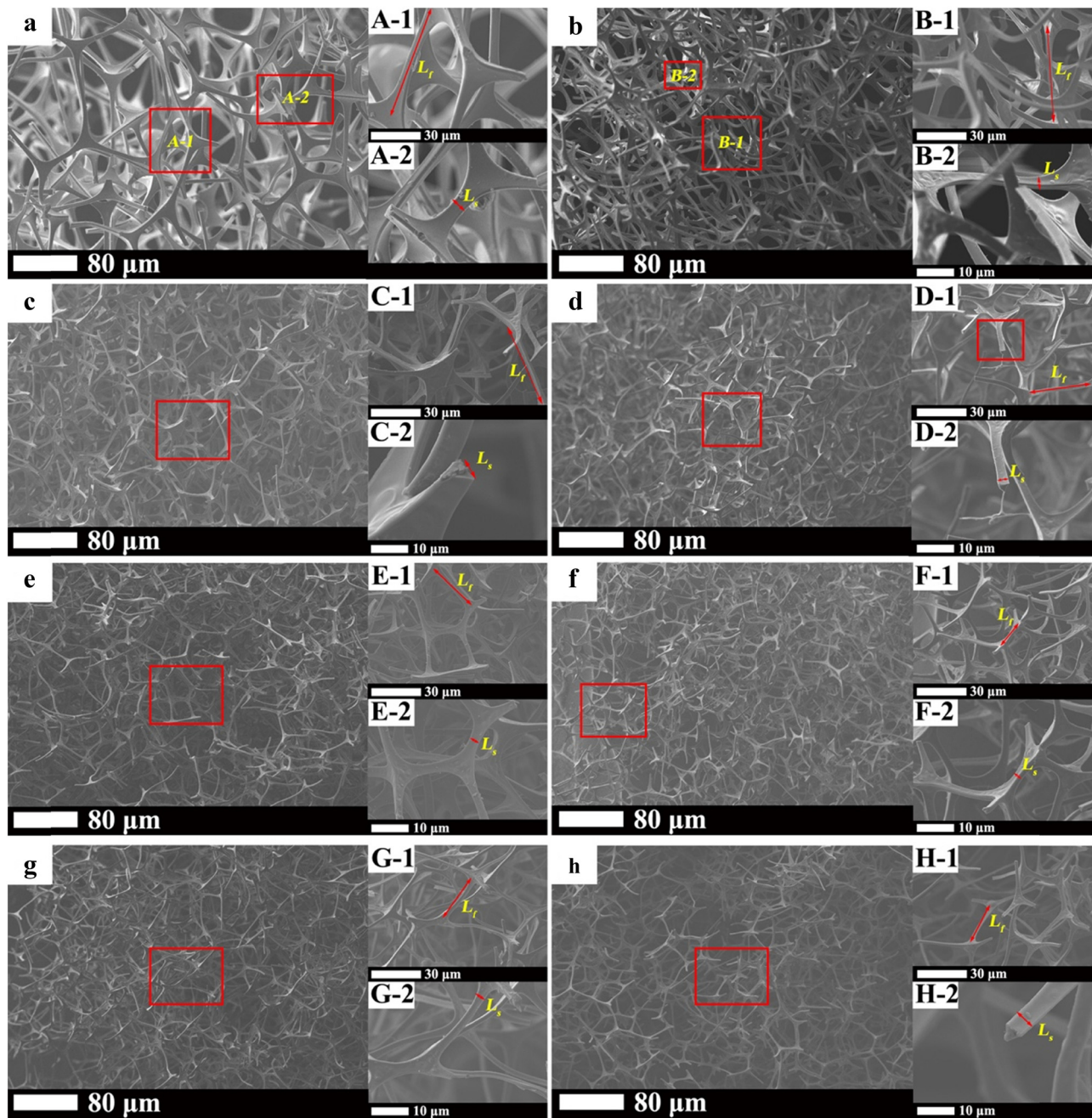


Figure 16: The influencing elements of MR molecular structure and MF pore structure.



**Figure 17:** Microstructures of carbon foam samples were obtained at different pyrolysis temperatures. (a) MF, (b) 450°C, (c) 550°C, (d) 650°C, (e) 750°C, (f) 850°C (g) 950°C, and (h) 1,050°C.

MF to be successfully applied in the fields of oil and water separation, VIP, building materials, and automotive sound-proofing materials. With the continuous exploration of experts and scholars, the application scope of MF has been gradually expanded. This chapter will summarize the two main extended application areas (carbon foam and carbon aerogel [112,113]) of MF foam.

## 4.1 Carbon foam

### 4.1.1 Thermal insulation

Carbon foams, which can be produced through direct carbonization of MF, exhibit promising potential in supporting thermal insulation systems due to their exceptional

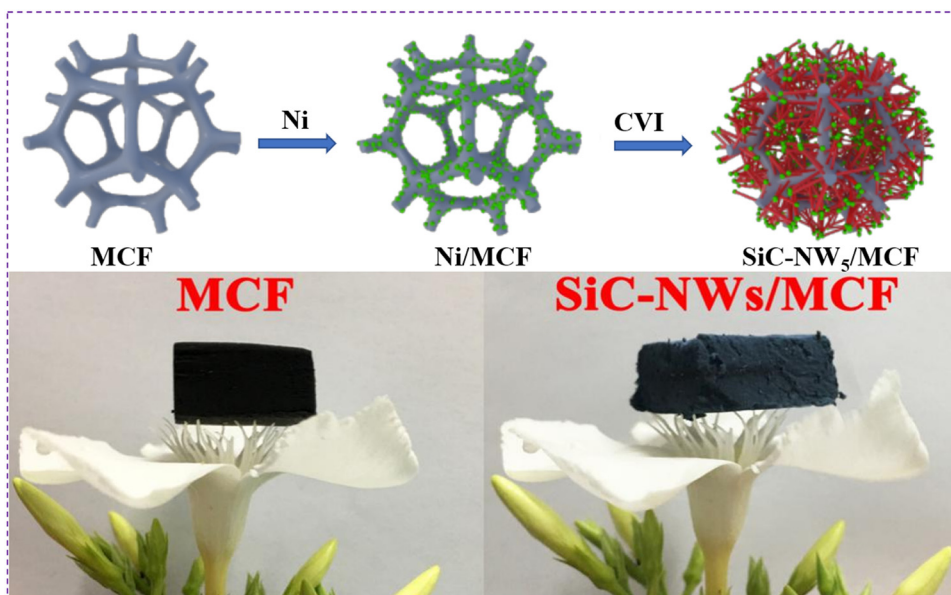


Figure 18: Preparation of SiC nanowires-reinforced MCF foams [119].

form-stable characteristics in photothermal conversion and storage [114–116]. For instance, the pyrolysis process of carbon foam matrices prepared from MFs was investigated by Chen's team at Nanjing University of Aeronautics and Astronautics [117,118]. The SEM images presented in Figure 17 depict the microstructure of carbon foam under various pyrolysis temperatures ranging from 450 to 1,050°C. As can be seen from Figure 17a, the pore size of the non-pyrolyzed dense amine foam sample was large, with the pore size ranging from 100 to 200  $\mu\text{m}$ , and the pore wall was thick and long. However, with the increase in the pyrolysis temperature, the pore structure participation of the prepared

carbon foam was obviously different from that of the dense amine foam sample. As shown in Figure 17b–f, it can be clearly seen that the carbon foam still maintained the three-dimensional mesh open-pore skeleton structure in the MF sample. However, with the increase in the pyrolysis temperature, the number of three-dimensional mesh pores in the field of view enhanced significantly, and the corresponding pore size also shrank gradually, from 100–200 to 20–80  $\mu\text{m}$  in the sample of dense amine foam. At the same time, the pore wall of carbon foam gradually became more thinner and shorter. The small pores in the three-dimensional network increased the scattering, reflection, and absorption of radiation energy passing through the porous structure, and effectively reduced the thermal conductivity of melamine-derived carbon foam (MDCF) samples.

In the previous work of Chen *et al.* [117,118], the influence of SiC nanowire parameters on the structure and properties of composite foam under various chemical vapor infiltration processes (Figure 18) was investigated. A novel type of SiC nanowire-reinforced carbon foam composites was proposed, which was composed of a carbon foam matrix as well as the filled SiC nanowires in the carbon foam, with an average pore diameter of 12–14 nm and a thermal conductivity of 0.036 W/(m K) at 100°C and 0.243 W/(m K) at 900°C. And, its bulk density can be as low as 5.56 mg/cm<sup>3</sup>. At present, the stabilization carbon foam has been successfully applied in the first-generation rover insulation board as reinforcing material. And it also has passed all of the technical indicators assessment requirements of the China academy of space technology, Beijing satellite manufacturer Co., Ltd.

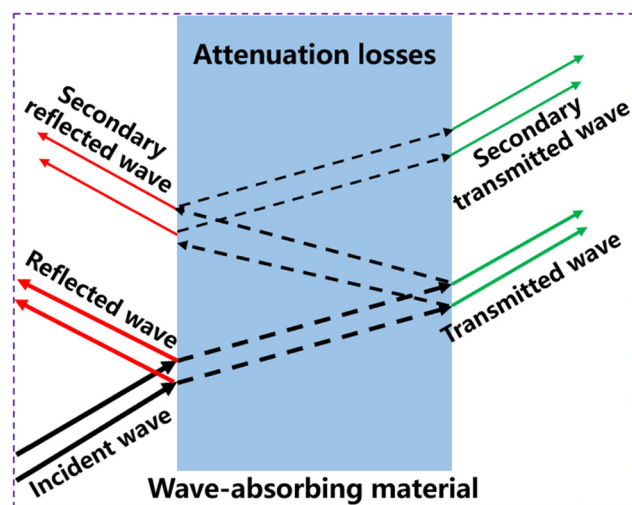
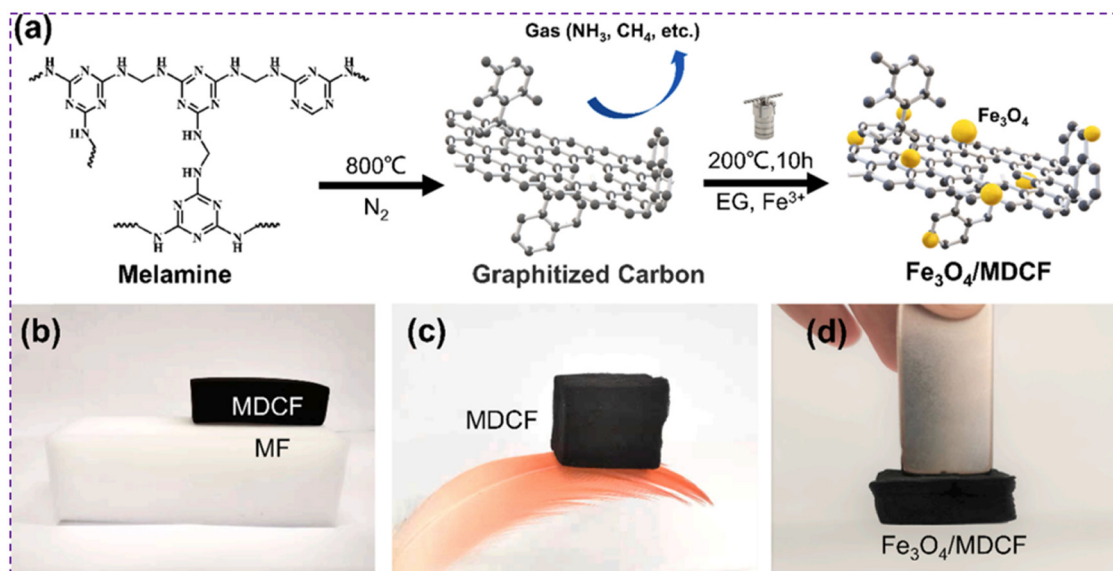


Figure 19: Electromagnetic wave and material interaction principle.



**Figure 20:** (a) Carbonization and synthetic process of Fe<sub>3</sub>O<sub>4</sub>/MDCF composite, (b) optical photograph of MF and MDCF, (c) an ultralight MDCF standing on feather, and (d) Fe<sub>3</sub>O<sub>4</sub>/MDCF composite is attracted by a magnet [123].

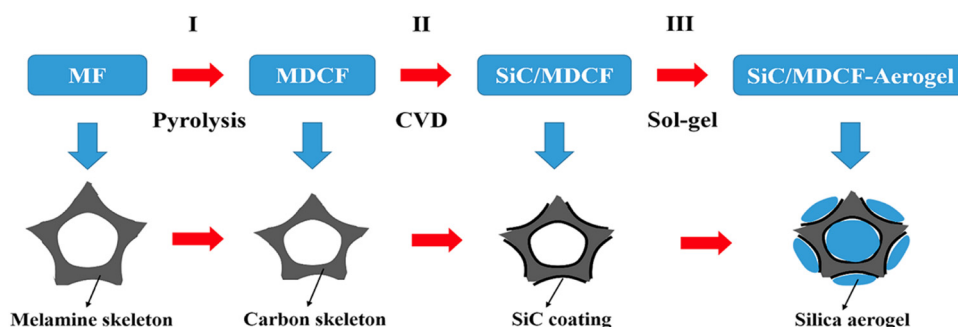
The SiC nanowire-reinforced carbon foams have provided help for implementation of the Mars exploration program and the successful launch of the “Tianwen” Mars exploration rover.

#### 4.1.2 Electromagnetic wave absorption materials

Carbon foam made of MF is not only used in thermal insulation systems of spacecraft, but also widely used in electromagnetic wave absorption materials due to its high porosity and low density [24,120,121]. Lyu *et al.* [122] constructed a three-dimensional carbon foam material embedded with CuNi alloy nanoparticles for electromagnetic wave absorption. In the synthetic procedure, Ni<sup>2+</sup> and Cu<sup>2+</sup> were first adsorbed into the MF to form Ni<sup>2+</sup>-Cu<sup>2+</sup>/MF composites, which were then subject to the pyrolysis treatment, resulting in the production of mesoporous CuNi alloy/CF.

The experiment results showed that the CuNi/CF with CuNi alloy nanoparticles evenly distributed in the 3D carbon matrix inherits the characteristic of the 3D structured N-rich MF, which induces masses of dipole polarization, interfacial polarization, and electromagnetic wave scattering. The CuNi<sub>11</sub> exhibited excellent electromagnetic wave absorption and insulation performance with the reflection loss (RL) of -50.20 dB at a thin thickness of 1.6 mm. The electromagnetic wave and wave-absorbing material interaction principle is shown in Figure 19.

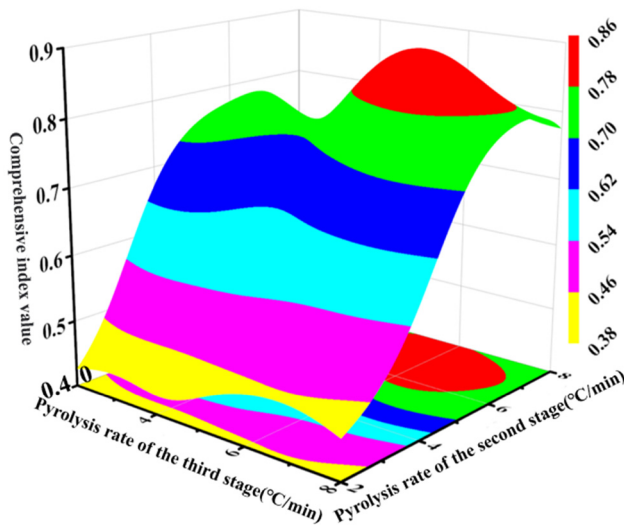
Jiang *et al.* [123] synthesized the ferroferric oxide/MDCF (marked as Fe<sub>3</sub>O<sub>4</sub>/MDCF) by carbonization and *in situ* growth strategy. The results showed that the minimum values of RL, the effective frequency width of absorption, and the density of the Fe<sub>3</sub>O<sub>4</sub>/MDCF were -26.45 dB (7.76 GHz), 4.28 GHz, and 13.1 mg/cm<sup>3</sup>, respectively. The continuous 3D (three dimensional) conductive network and the expanded



**Figure 21:** Manufacturing processes of SiC/MDCF-Aerogel. The main conclusions were as follows:.

**Table 5:** Single performance index corresponding to the optimal comprehensive performance

| Number              | Rate of change of volume density (%) | Compression modulus (kPa) | Absorbing ability (kJ/m <sup>3</sup> ) | Specific energy absorption (kJ/kg) |
|---------------------|--------------------------------------|---------------------------|----------------------------------------|------------------------------------|
| Optimal performance | 28.81                                | 204.14                    | 7.05                                   | 1.63                               |

**Figure 22:** The three-dimensional surface diagram of comprehensive performance index of carbon foam.

interface in the carbon/magnetic heterostructure improved the interfacial polarization relaxation, and enhanced the electromagnetic loss and wave absorbing capability. The preparation process of the  $\text{Fe}_3\text{O}_4/\text{MDCF}$ , the physical photos of the MF, and the  $\text{Fe}_3\text{O}_4/\text{MDCF}$  composite are shown in Figure 20.

## 4.2 Aerogel areas

Carbon aerogel is a prominent area of MF expansion in the field of multifunctional material [124–128]. With the growing

complex operating environment of the modern weapon systems, the electromagnetic interference phenomenon has become an increasingly serious concern as a single performance index cannot meet the usage requirements of the highly sophisticated field for performance optimization and functional pluralism materials. In line with this perspective, Chen *et al.* [129–139] proposed a carbon/silicon carbide/silica aerogel composite with lightweight, high strength, heat insulation, and microwave absorption performances. The carbon foam matrix made of MF was optimized using a segmented pyrolysis process and a neural network model. Then, the uniform silicon carbide coating was deposited on the surface of the carbon foam skeleton by the chemical vapor deposition technology to prepare the carbon/silicon carbide foam. Finally, the silica aerogel was filled in the pores of the carbon/silicon carbide foam by the sol–gel technique, resulting in the carbon/silicon carbide/silica aerogel composite after drying under normal pressure. The manufacturing process is shown in Figure 21.

1) The structure and performance of melamine-based carbon foam was primarily determined by the pyrolysis process, and the constructed backpropagation neural network model can optimize the pyrolysis process of carbon foam to obtain the carbon foam with the best comprehensive performance. The single performance index corresponding to the optimal comprehensive performance is shown in Table 5, and the three-dimensional surface diagram of comprehensive performance index of carbon foam is depicted in Figure 22.

2) The silicon carbide coating on the surface of carbon foam can be prepared via a chemical vapor deposition

**Table 6:** Electromagnetic absorbing performance of silicon carbide matrix composites

| Samples                   | Min RL (dB) | Min frequency (GHz) | Min matching thickness (mm) | Effective absorption band (GHz) | Reference |
|---------------------------|-------------|---------------------|-----------------------------|---------------------------------|-----------|
| C/SiC-16h                 | −29.74      | 11.12               | 1.75                        | 3.62                            | [132,133] |
| C/SiC-T900                | −51.58      | 8.96                | 3.60                        | 10.84                           | [131,132] |
| SiC <sub>fiber</sub> /SiC | −47.50      | 11.40               | 2.50                        | 2.80                            | [140]     |
| SiC aerogel               | −43.00      | 13.00               | 2.00                        | 4.00                            | [141]     |
| SiC nanowire foam         | −49.10      | 12.20               | 2.85                        | 4.20                            | [142]     |
| Graphite/SiC aerogel      | −47.30      | 10.52               | 3.00                        | 4.70                            | [143]     |
| SiC mixed carbon foam     | −43.20      | 4.70                | 5.00                        | 4.00                            | [144]     |

**Table 7:** Comparison of absorbing properties of SiC/SiO<sub>2</sub> composites

| Samples                                   | Min RL (dB) | Min frequency (GHz) | Min matching thickness (mm) | Effective absorption band (GHz) | Reference |
|-------------------------------------------|-------------|---------------------|-----------------------------|---------------------------------|-----------|
| C/SiC/SiO <sub>2</sub> -T900              | -55.38      | 16.56               | 2.85                        | 8.16                            | [129]     |
| Fe-SiC/SiO <sub>2</sub>                   | -32.00      | 9.00                | 3.25                        | 4.20                            | [145]     |
| SiC/SiO <sub>2</sub> composites           | -52.00      | 10.80               | ~3.00                       | 4.20                            | [146]     |
| SiC/SiO <sub>2</sub> -1400                | -52.00      | 10.80               | 2.75                        | 4.00                            | —         |
| SiC/SiO <sub>2</sub> core-shell nanowires | -32.72      | 13.84               | 3.00                        | 5.32                            | [147]     |
| Activated carbon-based C/SiC/SiO          | -30.80      | ~15.20              | 1.50                        | ~4.80                           | —         |

process. The thickness of silicon carbide coating significantly influences the structure and mechanical properties, dielectric properties, and wave absorption properties of carbon/silicon carbide foam, and the heat treatment temperature can improve the properties of carbon/silicon carbide foam. The electromagnetic absorbing performances of silicon carbide matrix composites are shown in Table 6. Compared with the current traditional silicon carbide fiber or silicon carbide matrix, the C/SiC-T900 foam after 900°C heat treatment had lower minimum RL and wider effective absorption band, so it showed better microwave absorption performance, and had great application potential in the future military or civil microwave absorption system.

3) The sol-gel process employed for the preparation and infiltration of aerogel into the pores of carbon/silicon carbide foam. The structure and properties of carbon/silicon carbide/silicon dioxide aerogel are affected by the thickness of silicon carbide coating, and the heat treatment temperature can improve the properties of carbon/silicon carbide/silicon dioxide aerogel. The comparison of absorbing properties of

SiC/SiO<sub>2</sub> composites are shown in Table 7. It can be found that the C/SiC/SiO<sub>2</sub>-T900 aerogel obtained by heat treatment at 900°C had lower minimum RL, wider effective absorption band, better microwave absorption performance, and great application potential.

In addition to this study, many other scholars have also reported the applications of carbon aerogel made with MF. Such as Wang *et al.* [113] reported a very facile approach for the preparation of compressible, fatigue resistant, conductive, and pressure-sensitive carbon aerogels by pyrolysis of cellulose nano-fibers aerogel using MF as the skeleton. The resulting carbon aerogels exhibit excellent performance, including a low density of 11.23 mg/cm<sup>3</sup>, high electrical conductivity of 0.378 S/cm, high sensitivity, and outstanding mechanical properties. Moreover, the high nitrogen content and hydrophilic property enable the carbon aerogels to be used as compressible electrodes with a specific and areal capacitance of 92.2 F/g and 461 mF/cm<sup>2</sup>, respectively, showing the promising prospect of flexible supercapacitors. The specific applications of carbon aerogel prepared by MF are shown in Table 8.

**Table 8:** Specific application of carbon aerogel prepared by MF

| Applications                                                           | Methods or process                                                                     | Properties                                                                                                       |
|------------------------------------------------------------------------|----------------------------------------------------------------------------------------|------------------------------------------------------------------------------------------------------------------|
| MXene-based aerogel areas [125]                                        | Using MF as a robust template for MXene/reduced graphene oxide aerogel                 | High solvent absorption capacity, great electrothermal conversion rate, and good photothermal conversion ability |
| Solar energy utilization areas [127]                                   | Sol-gel polymerization, freeze drying, and carbonization                               | Excellent thermal stability and high thermal storage density                                                     |
| Organic absorption and high-temperature thermal insulation areas [128] | Sol-gel polymerization, ambient pressure drying, and co-carbonization                  | High compressive strength, ultra-low thermal conductivity, and high-volume organic absorption capability         |
| Electromagnetic wave absorption materials [129]                        | Pyrolysis, sol-gel method, and chemical vapor deposition                               | Excellent microwave absorption performance                                                                       |
| Flexible supercapacitors [113]                                         | Using MFs as the skeleton, and pyrolysis cellulose nanofibers aerogel                  | Low density, outstanding mechanical properties, high electrical conductivity, and sensitivity                    |
| Supercapacitor (nitrogen-doped carbon aerogel) [148]                   | Direct pyrolysis of bamboo cellulose nanofibers/melamine/graphene oxide hybrid aerogel | Superior mechanical durability and excellent electrochemical stability                                           |

## 5 Conclusion

Due to its excellent sound absorption, heat resistance, and preservation properties, as well as low density, MF has been widely utilized in the preparation of oil–water separation materials, supercapacitors, sound absorption materials, and vacuum insulated panels in recent years. However, its own shortcomings of MF such as hardness, poor toughness, fragility, and slag removal greatly limit its application scope. Therefore, this review addressed the recent toughening modification methods and application areas of MF. Specifically, future research should explore the following four aspects of MF toughening modification.

- 1) Exploring cost-effective modifiers and determining the optimal dosage to reduce the crosslink density of MR, thereby enhancing the toughness of MF without compromising its flame retardant and heat resistance properties.
- 2) By integrating different toughening and modification methods in accordance with specific application requirements, incorporating flexible chain segments with excellent properties into MR can lead to the synthesis of composite modified MF with improved toughness.
- 3) Optimizing the synthesis process and foaming process of MR from the perspectives of simplifying production procedures and reducing costs, aiming to enhance foam quality by achieving low density and uniform bubble pores.
- 4) Continuously advancing research on MF modification mechanisms to facilitate functionalization and refinement, thus promoting further development in this field.

Moreover, considering the low density, high porosity, and exceptional heat insulation characteristics of MF, it is imperative to further investigate the potential application of carbon foam and aerogel derived from MF in the aviation industry and electromagnetic wave absorption field. It is anticipated that through extensive research on MF synthesis, foaming technology advancements, and nanotechnology developments, there will be continuous enhancements in the mechanical properties, microwave absorbing capabilities, and thermal insulation performance of MF derivatives. Consequently, commercialization and serialization of MDCF and aerogel can be expected in the near future.

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