

## Research Article

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# Prediction of lap shear strength of GNP and TiO<sub>2</sub>/epoxy nanocomposite adhesives

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**Abstract:** In this study, graphene nanoplatelets (GNPs) and titanium dioxide nanofillers were added to epoxy resin P-5005 at five different weight percentages (wt%), *viz.*, 1, 5, 10, 15, and 20 wt%. The tensile properties of the nanocomposites were experimentally tested following ASTM D638-14. Then, the above-mentioned nanocomposites were applied as adhesives for an overlap joint of two A5055 aluminum sheets. The apparent shear strength behavior of joints was tested following ASTM D1002-01. Moreover, experimentally obtained results were applied to train and test machine learning and deep learning models, *i.e.*, adaptive neuro-fuzzy inference system, support vector machine, multiple linear regression, and artificial neural network (ANN). The peak tensile strength (TS) and joint failure load (FL) values were observed in epoxy/GNP samples. The ANN model exhibited the least error in predicting the TS and FL of the considered nanocomposites. The epoxy/GNP nanocomposites exhibited the highest TS of 28.49 MPa at 1 wt%,

and the peak overlap joints exhibited an FL of 3.69 kN at 15 wt%.

**Keywords:** graphene nanoplatelets, titanium dioxide, mechanical characteristics, machine learning

## 1 Introduction

Epoxy resin is one of the most vital structural adhesives, widely popular in aerospace, automotive, electronics, civil, and packaging industries. Recently, the adhesive industry has changed significantly by including new formulations, raw materials, substrates, operation conditions, applications, and curing processes [1]. Therefore, properties such as resistance to failure at vibration and fatigue loading, resistance to thermal cycling and high service temperatures, and optimum curing conditions have to be considered in epoxy adhesives. Numerous studies aimed to develop multifunctional epoxy adhesives by mixing epoxy with a second material as a filler, such as carbon nanotubes (CNTs), graphene, Al<sub>2</sub>O<sub>3</sub>, CaCO<sub>3</sub>, SiO<sub>2</sub>, ZrO<sub>2</sub>, titanium dioxide (TiO<sub>2</sub>) and so on. In recent years, the size of fillers has shifted from micro- to nano-scale, resulting in much superior multifunctional characteristics in nanocomposite adhesives compared to neat adhesives and their composites having conventional micro-particles [2,3]. The addition of nanoparticles to epoxy adhesive joints improves their mechanical properties, joints' failure load (FL), thermal stability, and electrical conductivity [4–7]. However, it varies based on several parameters, such as particle characteristics, functional groups on the nanoparticle surface, size distribution, size, content, and shape, defining its compatibility with the epoxy matrix [8–12].

Graphene is a carbon allotrope that consists of a one-atom layer and its lattice nanostructure is arranged in a 2D honeycomb structure. Since 2004, it has been considered one of the most stiff and strong materials. The Young's modulus and tensile strength (TS) of graphene are around 1 TPa and 130 GPa, respectively [13]; moreover, the thermal conductivity is around 5,000 W/mK [14]. Multilayered

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graphene, also known as graphene nanoplatelets (GNPs), has become a very desirable polymer matrix filler with a great price-to-performance ratio and is utilized in diverse fields, *i.e.*, energy, defense, electronics, transport, medicine, *etc.*  $\text{TiO}_2$ , which is also known as titanium, is a natural mineral available in various crystalline structures, and its applications include the textile industry, food products, coatings, plastics, pharmaceuticals, and so on.

Multiple research studies have investigated the mechanical, thermal, and electrical characteristics of epoxy adhesives reinforced with GNP and metal-based  $\text{TiO}_2$ . Singh *et al.* [15] studied the effect of GNP and multi-walled carbon nanotubes (MWCNTs) on the mechanical characteristics of epoxy-based nanocomposites, and the results showed that the epoxy/GNP composite provided greater tensile and compressive strengths compared to MWCNT/epoxy composites, while the latter exhibited a greater toughness compared to epoxy/GNP. Hence, graphene possesses an outstanding specific surface area, in addition to significant mechanical and electrical properties, which makes graphene the greatest compared to other carbon allotropes for developing multi-functional and structural-reinforced composites [16,17]. Moreover, the dispersion of graphene in the composites and the surface friction force of graphene are the two main properties that have an impact on the ability of graphene to improve damping [18–20].

Mustafa *et al.* [21] have studied the influence of MWCNTs on mechanical characteristics and thermal stability of hybrid nanocomposites. The hybrid polymer structure was made from epoxy resin mixed with different weight percentages of zirconium dioxide ( $\text{ZrO}_2$ ) and yttrium oxide ( $\text{Y}_2\text{O}_3$ ). The structure was further strengthened with 0.1 wt% MWCNT through a hand lay-up casting process. It was reported that the TS and Young's modulus were improved by 24 and 37% in comparison with pure epoxy resin. Kumar *et al.* [22] investigated the tensile and dynamic mechanical properties of epoxy/MWCNT/ $\text{TiO}_2$  hybrid nanocomposites at 0.25, 0.5, 0.75, 1, and 1.5 wt%. The results revealed that the TS of epoxy was increased by 24% by the addition of 1 wt% MWCNTs. Modeling and simulation as well as artificial intelligence have emerged in a wider spectrum for diverse science and engineering applications [23–25]. Even though experimental testing is drastically crucial for the development of new material, machine learning reduces the computational time and cost, since the needed platforms to run machine learning algorithms mainly have free access and can be found easily [26–28]. Artificial intelligence was recently implemented by numerous studies to analyze the characteristics of composite materials, natural fiber composites, and nanocomposites [29–33]. Pati [34] utilized artificial neural networks (ANNs) to predict the wear properties of glass/

epoxy composites, and the input data included the RBD content, erodent size, erodent temperature, impact velocity, and impingement angle. Antil *et al.* [35] applied the response surface metamodel and ANN to evaluate the erosion characteristics of S glass composites, and the input parameters consisted of the impingement angle, nozzle diameter, and slurry pressure. Jayaganthan *et al.* [36] classified the conductivity of epoxy reinforced with 66 wt% silica fillers, 0.7 wt % ion trapping particles and coated with four different coal types by implementing a support vector machine (SVM), logistic regression model,  $K$ -nearest neighbors, and a multi-layer perceptron approach in their LIBS spectral data. Rahman *et al.* [37] applied a convolutional neural network for the analysis of the pull-out force of epoxy/CNT nanocomposites.

The goal of the current research was to study the characteristics of epoxy reinforced with GNP and  $\text{TiO}_2$  nanoparticles for aluminum single-lap joint application by testing the tensile properties of the considered nanocomposites, examining nanocomposite joints' FL, and developing an SVM, multiple linear regression (MLR)-based metamodel, adaptive neuro-fuzzy inference system (ANFIS), and ANNs. The effect of increasing filler weight percentages of 1, 5, 10, 15, and 20 wt% was examined.

## 2 Methodology

$\text{TiO}_2$  and GNP nano-particles were added into an epoxy matrix with five different percentages, *i.e.* 1, 5, 10, 15, and 20 wt%. First, the tensile behaviors of these nanocomposites were evaluated following the ASTM D638-14 tensile testing standard, followed by scanning electron microscopy for testing the fracture surface morphology. Moreover, the aforementioned nanocomposites were applied as an overlap adhesive between two aluminum plates. The apparent shear strength properties of the adhesives were tested following ASTM D1002-01. The experimental results were utilized to train and test machine learning and deep learning models, *i.e.*, ANFIS, SVM, MLR, and ANN to evaluate the ability of the aforementioned to predict the mechanical properties of nanocomposites.

### 2.1 Experiment

Epoxy resin P-5005 by Polymex was utilized in this study as a matrix for the nanocomposite adhesive. The resin-to-hardener mixing ratio (2:1) and matrix-to-filler

compositions were prepared using a high-precision electronic balance. Thus, in order to achieve a homogeneous nanocomposite, the nanocomposites were prepared *via* the following steps: a specific amount of epoxy was added into the mixing container, then nanoparticles were added upon a desired weight percentage, and the mixture was stirred with a plastic spoon for 25 min. To ensure that the nanoparticles in the epoxy were evenly distributed, the mixture was left at room temperature for an extra 2 h. After the addition of the curing agent, the mixture was mixed for 3 min, and then poured into a mold to develop tensile test specimens and to prepare lap-shear test samples. The A5055 aluminum sheets (1 mm thick) were considered as an adherent material. Moreover, the sheets' surfaces were wiped and treated aiming to enhance the bonding between the nanocomposite material and the aluminum sheets. GNPs and TiO<sub>2</sub> were supplied from Nanografi. GNPs had a purity of 99.9%, 5 nm thickness, 30  $\mu$ m diameter, gray in color, a conductivity between 1,100 and 1,600 S/m, and a specific surface area of 170 m<sup>2</sup>/g. TiO<sub>2</sub> nanoparticles were of 20 nm size, 99.9% purity, 4.26 g/cm<sup>3</sup> density, white in color, and 79.87 g/mol molecular weight.

### 2.1.1 Tensile test

The TS values of GNP and TiO<sub>2</sub>-reinforced epoxy nanocomposites were tested following the ASTM standard D638-14. Dog bone-shaped samples were prepared with 200 mm length, 57 mm width, 3.2 mm thickness, and a gauge length of 50 mm. First, the epoxy resin and its hardener were mixed in a 2:1 ratio and then the nanoparticles were added and stirred until the mixture became homogeneous. A layer of releasing agent was sprayed into the wooden mold before pouring the mixture. Then, the samples were cured at room temperature for 12 h. The test was conducted on

an INSTRAM 338571 tensile machine with a testing speed of 5 mm/min.

### 2.1.2 Lap joint

A5055 aluminum was selected as a substrate material. Single lap joint specimens were prepared by first slightly roughening the overlap area using an emery cloth to strengthen the bonding between the adhesive and substrates, followed by immersing the latter in a detergent to prevent any unwanted dirt, grease, or oil. Then, the samples were removed from the solution, washed with deionized water, and dried using a tissue. An epoxy/nano-particles mixture was then spread for a 25 mm  $\times$  25 mm overlap area, keeping a grip length of 200 mm (Figure 1). Thus, the thickness of the cured adhesive was around 0.5 mm. In order to ensure the results' accuracy and avoid undesired errors, three replicates were prepared for each nanocomposite's composition, *i.e.*, 1, 5, 10, 15, and 20 wt%. An INSTRAM 338571 tensile machine (10 kN) was utilized for conducting the single lap joint test at a testing rate of 1 mm/min.

## 2.2 Machine learning and deep learning predictions

Machine learning is derived from artificial intelligence, which is a method that trains the computer to perform tasks that are usually specific for humans and acquired through experience. Increasing the amount of learning samples usually enhances the precision of the algorithm. Moreover, the implementation of deep learning in research attained wide popularity since 2006, as it was applied to

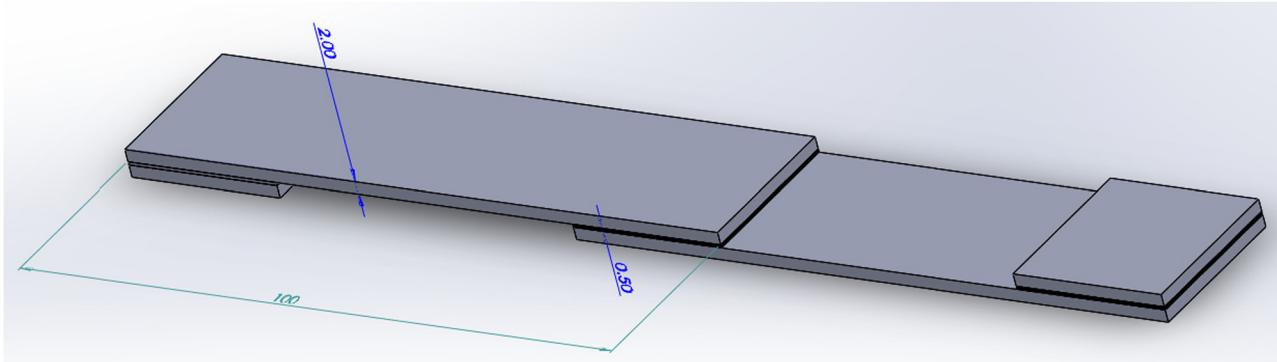


Figure 1: The geometry of adhesively bonded aluminum joints.

object recognition, machine translations, speech recognition, and image segmentation. Basically, due to their neural network structures, most deep learning methods could be considered deep neural networks. Although there are supervised and unsupervised machine learning algorithms, the former is more suitable for manufacturing implementations due to the labeled data it provides:

$$\text{Prediction error \%} = \frac{|\text{Expt. Value} - \text{Pred. Value}|}{\text{Expt. Value}} \times 100. \quad (1)$$

An easy approach for evaluating the training model precision is called prediction error, where the aforementioned is validated using novel input data, which were not considered in the testing stage, which thereby contributes to defining the error percentage. Hence, root mean square error (RMSE) is a prevalent method used to define a model's error:

$$\text{RMSE} = \sqrt{\frac{1}{N} \sum_{i=1}^N (p_i - q_i)^2}, \quad (2)$$

where  $N$  is the whole training data,  $p_i$  is the predicted deliberate information, and  $q_i$  is the real value.

In this study, ANFIS, SVM, ANNs, and the response surface metamodel were applied to evaluate and define the most suitable approach for the prediction of the lap shear strength and TS for input parameters not considered in the experiment, as well as to determine the design space based on the considered parameters. The proposed model employs nanoparticle types and weight percentages as input. The tensile test and shear lap experimentally obtained results were considered for training, validation, and testing. The

precision of the prediction models was assessed by the mean value of absolute percentage error.

### 3 Results and discussion

The tensile and shear characteristics obtained from experimental and machine learning of  $\text{TiO}_2$  and GNP-reinforced epoxy are presented in this section.

#### 3.1 Tensile test

In this section, tensile test results are presented, which mainly highlight the TS values of epoxy/GNP and epoxy/ $\text{TiO}_2$  nanocomposites for five different compositions. Figure 2 illustrates the impact of increasing  $\text{TiO}_2$  and GNP nanoparticles from 1 to 20 wt% on the TS of the end nanocomposites.

As exhibited in Figure 2, increasing the  $\text{TiO}_2$  content increased the TS from 17.96 to 25.23 MPa at 10 wt%, which was the highest TS observed in  $\text{TiO}_2$ -reinforced nanocomposites. Meanwhile, the addition of GNP in the epoxy matrix reached the highest TS value of 28.49 MPa at 1 wt% followed by a gradual decrease to reach 20.22 MPa at 20 wt%. Close TS values of 25.23 and 25.07 MPa were observed in both nanocomposites at 10 wt%, respectively. Thereby, GNP and  $\text{TiO}_2$  nanocomposites followed a descending trend to reach 20.22 and 20.37 MPa at 20 wt%. The highest TS value recorded in this study was observed in GNP-reinforced epoxy at 1 wt%, which was quite similar to the results obtained by Nitesh *et al.*

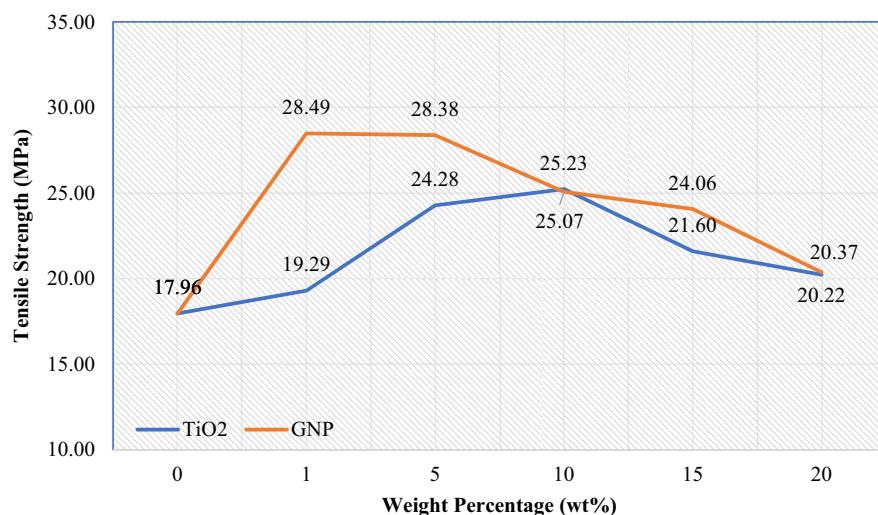
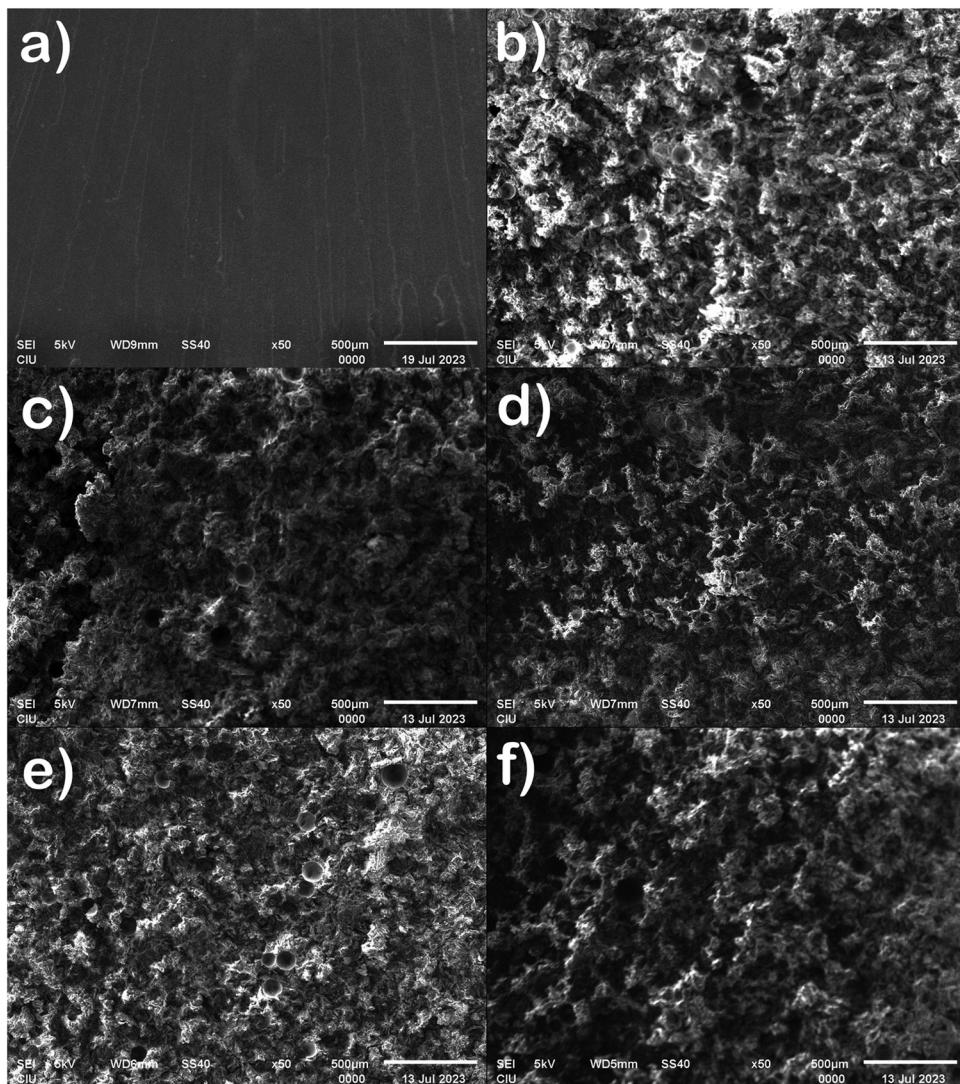


Figure 2: TS of epoxy/ $\text{TiO}_2$  and epoxy/GNP for 5 wt%.



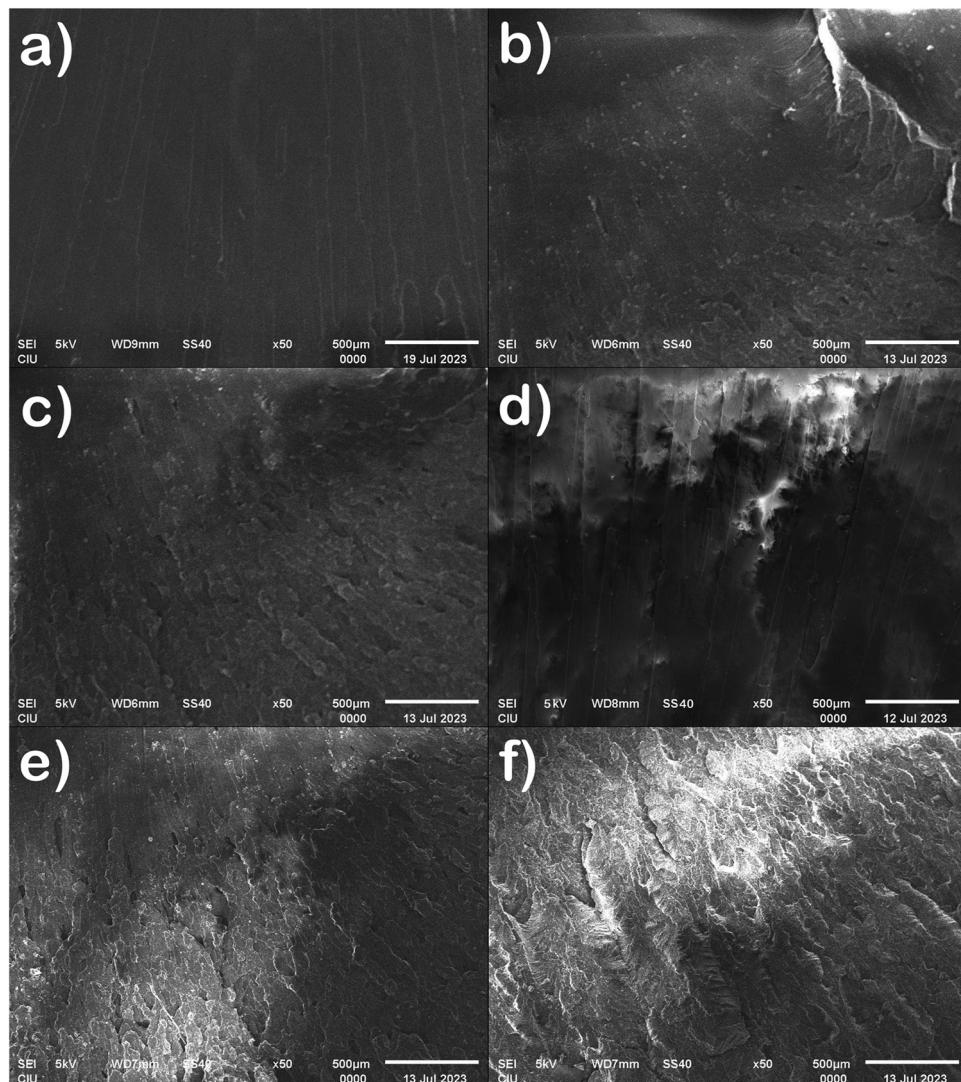
**Figure 3:** Fracture surface of nanocomposite epoxy with different GNP contents: (a) 0 wt%, (b) 1 wt%, (c) 5 wt%, (d) 10 wt%, (e) 15 wt%, and (f) 20 wt%.

[38], where a 1 wt% nanoparticle concentration exhibited the highest tensile behavior, which was ~60% higher than the corresponding values obtained for neat epoxy. The decrease in the TS when the filler content is further increased above the optimal concentration is caused by the agglomeration of the nanofillers, which thereby leads to weaker interface bonding between the matrix and the fillers. Figure 3 illustrates SEM micrographs of the epoxy fracture surface at 1, 20, 5, 10, and 15 wt% GNP.

Larger agglomerates were observed when the GNP content exceeded 1%. The 20% GNP ductile epoxy specimen exhibited a significant agglomeration. Referring to the tensile test results, GNP/epoxy samples with GNP contents exceeding 5% exhibited decreased tensile modulus and TS, which could be due to the increase in the size and amount of agglomerates inside the nanocomposite [39].

The fracture surface of the nanocomposites became markedly rougher upon the increase of the nanofiller content. Figure 4 shows the SEM of  $\text{epoxy}/\text{TiO}_2$  at five different weight percentages.

As clearly shown in Figure 4,  $\text{TiO}_2$  exhibited a good dispersion along with a good interface compatibility with the epoxy matrix. Furthermore, scanning electron micrography of  $\text{TiO}_2$  showed that the manufactured nanoparticles were made up of small, irregularly shaped, fine particles. Peeling away from the grain boundaries indicates brittle fracture, whereas the presence of many dimples indicates ductile fracture surface. Fracture in composites is caused by the initiation and expansion of dimples, fracture of the layers or reinforcing particles, and interface detachment. The SEM graphs show that these fracture surfaces exhibit ductile and cleavage modes of fracture.



**Figure 4:** Fracture surface of the nanocomposite epoxy with different  $\text{TiO}_2$  contents: (a) 0 wt%, (b) 1 wt%, (c) 5 wt%, (d) 10 wt%, (e) 15 wt%, and (f) 20 wt%.

### 3.2 Lap joint

The results of the apparent shear strength of aluminum specimens adhesively bonded with single lap joints are given in this section. GNP and  $\text{TiO}_2$  nanocomposites at five different weight fractions were considered adhesives for aluminum overlap joints. Figure 5 shows FL *versus* displacement of epoxy/ $\text{TiO}_2$  at 10 wt%.

As exhibited in Figure 5, ductile behavior was observed in epoxy/ $\text{TiO}_2$  overlap joints; the force *vs* displacement behavior started with a linear elastic increase to reach a force of 1.108 kN at 0.075 mm. Then, the force increased gradually along with a notable increase in the displacement, where it attained a maximum value of 1.877 kN at 0.567 mm, right before its failure. Figure 6

displays the FL and displacement behavior of the epoxy/GNP lap joint at 15 wt%.

As shown in Figure 6, the force *vs* displacement of the epoxy/GNP nanocomposite at 15 wt% increased linearly to reach a value of 1.553 kN at 0.119 mm. Then, it followed an ascending trend to exhibit its peak force of 3.081 kN at 0.806 mm, which was straight before its ductile failure. Although the nanoparticles were evenly distributed throughout the hardened resin, the TS of the corresponding modified resin likewise significantly decreased [40]. The poorer interlocking system caused by inadequate dispersion within the composite matrix may be the cause of the decreased bond strength shown in specimens.

Figure 7 shows the comparison of the impact of increasing the filler content on the maximum FL.

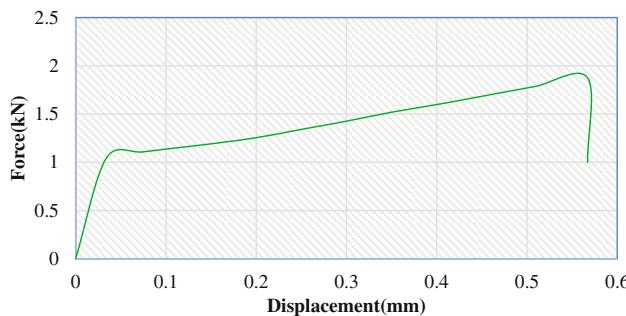


Figure 5: Force vs displacement of the epoxy/ $\text{TiO}_2$  lap joint at 10 wt%.

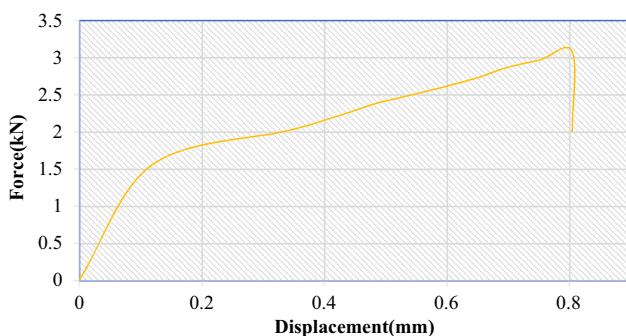


Figure 6: Force vs displacement of the epoxy/GNP lap joint at 15 wt%.

As displayed in Figure 7, the addition of 1 and 5 wt%  $\text{TiO}_2$  nanoparticles decreased the joint shear strength compared to pure epoxy, revealing average FLs of 1.73 and 1.57 kN, respectively. Further increasing the weight percentage of  $\text{TiO}_2$  led to an increase in the FL, which exhibited its peak value of 3.07 kN at 20 wt%. A common behavior was observed at 1 wt% GNP, where the joint FL decreased to 1.52 kN, which was around 60% lower than that of pure epoxy joints. Hence, increasing the GNP content contributed to increasing joint FL to reach its highest FL of 3.69 kN in the nanocomposite with 15 wt% GNP, followed by a decrease at 20 wt%. Throughout the considered compositions, 20 wt% revealed peak FL in  $\text{TiO}_2$  nanocomposites, whereas 15 wt% was an optimum composition for epoxy/GNP nanocomposites, which was also considered as highest FL in this study (3.69 kN). The opposite FL behavior of both nanocomposites at 20 wt% can be explained by adapting some of the important properties of nanocomposites, *i.e.*, matrix to particle interface quality. In other words, if the interaction between the matrix and nanoparticles is weak, the particles are unable to withstand a part of the applied external loads to the nanocomposite, thereby the yielding of amorphous glassy polymers switches from cavitation to shear, leading to a brittle to ductile transition. It is worth mentioning that there are two types of damage, *i.e.*, adhesion and cohesion, and their occurrence is strongly correlated to the

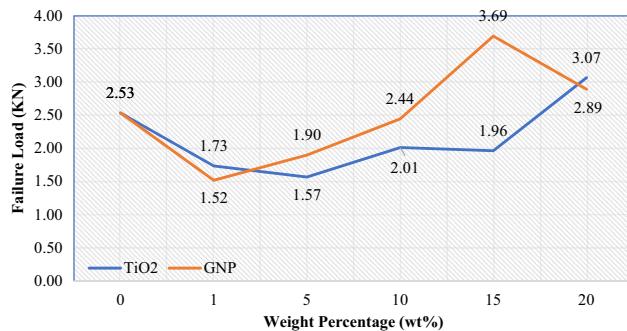


Figure 7: Joint FL of the epoxy/ $\text{TiO}_2$  and epoxy/GNP for 5 wt%.

homogeneous dispersion of nanoparticles in the epoxy matrix. Therefore, the adhesion strength can be affected by physical and chemical reactions between the epoxy and nanofiller materials.

### 3.3 Machine learning and deep learning

MLR, ANNs, ANFIS, and SVM were applied in the current study to define the most suitable machine learning method to predict shear force and TS of nanocomposites, as well as specify design spaces of corresponding parameters. The results of the aforementioned machine learning tools in predicting TS and overlap joint FL of GNP and  $\text{TiO}_2$  nanocomposites are listed in this section. A regression model was applied in this study by the curve fitting tool (Cftool) in MATLAB. Response surface models of GNP and  $\text{TiO}_2$  nanocomposites were projected via cubic polynomial prediction functions. Moreover, output data were TS and FL, while input data were weight percentage and fiber type. Furthermore, the response surface metamodel creates a surface fit, which considers the whole design space, providing the ability to predict responses using input parameters not considered in the experiment. Figure 8a and b shows the developed response surfaces.

The Cftool in MATLAB was used to develop the RSM for the TS and overlap FL of  $\text{TiO}_2$  and GNP nanocomposites. The RMSE, *R*-square adjusted, and sum of square error (SSE) were considered to evaluate goodness of fit. For proper surface fit, the values of SSE and RMSE should be as close as possible to 0. Meanwhile, the adjusted value of the *R*-square ranges from 0 and 1, and a good fit should be close to 1. Moreover, the function utilized to develop the response surface of the TS of nanocomposites is

$$f(x, y) = p_{00} + p_{10} \times x + p_{01} \times y + p_{11} \times x \times y + p_{02} \times y^2 + p_{12} \times x \times y^2 + p_{03} \times y^3,$$

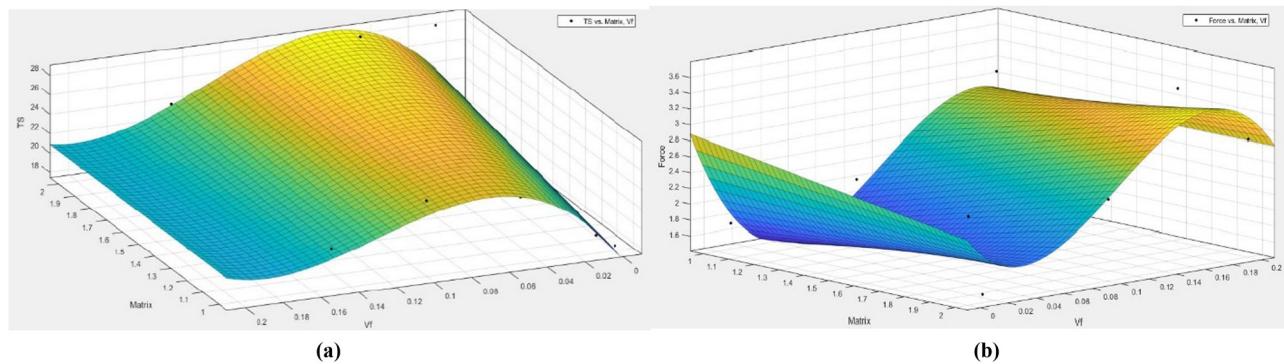


Figure 8: Response surface fitting for the (a) TS and (b) joint FL.

where  $x$  is the nanoparticle type,  $y$  is the weight percent, and  $p_{00} = 12.92$ ,  $p_{10} = 4.418$ ,  $p_{01} = 285.6$ ,  $p_{11} = -22.2$ ,  $p_{02} = -2,560$ ,  $p_{12} = 7.554$ , and  $p_{03} = 6,555$ . The goodness of fit attained an RMSE of 2.896, adjusted  $R$ -square of 0.3909,  $R$ -square of 0.7232, and SSE of 41.94.

For the RSM model of overlap joints FL, the applied function was

$$f(x, y) = p_{00} + p_{10} \times x + p_{01} \times y + p_{11} \times x \times y + p_{02} \times y^2 + p_{12} \times x \times y^2 + p_{03} \times y^3,$$

where  $x$  is the nanoparticle types,  $y$  is the weight percent, and  $p_{00} = 2.708$ ,  $p_{10} = -0.3022$ ,  $p_{01} = -62.74$ ,  $p_{11} = 22.09$ ,  $p_{02} = 555$ ,  $p_{12} = -97.93$ , and  $p_{03} = -1,213$ . In addition,  $SSE = 1.168$ ,  $RMSE = 0.4832$ , adjusted  $R$ -square = 0.4671, and  $R$ -square = 0.7578.

All components of the prediction model in this study were trained using the Levenberg–Marquardt algorithm, which exhibited a stable and swift convergence. Figure 9 illustrates the ANN model design: it consisted of 2 inputs, 10 hidden layers, and a single output.

A neural network fitting tool in MATLAB was utilized for the generation of the ANN models. Nanoparticle types and weight percentages were considered as input data, whereas the TS and FL were the model outputs. Experimental results of the tensile test and overlap shear test were considered to generate two corresponding models, and 70% of the data

were utilized for training, 15% for validation, and 15% for testing. Figure 10 shows the schemes of TS regressions for  $TiO_2$  and GNP nanocomposites for five different weight percentages, *i.e.*, 1, 5, 10, 15, and 20 wt%. This graph provides a correlation between experimental data (target) and ANN output data.

As shown in Figure 10, the dotted line denotes the best achievable correlation, while the solid line represents the correlation between output and target values. The overall regression coefficient of the TS ANN model was 0.93712 and that of the FL ANN model was 0.92341, which can be considered satisfactory as the values are close to 1. The MATLAB regression learner tool was used to generate the SVM model, and input parameters included two nanoparticle types and five weight percentages. Testing and training output datasets created by the FL SVM model are shown in Figure 11: yellow dots illustrate predicted data, while blue dots are the true data.

Furthermore, SVM models were trained using tensile and FL experimental results, Gaussian was considered as the kernel function, and fine Gaussian SVM was the selected preset. However, for the TS SVM model  $RMSE = 4.2839$ ,  $MSE = 18.352$ , and  $R$ -square = -0.36, while for the FL SVM model  $RMSE = 0.75973$ ,  $MSE = 0.57719$ , and  $R$ -square = -0.13. The considered ANFIS models include two inputs (nanoparticle type and weight percentage), and two and five membership

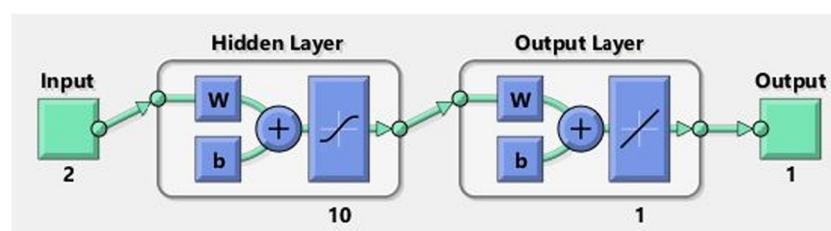


Figure 9: The ANN model structure.

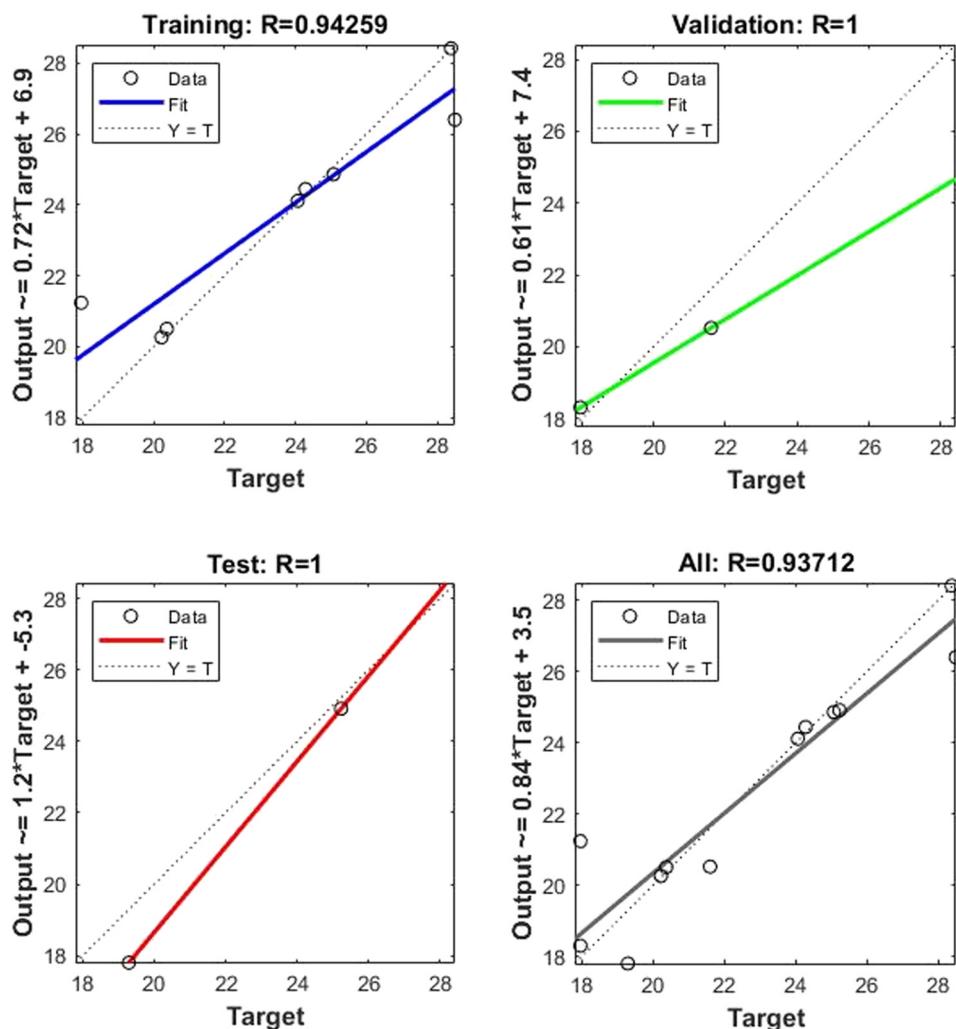


Figure 10: Regression plot of the ANN model.

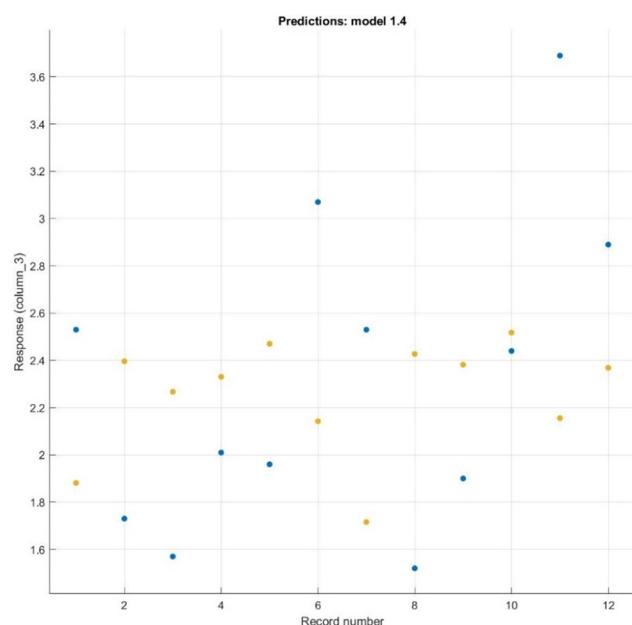


Figure 11: FL predict versus real SVM data.

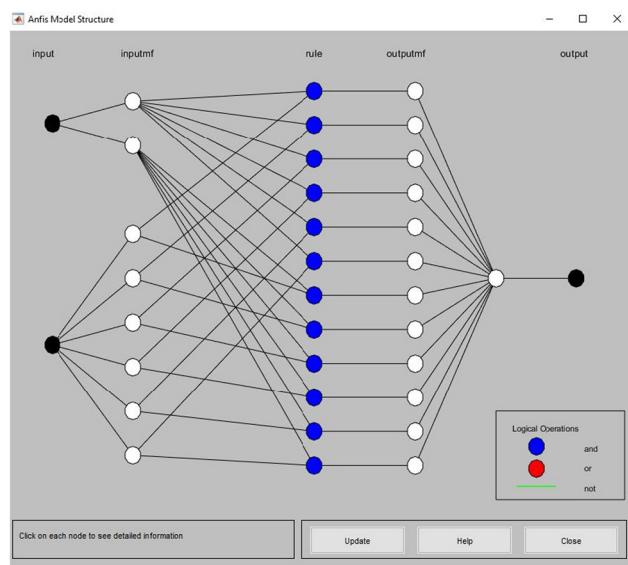
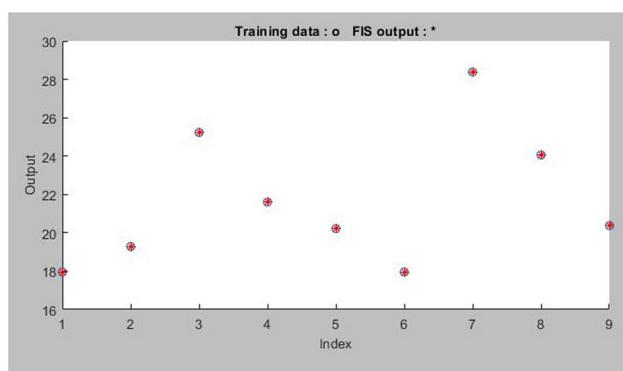


Figure 12: The ANFIS model structure.



**Figure 13:** Plot of ANFIS for TS prediction.

functions for the first and second inputs, respectively. Figure 12 illustrates the ANFIS model structure.

ANFIS models were generated using the MATLAB Neurofuzzy designer tool. The models were trained using 75% of

tensile and joint FL experimental results, yet the remaining 25% were for testing. The Gaussmf membership function with fixed output was used to create the FIS model. Training was completed at epoch 1, with an average testing error of  $1.5265 \times 10^{-6}$ . Figure 13 displays the TS ANFIS model plot, which shows the FIS output and training data.

### 3.3.1 ML vs experimental results

Table 1 lists TS results and predicted data using machine learning models of  $\text{TiO}_2$  and GNP nanocomposites. As shown in Table 1, the predicted TS using ANN showed a drastic compliance with experimental TS results, exhibiting 3.67% prediction error, while TS values obtained from MLR, SVM, and ANFIS revealed a significant agreement with errors of 5.27, 5.37, and 6.74%, respectively. This highlights the ability of these models to be trained and

**Table 1:** Predicted TS values using MLR, ANN, SVM, and ANFIS

Matrix	Wt%	Experiment (MPa)	MLR (MPa)	ANN (MPa)	SVM (MPa)	ANFIS (MPa)
$\text{TiO}_2$	0	17.96	17.34	18.3	18.9419	17.96
$\text{TiO}_2$	1	19.29	19.72	17.8	19.5465	19.29
$\text{TiO}_2$	5	24.28	24.95	24.44	23.8802	22.3731
$\text{TiO}_2$	10	25.23	24.71	24.91	24.8235	25.23
$\text{TiO}_2$	15	21.60	21.54	20.52	22.0001	21.6
$\text{TiO}_2$	20	20.22	20.36	20.26	20.623	20.22
GNP	0	17.96	21.76	21.24	22.5004	17.96
GNP	1	28.49	23.92	26.4	23.8548	21.1485
GNP	5	28.38	28.27	28.41	27.2574	28.38
GNP	10	25.07	26.98	24.86	24.6729	13.2474
GNP	15	24.06	22.80	24.11	23.667	24.06
GNP	20	20.37	20.64	20.5	20.7634	20.37
Error%			5.27%	3.67%	5.37%	6.74%

**Table 2:** Predicted joint FL values using MLR, ANN, SVM, and ANFIS

Matrix	Wt%	Experiment (kN)	MLR (kN)	ANN (kN)	SVM (kN)	ANFIS (kN)
$\text{TiO}_2$	0	2.53	2.41	2.475	2.459	2.53
$\text{TiO}_2$	1	1.73	2.04	1.909	2.2343	1.73
$\text{TiO}_2$	5	1.57	1.36	1.573	1.6695	1.57
$\text{TiO}_2$	10	2.01	1.70	2.014	2.0799	1.3538
$\text{TiO}_2$	15	1.96	2.50	1.959	2.0263	1.96
$\text{TiO}_2$	20	3.07	2.85	2.664	3.0036	3.07
GNP	0	2.53	2.10	1.952	2.4637	2.53
GNP	1	1.52	1.95	1.937	2.2611	1.52
GNP	5	1.90	1.92	1.901	1.9664	1.5719
GNP	10	2.44	2.63	2.223	2.3993	2.44
GNP	15	3.69	3.31	3.692	3.0663	3.69
GNP	20	2.89	3.05	2.887	2.8201	0.8201
Error%			13.00%	7.21%	10.33%	10.24%

used for the prediction of the TS of nanocomposites. Moreover, GNP nanocomposite at 5 wt% exhibited the highest TS between all implemented machine learning models. The most convenient machine learning technique for the prediction of the TS of nanocomposites is ANN as it exhibited the least prediction error of 3.67%. Table 2 displays experimental overlap FL results and predicted data using machine learning techniques of  $\text{TiO}_2$  and GNP nanocomposites.

As shown in Table 2, MLR, SVM, and ANFIS models exhibited prediction errors of 13.00, 10.33, and 10.24%, respectively. However, the overlap FL obtained from the ANN model showed the least prediction error, a value of 7.21%, which therefore can be considered the most suitable prediction approach across all the utilized models for predicting the overlap joint FL of the nanocomposites. Peak FL values were observed in the GNP nanocomposite joints at 15 wt%.

## 4 Conclusion

In this study, the tensile properties of  $\text{TiO}_2$  and GNP nanocomposite were evaluated following the ASTM standard D638-14. Moreover, the considered nanocomposites were utilized as an adhesive for an overlap joint of two A5055 aluminum sheets. Then, their apparent shear strength properties were evaluated following the ASTM D1002-01 standard. Five weight percentages of different nanoparticles were considered (1, 5, 10, 15, and 20 wt%). Furthermore, for evaluating the capability of deep learning and machine learning models in predicting the mechanical characteristics of nanocomposites, the results obtained from the experiment were applied to train and test the considered models, *i.e.*, ANFIS, SVM, MLR, and ANN.

However, a maximum TS behavior was witnessed in epoxy/GNP at 1 wt%, which exhibited a value of 28.49 MPa, whereas for the overlap joints FL, the highest value recorded in this study was 3.69 kN, which was observed in epoxy with 15 wt% GNP. ANN can be considered a convenient machine learning model for the prediction of the joint failure shear load and TS of nanocomposites since it showed the least prediction error of 3.67% in predicting TS and 7.21% in the prediction of FL.

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