

Research Article

Dunzhu Lu, Ying Zeng, Qizhi Yan, Qiyu Chen, Weiliang Ma, Xiao Luo, Ming Xu, Xiaosheng Yang* and Peining Li*

Tunable hyperbolic polaritons with plasmonic phase-change material In₃SbTe₂

https://doi.org/10.1515/nanoph-2023-0911 Received December 12, 2023; accepted February 4, 2024; published online February 13, 2024

Abstract: Hyperbolic polaritons that originate from the extreme optical anisotropy in van der Waals (vdW) crystals have gained much attention for their potential in controlling nanolight. For practical use, there has been a strong interest to develop various manipulation strategies to customize the propagation of hyperbolic polaritons on a deeply sub-diffractional scale. In this regard, phase-change materials (PCMs) that possess two phases with different refractive indices offer suitably a tunable dielectric environment. Here, we report on the tuning of hyperbolic phonon polaritons in natural vdW crystals, hexagonal boron nitride (hBN), and alpha-phase molybdenum trioxide (α -MoO₃), using the plasmonic phase-change material In₃SbTe₂ (IST). Unlike conventional PCMs whose both phases are dielectric, IST features a metallic crystalline phase that is stable at room temperature. The coupling between polaritons with their mirror charges in the underneath crystalline IST triggers an even stronger field confinement for polaritons.

crystalline IST, we show an all-optical material platform in which crystalline IST boundaries efficiently excite and focus hyperbolic phonon polaritons in $\alpha\text{-MoO}_3$. Our experiments highlight the possibility to obtain new degrees of freedom in polariton engineering with plasmonic PCMs, thereby expanding the toolkit of tunable nanophotonics with flexible, on-demand fabrication and reconfiguration capabilities.

Moreover, benefited from the metallicity of laser-writable

Keywords: hyperbolic polaritons; van der Waals materials; phase-change materials; In₃SbTe₂

1 Introduction

Polaritons in natural van der Waals (vdW) materials provide vast opportunities for manipulating light on the nanoscale. This manipulation is achieved by exploiting their enhanced electromagnetic energy concentration, extreme anisotropic propagation, and strong inherent nonlinearities [1]-[10]. One particular research focus lies with hyperbolic phonon polaritons (HPPs) originating from the coupling between photons and lattice vibrations. These phenomena occur in vdW materials with permittivities of opposite signs along different directions, e.g., hexagonal boron nitride (hBN) [4], [11]–[13] and alpha-phase molybdenum trioxide (α -MoO₃) [1], [14], [15]. HPPs feature ultra-confined and highly directional light beam with hyperbolic dispersion, enabling novel nanophotonic applications such as hyperlensing [16], negative refraction [17], [18], molecular sensing [19]-[22], and nanofocusing [23], [24].

In the ongoing quest for reconfigurable control of HPP properties [18], [25], [26], phase change materials (PCMs) emerge as an appealing platform. Their optical properties change significantly between the amorphous and crystalline phases subject to external stimuli, such as temperature control, laser pulses, or electrical bias [27]–[29]. As the electric field profile of polaritons is directly influenced by the dielectric environment, nonvolatilely switchable chalcogenide PCMs like $Ge_3Sb_2Te_6$ have been often employed in

Dunzhu Lu, **Qizhi Yan**, **Qiyu Chen and Weiliang Ma**, Wuhan National Laboratory for Optoelectronics and School of Optical and Electronic Information, Huazhong University of Science and Technology, Wuhan 430074, China; and Optics Valley Laboratory, Hubei 430074, China. https://orcid.org/0000-0001-5440-5888 (Q. Yan)

Ying Zeng, School of Information Engineering, Wuhan University of Technology, Wuhan 430070, China; National Engineering Research Center of Fiber Optic Sensing Technology and Networks, Wuhan University of Technology, Wuhan 430070, China; and Hubei Key Laboratory of Broadband Wireless Communication and Sensor Networks, Wuhan University of Technology

Xiao Luo and Ming Xu, School of Integrated Circuits, Huazhong University of Science and Technology, Wuhan 430074 China. https://orcid.org/0000-0002-2730-283X (M. Xu)

nology, Wuhan 430070, China

^{*}Corresponding authors: Xiaosheng Yang and Peining Li, Wuhan National Laboratory for Optoelectronics and School of Optical and Electronic Information, Huazhong University of Science and Technology, Wuhan 430074, China; and Optics Valley Laboratory, Hubei 430074, China, E-mail: yang_xs@hust.edu.cn (X. Yang), lipn@hust.edu.cn (P. Li). https://orcid.org/0000-0002-7632-0401 (X. Yang). https://orcid.org/0000-0003-3836-3803 (P. Li)

a heterostructure with the hyperbolic material to actively tune HPPs [28]. However, the tuning range of HPP properties in this manner is limited by the permittivity contrast between the crystalline and amorphous phases of PCM. Moreover, to effectively launch HPP in the vdW/PCM heterostructure, an edge of the vdW material or a lithographically fabricated metallic structure is required [30], which makes the applications of polaritonic devices less flexible.

Here, we study the all-optical control of HPP propagation using "plasmonic PCM" In₃SbTe₂ (IST) [31]-[38] in vdW/PCM heterostructures. IST has a crystalline phase (cIST) that features properties similar to noble metal, while its amorphous phase (aIST) remains dielectric, thus providing a significant optical contrast over infrared and terahertz spectral range [39]. By introducing an IST film in conjunction with hBN (α -MoO₃) that support in-plane isotropic (anisotropic) HPPs, HPPs in the vdW material hybridize with their electromagnetic image charges in the metallic cIST [40]-[43]. This hybrid polariton mode, referred to as hyperbolic "image polaritons" (HIPs) [42]-[44], not only enhances the field confinement but also provides reconfigurability of the polaritonic response. We also demonstrated that, with laser-switched crystalline microdomains in amorphous IST, the polariton propagation is conveniently manipulated in which polaritons can be effectively launched from a cIST boundary or focused by the end of a cIST antenna.

2 Results and discussion

We first investigate the polariton propagation in a heterostructure that comprises an hBN layer (62 nm), a natural vdW material that supports in-plane isotropic HPPs, on an IST film (80 nm). In the latter, crystalline and amorphous phases coexist (Figure 1A, see Section 4). To visualize polaritons, we used a scattering-type scanning near-field optical microscope (s-SNOM). Under incident infrared field $E_{\rm in}$, the light is scattered from the atomic force microscope (AFM) tip of s-SNOM, thereby the polariton wavefront E_s is mapped as a function of tip position. Figure 1B shows the measured AFM topography image and near-field optical image of two adjacent regions of hBN/cIST and hBN/aIST. In the topography image, the dark and bright areas correspond to IST film and IST covered by hBN, respectively. While no significant difference in height is found between cIST (red dashed trapezoid) and aIST (blue dashed trapezoid) regions, the near-field image measured at 1460 cm⁻¹ shows typical polariton fringe patterns of different periodicities on hBN/IST in these two regions.

To quantify the difference of propagating polaritons due to IST's different phases in the heterostructure, we extracted near-field amplitude profiles along the white lines from Figure 1B and the results are shown in Figure 1C. Here, two different excitation sources for polaritons need to be considered [30]: the first is the hBN edge and the second is the AFM tip, which induces polariton interference fringes with spacing λ_n and $\lambda_n/2$, respectively, where λ_n is the polariton wavelength. With this analytical model, we fit the data to find the complex wavevectors k of polaritons on hBN/cIST and hBN/aIST to be $(11.1 + 0.4i)k_0$ and $(8.1 + 0.5i)k_0$. The error bars are determined from the half width at half maximum (HWHM) obtained from the Fourier transform [45], [46]. These values indicate that the polaritons propagating on hBN/cIST exhibit larger field confinement compared to that on hBN/aIST. Meanwhile, the propagation loss of the former is slightly smaller judging from the imaginary parts. We note that polariton wavefronts observed at the boundary region between cIST and aIST are predominantly edge-excited. We also changed the illumination frequency between 1420 cm⁻¹ and 1460 cm⁻¹, namely within the type-II Reststrahlen band (1360-1610 cm⁻¹) of hBN [4], and extracted the wavevector k of polaritons on hBN/cIST and hBN/aIST with the same approach from nearfield images. The experimental data (squares and triangles in Figure 1D) are in good agreement with the analytically calculated dispersion (Section 4).

To understand the different polariton propagation constants on hBN/cIST and hBN/aIST heterostructures, we conducted numerical simulations (Section 4) for the electric field distribution across interface, see Figure 1E. With dielectric aIST, the electric field of HPP penetrates into both air and aIST. With metallic cIST, the presence of image charges in cIST induces the so-called HIP mode [44], in which the penetration of electric field into cIST is strongly suppressed and companied by a phase inversion. The metallicity of cIST is also evident when the figure of merit (FOM) of propagating polaritons, Re(k)/Im(k), is compared for heterostructures of hBN on various materials as a function of frequency (Figure 1F). Here, we have considered cIST, crystalline GST (cGST), crystalline GeTe (cGeTe), or gold as substrate for an 80-nm-thick hBN layer. Comparing the FOM values of those, cIST has the lowest relative loss among the different crystalline PCMs and demonstrate its potential as substitution for noble metals in infrared nanophotonics.

We now turn to the polariton propagation in an α -MoO₃/IST heterostructure. Comparing with hBN that supports in-plane isotropic HPPs, α-MoO₃ supports in-plane anisotropic HPPs with rich phonon modes and advantages of low loss and long lifetime. The in-plane dispersion of HPPs in α -MoO₃ under the thin-film approximation [1] ($k_0d \ll$

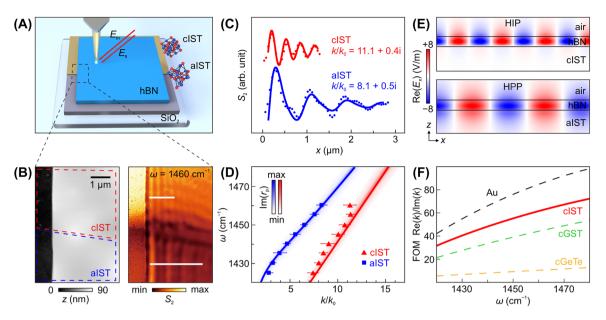


Figure 1: Near-field nanoimaging and dispersion analysis for phonon polaritons in hBN/IST heterostructure. (A) Schematic of s-SNOM nanoimaging experiment. (B) AFM topography image and near-field amplitude image ($\omega = 1460 \text{ cm}^{-1}$) of the hBN (62 nm)/IST (80 nm) heterostructure. (C) Near-field amplitude profiles extracted along the white lines in (B) and corresponding fits. (D) Dispersion of phonon polaritons in hBN/cIST and hBN/aIST. False-color plot represents imaginary part of the Fresnel reflection coefficient, $\text{Im}(r_p)$, whose maxima are shown as solid lines. Triangles and squares are experimental data extracted from monochromatic s-SNOM measurements. (E) Cross-sectional electric field distribution $\text{Re}(E_x)$ of the HIP (HPP) mode propagating along hBN/cIST (hBN/aIST). (F) Theoretical FOM values for hBN/Au and different hBN/PCM heterostructures.

1, where d is the α -MoO₃ thickness) can be analytically expressed as [47]:

$$k = \frac{\rho}{k_0 d} \left[\arctan\left(\frac{\epsilon_1 \rho}{\epsilon_z}\right) + \arctan\left(\frac{\epsilon_2 \rho}{\epsilon_z}\right) + \pi l \right], \quad l = 0, 1, 2 \dots$$

where $\rho = i\sqrt{\varepsilon_z/(\varepsilon_x \cos^2\theta + \varepsilon_y \sin^2\theta)}$, θ is the angle between the x axis and the in-plane HPP wavevector k, $\varepsilon_{x,y,z}$ is the permittivity of α-MoO₃ along different orthogonal axis, and $\varepsilon_1(\varepsilon_2)$ is the permittivity of the media on top (at bottom) of the α -MoO₃ film. We performed near-field nanoimaging experiment for a rectangle α-MoO₃ flake with thickness d = 240 nm, under which both aIST and cIST regions are present. As shown in Figure 2A, the near-field amplitude image taken at 990 cm⁻¹ shows periodic fringe patterns parallel to all flake edges along the [100] (x axis) and [001] (y axis) directions, as illustrated in the enlarged views (inset). In contrast, in Figure 2B, the image taken at 885 cm⁻¹ shows similar fringe patterns exclusively parallel to the flake edge along the [001] direction. These results are in good agreement with previous reports on phonon polaritons of α-MoO₃ exhibiting in-plane elliptic and hyperbolic dispersions in the upper (958-1010 cm⁻¹) and lower (820–972 cm⁻¹) Reststrahlen bands [1], respectively.

To quantitatively compare the difference of phonon propagation properties for α -MoO₃/aIST and α -MoO₃/cIST,

we took multiple monochromatic s-SNOM images. Using the same analytical model for hBN that considers both the edge- and tip-excited modes, we extracted the wavevector values from line scans perpendicular to flake edges. The polariton dispersions for both cases and both Reststrahlen bands were plotted in Figure 2C and D, and an overall good agreement between theory and experiment is found. Notably, in the upper Reststrahlen band (Figure 2C), the inplane anisotropy of phonon polaritons is reflected by dispersion curves of different slopes along the [100] and [001] directions. However, the difference of the in-plane polariton wavevector k for aIST and cIST cases are not pronounced. In the lower Reststrahlen band (Figure 2D), the polariton wavevectors supported by α -MoO₃/aIST and α -MoO₃/cIST heterostructures differ more, with the latter being clearly larger.

We also simulated the in-plane field distributions around an electric dipole source for α -MoO₃/aIST and α -MoO₃/cIST, see Figure 2E. The predicted wavefronts along the x axis for α -MoO₃/cIST (Figure 2E, right panel) carries larger momentum than that for α -MoO₃/aIST (Figure 2E, left panel), corresponding to the HIP and HPP mode in analogy to our analysis for hBN/cIST and hBN/aIST heterostructures (Figure 1E). The enhanced field confinement due to the presence of image charges in cIST is also evident in the calculated isofrequency contours for α -MoO₃/cIST and

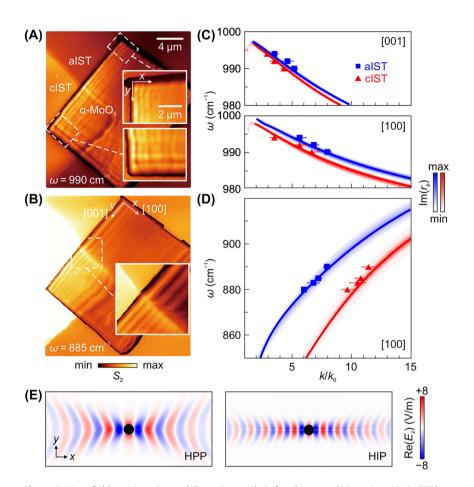


Figure 2: Near-field nanoimaging and dispersion analysis for phonon polaritons in α -MoO $_3$ /IST heterostructure. (A, B) Near-field amplitude image of the α -MoO $_3$ (240 nm)/IST (80 nm) heterostructure, measured at (A) ω = 990 cm $^{-1}$ and (B) ω = 885 cm $^{-1}$. Insets show enlarged view of the dashed rectangular regions. (C, D) Dispersion of phonon polaritons in α -MoO $_3$ /IST within the (C) upper and (D) lower Reststrahlen bands. Labels on the top right corner mark the corresponding orientations of α -MoO $_3$. False-color plot represents imaginary part of the Fresnel reflection coefficient, Im(r_p), whose maxima are shown as solid lines. Triangles and squares are experimental data extracted from monochromatic s-SNOM measurements. (E) Simulated in-plane field distributions (ω = 885 cm $^{-1}$) around an electric dipole source for HPP mode of α -MoO $_3$ /aIST and HIP mode of α -MoO $_3$ /cIST.

 α -MoO₃/aIST at ω = 885 cm⁻¹, see Figure 3A. Here, \mathbf{k}_{θ} indicates an in-plane polariton wavevector with angle θ with respect to the x axis. When θ equals zero, the momentum difference of polaritons for α -MoO₃/cIST and α -MoO₃/aIST is maximized, with the former being approximately 1.3 times larger. As θ increases, this difference is gradually suppressed and both cases now enable a wavevector approaching infinity with an ultra-short wavelength. From a practical perspective, it is worth to compare the FOM values (Re(k)/Im(k)) of polaritons as a function of azimuthal angle for heterostructures of α -MoO₃ on various PCMs (cIST, aIST, cGST, and cGeTe), as displayed by the polar plot in Figure 3B. For all types of PCM-based heterostructure calculated at $\omega = 885 \text{ cm}^{-1}$, the phonon polaritons of α -MoO₃ are available within the range between $\theta = \pm 60^{\circ}$. The α -MoO₃/cIST heterostructure has overall largest FOM values

for polaritons, except being marginally smaller than the $\alpha\text{-MoO}_3\text{/Au}$ heterostructure.

Having established that phonon polaritons of different wavevectors can be excited from the flake edge of $\alpha\text{-MoO}_3$ on cIST or aIST, we further investigate polaritons that are launched by the boundary of cIST structures buried under $\alpha\text{-MoO}_3$. Figure 4A shows numerical simulations for polaritons launched from a linear boundary made of cIST, where beneath $\alpha\text{-MoO}_3$, the cIST and aIST regions are placed next to each other. The linear polariton wavefronts launched by the cIST edge are visible, whose amplitude gradually attenuates with increasing distance x. In contrast, with a rod antenna-like cIST, the polariton wavefronts on aIST are strongly deformed, see Figure 4B. Here, the interference of highly directional HPPs from different spatial position along the convex edge of cIST leads to the focus formation at a

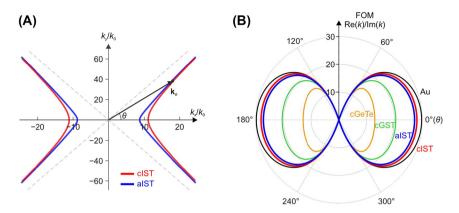


Figure 3: IFCs and FOM values of phonon polaritons for α -MoO₂/IST. (A) IFCs of phonon polaritons for α -MoO₂/cIST and α -MoO₃/aIST. Dashed lines indicate the two asymptotes. Black arrow represents the wavevector of phonon polaritons with angle θ from the x axis of α -MoO₃. (B) Comparing theoretical FOM values for α -MoO₃/Au heterostructure and various α -MoO₃/PCM heterostructures as a function of angle θ . Au and PCMs were modeled as infinite thick.

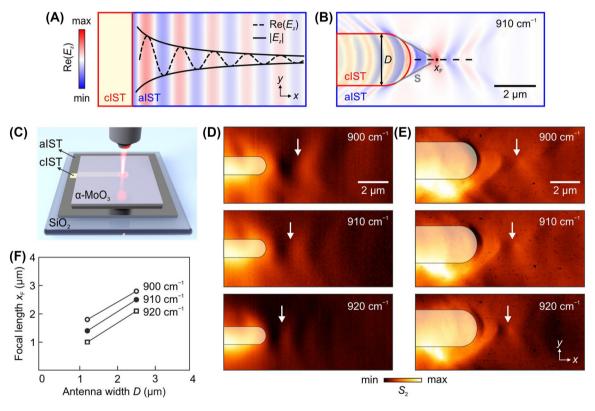


Figure 4: Launching and focusing HPPs with cIST boundary under α-MoO₃. (A) Simulated electric field distribution of HPPs launched by a linear cIST boundary. (B) Simulated electric field distribution of HPPs focused by a rod antenna made of cIST. (C) Schematic of laser-switching the cIST rod antenna in an aIST film under α-MoO₂. (D, E) Near-field amplitude images of the HPPs excited and focused by cIST rod antennas with widths (D) 1.2 μm and (E) 2.5 μm. Yellow-shaded areas represent where the crystallization of IST takes place. White arrows mark the foci positions. The illumination frequency was varied from 900 cm $^{-1}$ to 920 cm $^{-1}$ in experiment. The thicknesses of α -MoO $_3$ and IST were 410 nm and 80 nm, respectively. (F) Focal lengths x_F extracted from the experimental near-field images.

distance x_F from the antenna apex. This is similar to the case when a metallic (e.g., gold) disk is placed on top of α -MoO₃ [23], but the advantage brought by our approach is that on-demand reconfiguration of HPPs is now possible with any thin hyperbolic medium on IST.

To experimentally validate our proposal, we transferred an α -MoO $_3$ flake on an 80-nm-thick aIST film, in which rod antennas of cIST were subsequently laserswitched (Figure 4C, see Section 4). The orientation of the cIST antenna is aligned such that it is parallel to the [100]

direction (x axis) of α-MoO₃. In Figure 4D and E, cIST antennas of different widths, $D = 1.2 \mu m$ and 2.5 μm , were imaged by s-SNOM at various illumination frequency. Here, yellowdashed areas represent where the crystallization of IST takes place. Away from the cIST antenna's right end, the edge-launched HPP wavefronts propagate along the x direction, first narrowing to a focal point and then diverging as moving forward. This behavior is in agreement with the numerical simulation shown in Figure 4B and resembles the free-space focusing by conventional convex lenses. Finally, in Figure 4F, the dependency of focal length $x_{\rm F}$ on the antenna width and illumination frequency is shown. As the antenna width increases from 1.2 μ m to 2.5 μ m, $x_{\rm F}$ extends by approximately 1 µm. As the illumination frequency increases from 900 cm⁻¹ to 920 cm⁻¹, which correspond to a wavelength decrease of 0.24 μ m, x_F shrinks by 0.7-0.8 µm. We note that in Figure 4D and E, the brighter appearance of lower parts of antenna is due to an experimental artefact (see Figure S1). Our results show that both the size of phase-changed area as well as the incident infrared light can be utilized to fine-tune the planar subwavelength focusing of HPPs.

3 Conclusions

We have experimentally presented a reconfigurable heterostructure platform that uses the plasmonic PCM IST for tuning in-plane isotropic or anisotropic HPPs in hBN or α-MoO₃. These results underscore the universality of IST in manipulating HPPs, not confined to a specific type of hyperbolic behavior. In particular, the metallicity of crystalline IST introduces a new avenue for controlling polariton properties. On one hand, the coupling between polaritons in the hyperbolic medium and their image charges in crystalline IST, known as the HIP mode, enhances the optical field compression of conventional HPP modes. On the other, we showcased the ability to launch polaritons from the edge of highly conductive crystalline IST, which provides an alternative to using the vdW crystal edge or a metal antenna as the polariton launcher. Furthermore, we demonstrated that a rod antenna made of crystalline IST with a convex boundary achieves the focusing of HPPs without the need for lithography. We anticipate that, as the dielectric-to-metal phase transition in IST is triggered by laser pulses in a nonvolatile fashion, microstructures of IST with arbitrary shapes [48] can be engineered for on-demand, customized manipulation and steering of polaritons.

4 Methods

4.1 Sample fabrication

In₃SbTe₂ (IST) was deposited on 2-mm-thick SiO₂ substrates using stoichiometric targets by direct current magnetron sputtering $(2 \times 10^6 \text{ mbar background pressure}, 20 \text{ s.c.c.m.}$ Ar flow, 0.15 nm/s deposition rate). The thickness of IST film was controlled by a stylus profiler (Brucker DektakXT) and the as-deposited IST films were in the amorphous state. Isotopically (10^B) enriched hBN flakes were first prepared by the mechanical exfoliation using blue Nitto tape, before being transferred onto a polydimethylsiloxane stamp by a second exfoliation with the tape. We inspected the surface of the stamp under an optical microscope to identify the desired flake and transferred it onto an 80-nm-thick IST film on clean SiO₂ substrates. α-MoO₃ flakes were prepared and transferred with the same procedure.

4.2 Laser printing

A home-built setup was used to pattern the IST film by focusing a laser beam through a ×20 or ×50 objective (numerical aperture both 0.4) on the sample surface. A nanosecond laser diode source with a central wavelength of 660 nm provides single pulses with tunable output power up to 400 mW and pulse duration from 1 ns to 10 μ s. To crystallize the IST film, laser pulses of 70 mW power and 80 ns duration were used. An x-y movable sample stage with maximum range of 50 mm in each direction and minimal step size of 700 nm was used to prepare the cIST areas (Figures 1B and 2A, B) and cIST antennas (Figure 4D and E). The scanning speed was set at 1-4 mms⁻¹, while the repetition rate of the pulsed laser was not less than 3 kHz. While experimentally it is viable to write cIST after the transfer of vdW materials, for experiments in this work, the writing of cIST was performed first.

4.3 Near-field nanoimaging

For the infrared near-field nanoimaging, we employed a commercial s-SNOM system from Neaspec GmbH based on a tapping-mode AFM. The Pt-coated metallized tip (ArrowNCPt, NanoWorld) oscillates vertically with an amplitude of approximately 70 nm at a frequency $\Omega \approx 270$ kHz. The tip was illuminated by p-polarized light from a tunable continuouswave quantum cascade laser. When the illuminated tip approaches the sample surface, the near-field interaction modifies the tip-scattered light, which is then recorded with a pseudo-heterodyne Michelson interferometer. To suppress background contribution induced from the tip shaft and distant sample, the detector signals are demodulated at a higher harmonic frequency $n\Omega$ ($n \ge 2$), yielding the near-field optical amplitude S_n and phase φ_n . For the near-field amplitude images shown in this manuscript, the second-order signal S_2 was used.

4.4 Numerical simulations

Simulations were performed using finite-element method in the frequency domain with COMSOL Multiphysics. We used two-dimensional mode solver to calculate the electric field distribution across interface of hBN/IST heterostructures in Figure 1E. Wave optics module was used to obtain the in-plane field distributions around an electric dipole source and cIST rod antennas shown in Figures 2E and 4A, B, respectively. We used the transfer matrix method to calculate the eigenmodes of hyperbolic polaritons for various heterostructures. These eigenmodes are located in the poles of the Fresnel reflection coefficients for p-polarized light, $r_{\rm pp}$. We determine the poles by numerically solving the equation (1)/Abs $(r_{\rm nn}) = 0$. For spatially decaying modes, we considered real-valued frequencies and complex momenta. The dielectric permittivities of hBN, α-MoO₃, and IST were modelled after references [15], [20], [32].

Acknowledgments: We thank the Optoelectronic Micro & Nano Fabrication and Characterizing Facility, Wuhan National Laboratory for Optoelectronics of Huazhong University of Science and Technology for the support in sample fabrication.

Research funding: National Natural Science Foundation of China (62075070, 62174060); National Key Research and Development Program of China (2021YFA1201500, 2022YFB3807200); Hubei Provincial Natural Science Foundation of China (2022CFA053); Innovation Fund of WNLO.

Author contributions: D.L. and Y.Z. contributed equally to this manuscript. X.Y. and P.L. conceived the study. D.L., Q.C., and X.L. fabricated the samples. D.L. performed the measurements with the help of Y.Z. and W.M. D.L. and Q.Y. performed the theory analysis and simulations. M.X., X.Y., and P.L. coordinated and supervised the work. D.L., Y.Z., and X.Y. wrote the manuscript with input from all other co-authors. All authors have accepted responsibility for the entire content of this manuscript and approved its submission.

Conflict of interest: Authors state no conflicts of interest. Data availability: The datasets generated and/or analysed during the current study are available from the corresponding author upon reasonable request.

References

- [1] W. Ma, et al., "In-plane anisotropic and ultra-low-loss polaritons in a natural van der Waals crystal," Nature, vol. 562, no. 7728, p. 557,
- [2] D. N. Basov, A. Asenjo-Garcia, P. J. Schuck, X. Y. Zhu, and A. Rubio, "Polariton panorama," Nanophotonics, vol. 10, no. 1, pp. 549 – 577,
- [3] P. Li, et al., "Infrared hyperbolic metasurface based on nanostructured van der Waals materials," Science, vol. 359, no. 6378, p. 892, 2018.
- [4] S. Dai, et al., "Tunable phonon polaritons in atomically thin van der Waals crystals of boron nitride," Science, vol. 343, no. 6175, pp. 1125-1129, 2014.
- [5] S. Dai, et al., "Subdiffractional focusing and guiding of polaritonic rays in a natural hyperbolic material," Nat. Commun., vol. 6, no. 1, p. 6963, 2015.
- [6] J. D. Caldwell, et al., "Sub-diffractional volume-confined polaritons in the natural hyperbolic material hexagonal boron nitride," Nat. Commun., vol. 5, no. 1, p. 5221, 2014.

- [7] Y. Wu, et al., "Manipulating polaritons at the extreme scale in van der Waals materials," Nat. Rev. Phys., vol. 4, no. 9, pp. 578-594, 2022
- [8] W. Ma, et al., "Ghost hyperbolic surface polaritons in bulk anisotropic crystals," Nature, vol. 596, no. 7872, p. 362, 2021.
- [9] C. Hu, et al., "Source-configured symmetry-broken hyperbolic polaritons," eLight, vol. 3, no. 1, p. 14, 2023.
- [10] X. Zhang, et al., "Ultrafast anisotropic dynamics of hyperbolic nanolight pulse propagation," Sci. Adv., vol. 9, no. 34, p. eadi4407, 2023.
- [11] P. Li, et al., "Hyperbolic phonon-polaritons in boron nitride for near-field optical imaging and focusing," Nat. Commun., vol. 6, no. 1, p. 7507, 2015.
- [12] P. Li, et al., "Optical nanoimaging of hyperbolic surface polaritons at the edges of van der Waals materials," Nano Lett., vol. 17, no. 1, pp. 228-235, 2017.
- [13] F. J. Alfaro-Mozaz, et al., "Nanoimaging of resonating hyperbolic polaritons in linear boron nitride antennas," Nat. Commun., vol. 8, no. 1, p. 15624, 2017.
- [14] Z. Zheng, et al., "A mid-infrared biaxial hyperbolic van der Waals crystal," Sci. Adv., vol. 5, no. 5, p. eaav8690, 2019.
- [15] G. Álvarez-Pérez, et al., "Infrared permittivity of the biaxial van der Waals semiconductor α -MoO₃ from near- and far-field correlative studies," Adv. Mater., vol. 32, no. 29, p. 1908176, 2020.
- [16] M. He, et al., "Ultrahigh-resolution, label-free hyperlens imaging in the mid-IR," Nano Lett., vol. 21, no. 19, pp. 7921-7928, 2021.
- [17] G. Álvarez-Pérez, et al., "Negative reflection of nanoscale-confined polaritons in a low-loss natural medium," Sci. Adv., vol. 8, no. 29, p. eabp8486, 2022.
- [18] H. Hu, et al., "Gate-tunable negative refraction of mid-infrared polaritons," Science, vol. 379, no. 6632, pp. 558-561, 2023.
- [19] Z. Yuan, R. Chen, P. Li, A. Y. Nikitin, R. Hillenbrand, and X. Zhang, "Extremely confined acoustic phonon polaritons in monolayer-hBN/metal heterostructures for strong light - matter interactions," ACS Photonics, vol. 7, no. 9, pp. 2610-2617, 2020.
- [20] A. Bylinkin, et al., "Real-space observation of vibrational strong coupling between propagating phonon polaritons and organic molecules," Nat. Photonics, vol. 15, no. 3, pp. 197-202, 2021.
- I. Dolado, et al., "Remote near-field spectroscopy of vibrational strong coupling between organic molecules and phononic nanoresonators," Nat. Commun., vol. 13, no. 1, p. 6850, 2022.
- [22] T. Qin, W. Ma, T. Wang, and P. Li, "Phonon polaritons in van der Waals polar heterostructures for broadband strong light – matter interactions," *Nanoscale*, vol. 15, no. 28, pp. 12000 – 12007,
- [23] J. Martín-Sánchez, et al., "Focusing of in-plane hyperbolic polaritons in van der Waals crystals with tailored infrared nanoantennas," Sci. Adv., vol. 7, no. 41, p. eabj0127, 2021.
- [24] Z. Zheng, et al., "Controlling and focusing in-plane hyperbolic phonon polaritons in $\alpha\text{-MoO}_3$ with a curved plasmonic antenna," Adv. Mater., vol. 34, no. 6, p. 2104164, 2022.
- [25] S. Dai, et al., "Graphene on hexagonal boron nitride as a tunable hyperbolic metamaterial," Nat. Nanotechnol., vol. 10, no. 8, pp. 682-686, 2015.
- [26] T. G. Folland, et al., "Reconfigurable infrared hyperbolic metasurfaces using phase change materials," Nat. Commun., vol. 9, no. 1, p. 4371, 2018.

- [27] K. Chaudhary, et al., "Polariton nanophotonics using phase-change materials," Nat. Commun., vol. 10, no. 1, p. 4487, 2019.
- [28] P. Li, et al., "Reversible optical switching of highly confined phonon - polaritons with an ultrathin phase-change material," Nat. Mater., vol. 15, no. 8, pp. 870-875, 2016.
- [29] S. Dai, et al., "Phase-change hyperbolic heterostructures for nanopolaritonics: a case study of hBN/VO₂," Adv. Mater., vol. 31, no. 18, p. 1900251, 2019.
- [30] S. Dai, et al., "Efficiency of launching highly confined polaritons by infrared light incident on a hyperbolic material," Nano Lett., vol. 17, no. 9, pp. 5285-5290, 2017.
- [31] C. Meng, et al., "Broadband hyperbolic thermal metasurfaces based on the plasmonic phase-change material In₃SbTe₂," Nanoscale, vol. 15, no. 13, pp. 6306 – 6312, 2023.
- [32] A. Heßler, et al., "In₃SbTe₂ as a programmable nanophotonics material platform for the infrared," Nat. Commun., vol. 12, no. 1, p. 924, 2021.
- [33] L. Conrads, et al., "Reconfigurable and polarization-dependent grating absorber for large-area emissivity control based on the plasmonic phase-change material In₃SbTe₂," Adv. Opt. Mater., vol. 11, no. 8, p. 2202696, 2023.
- [34] A. Heßler, L. Conrads, K. G. Wirth, M. Wuttig, and T. Taubner, "Reconfiguring magnetic infrared resonances with the plasmonic phase-change material In₃SbTe₂," ACS Photonics, vol. 9, no. 5, pp. 1821-1828, 2022.
- [35] A. Heßler, S. Wahl, P. T. Kristensen, M. Wuttig, K. Busch, and T. Taubner, "Nanostructured In₃SbTe₂ antennas enable switching from sharp dielectric to broad plasmonic resonances," Nanophotonics, vol. 11, no. 17, pp. 3871 – 3882, 2022.
- [36] I. A. Oliveira, I. L. G. de Souza, and V. F. Rodriguez-Esquerre, "Programmable nanophotonic planar resonator filter-absorber based on phase-change InSbTe," Sci. Rep., vol. 13, no. 1, p. 13225,
- [37] W. Zha, et al., "Nonvolatile high-contrast whole long-wave infrared emissivity switching based on In₃SbTe₂," ACS Photonics, vol. 10, no. 7, pp. 2165-2172, 2023.
- [38] S. Zhou, et al., "Continuous programmable mid-infrared thermal emitter and camouflage based on the phase-change material In₃SbTe₂," Opt. Lett., vol. 48, no. 16, pp. 4388 – 4391, 2023.

- [39] Y. Zeng, et al., "Laser-printed terahertz plasmonic phase-change metasurfaces," Adv. Opt. Mater., vol. 11, no. 10, p. 2202651, 2023
- [40] W. Ma, et al., "Anisotropic polaritons in van der Waals materials," InfoMat, vol. 2, no. 5, pp. 777-790, 2020.
- [41] Q. Yan, R. Chen, Z. Yuan, P. Li, and X. Zhang, "Canalization acoustic phonon polaritons in metal-MoO₃-metal sandwiched structures for nano-light guiding and manipulation," J. Opt., vol. 24, no. 2, n 024006 2022
- [42] S. G. Menabde, et al., "Low-loss anisotropic image polaritons in van der Waals crystal α-MoO₃," Adv. Opt. Mater., vol. 10, no. 21, p. 2201492, 2022.
- [43] S. G. Menabde, et al., "Near-field probing of image phonon-polaritons in hexagonal boron nitride on gold crystals," Sci. Adv., vol. 8, no. 28, p. eabn0627, 2022.
- [44] I.-H. Lee, et al., "Image polaritons in boron nitride for extreme polariton confinement with low losses," Nat. Commun., vol. 11, no. 1, p. 3649, 2020.
- [45] V. E. Babicheva, S. Gamage, L. Zhen, S. B. Cronin, V. S. Yakovlev, and Y. Abate, "Near-field surface waves in few-layer MoS2," ACS Photonics, vol. 5, no. 6, pp. 2106-2112, 2018.
- [46] F. Gesuele, et al., "Towards routine near-field optical characterization of silicon-based photonic structures: an optical mode analysis in integrated waveguides by transmission AFM-based SNOM," Phys. E Low-Dimens. Syst. Nanostruct., vol. 41, no. 6, pp. 1130-1134, 2009.
- [47] G. Álvarez-Pérez, K. V. Voronin, V. S. Volkov, P. Alonso-González, and A. Y. Nikitin, "Analytical approximations for the dispersion of electromagnetic modes in slabs of biaxial crystals," Phys. Rev. B, vol. 100, no. 23, p. 235408, 2019.
- [48] L. Conrads, L. Schüler, K. G. Wirth, M. Wuttig, and T. Taubner, "Direct programming of confined surface phonon polariton resonators using the plasmonic phase-change material In₃SbTe₃," arXiv preprint arXiv:2310.12841, 2023. https://doi.org/10.48550/arXiv 2310 12841

Supplementary Material: This article contains supplementary material (https://doi.org/10.1515/nanoph-2023-0911).