

# High-Temperature Mass Spectrometric Study of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> System

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## ABSTRACT

The CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system, containing about 10.5 and 16.5 mole % TiO<sub>2</sub> was examined at temperatures 1850-2200 K using the Knudsen effusion method. Partial pressures of SiO and O over samples were determined by the complete isothermal vaporization method and by the ion current comparison method at 1850 K. The SiO<sub>2</sub> activities in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system were found to have negative deviations from the ideality for all compositions except the sample with high SiO<sub>2</sub> concentration (CaO/SiO<sub>2</sub>=0.6, 16 mole % TiO<sub>2</sub>). Replacement of CaO with TiO<sub>2</sub> in slags of relatively high basicity ( $x_{\text{CaO}}/x_{\text{SiO}_2} > 1$ ) causes a sharp increase in the SiO<sub>2</sub> activity and activity coefficient. In more acidic slags, addition of TiO<sub>2</sub> to the CaO-SiO<sub>2</sub> system does not have a strong effect on the SiO<sub>2</sub> activity and activity coefficient.

## INTRODUCTION

Thermodynamic properties of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system are important from the viewpoint of smelting of titanium containing iron ores and titanium extraction processes.

According to Froberg *et al.* [1,2], addition of TiO<sub>2</sub> to the CaO-SiO<sub>2</sub> system increases the fluidity of slags. The viscosities of CaO-TiO<sub>2</sub>-SiO<sub>2</sub> melts are in the range from 0.5 to 4 Pa s at temperatures of about 1800 K. The values of the specific electrical conductivity of this system at 1470-1740 K were found by Mori [3] to be equal to 0.1-1.0 Ohm<sup>-1</sup>cm<sup>-1</sup>. It was concluded that these melts have the ionic

conductivity, which increases with TiO<sub>2</sub> content at constant CaO:SiO<sub>2</sub> ratio and with increasing CaO:SiO<sub>2</sub> ratio at constant TiO<sub>2</sub> content.

It has been noted [4] that various aspects of the behaviour of TiO<sub>2</sub> in slags can be accounted for by the specific structure of the electron shell of a titanium atom and by the high polarizability of the Ti<sup>4+</sup> cation.

Thermodynamic properties of multi-component slags based on the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system were studied by the gas-slag equilibrium technique with addition of FeO and MnO [4], and MgO and Al<sub>2</sub>O<sub>3</sub> [5]. The following oxides were identified in the condensed phase: Ti<sub>2</sub>O, TiO, Ti<sub>2</sub>O<sub>3</sub>, Ti<sub>3</sub>O<sub>5</sub>, Ti<sub>4</sub>O<sub>7</sub>, TiO<sub>2</sub>, Ti<sub>5</sub>O<sub>9</sub> and Ti<sub>10</sub>O<sub>19</sub>.

Titanium partitioning between Ti<sup>3+</sup> and Ti<sup>4+</sup> valency states in the CaO-SiO<sub>2</sub>-TiO<sub>x</sub> system as a function of temperature, oxygen potential, titanium oxide content and CaO:SiO<sub>2</sub> ratio was examined by Tranell *et al.* [5]. It was found that the Ti<sup>3+</sup>/Ti<sup>4+</sup> ratio decreases with the increasing CaO/SiO<sub>2</sub> ratio, increasing oxygen potential and decreasing temperature.

Phase equilibria in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system have been considered in Ref. 6 and in Ref. 7 using microprobe analysis (beam size of 200 μm). Sphene (CaTiSiO<sub>5</sub>) with the melting point of 1655 K was the only ternary compound identified in the condensed phase of this system.

Literature data on the vaporization and thermodynamic properties of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system are extremely limited. The nitride capacity of this system, studied by Sakai and Suito [8] at 1873K was found to be strongly affected by titanium oxide. They reported a significant loss of SiO<sub>2</sub> at low oxygen partial pressures (log p<sub>O2</sub> = -11 to -14 atm). The CaO-Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> system was also examined by Tanabe and

Suito /9/ using the gas-slag equilibrium technique, and thermodynamic data on titanium oxides activities and slag nitride capacity were obtained.

Vaporization and thermodynamic properties of titanium oxides by high temperature mass spectrometric method were examined by Semenov /10/ and Gilles *et al.* /11-26/. The high temperature mass spectrometric study of the corresponding binary systems of the ternary CaO-SiO<sub>2</sub>-TiO<sub>x</sub> system was carried out in the following temperature ranges: for the CaO-SiO<sub>2</sub> system at 1933-2133 K /27/, for the TiO-TiO<sub>2</sub> /28-30/, Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> /28-30/ and CaO-Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> /31/ systems at 1700-2200 K. These systems were vaporised from molybdenum effusion cells and the following species were identified in the vapour phase: (i) the CaO-SiO<sub>2</sub> system: Ca, CaO, SiO, SiO<sub>2</sub>, CaSiO<sub>3</sub>, and O<sub>2</sub>; (ii) TiO-TiO<sub>2</sub>, Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> and CaO-Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> systems: Ti, TiO, TiO<sub>2</sub>, and O<sub>2</sub>. In both cases vaporization of the MoO<sub>3</sub> and MoO<sub>2</sub> species was also observed. Thus the composition of vapour over the systems studied included the same species as those over individual oxides CaO /27, 32/, SiO<sub>2</sub> /27,32/ and TiO, Ti<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> /32/. In the CaO-SiO<sub>2</sub> and CaO-Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> systems, negative deviations from the ideality were observed /27, 31/. The sign of deviations from the ideality in the TiO-TiO<sub>2</sub> and Ti<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> systems changes depending on the concentration of titanium in various oxidation states /28-30/.

The aim of the present study was identification of the vapour composition and determination of thermodynamic properties in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system using the high-temperature mass spectrometric method.

## EXPERIMENTAL

Two series of samples were studied. In the first series, the titanium oxide content was 10.5 mole % and CaO/SiO<sub>2</sub> ratio was between 0.82 and 1.30. The second series of samples contained 16-17 mole % TiO<sub>2</sub> at CaO/SiO<sub>2</sub> ratio from 0.60 to 1.30. Preparation of samples was described elsewhere /5/. Chemical compositions of the samples, determined by Inductively Coupled Plasma-Atomic Emission Spectroscopy, are presented in Table 1. The content of titanium in the reduced state (in the form of Ti<sub>2</sub>O<sub>3</sub>) in all samples was no more than 0.5 mole %.

This study was carried out using the MSI301 magnetic mass spectrometer, designed for Knudsen effusion high-temperature mass spectrometry of low volatile inorganic substances and manufactured by the Institute of Analytical Instrumentation of Russian Academy of Sciences, St Petersburg. The experimental set-up is described in detail in Ref. 33. The mass resolution ( $k 0.1$ ) of the MSI301 mass spectrometer on the lines intensities level of 10% is 600. Experiments were carried out at the ionization energy of 12 and 25 eV. The sensitivity of the mass-spectrometer determined for the gold vapour was not less than  $10^{-9}$  atm. A molecular beam from the Knudsen effusion cell was perpendicular to the electron beam from the ion source. The mass spectrometer was separated from the Knudsen effusion cell by a moving shutter to filter background peaks appearing in the mass spectrometer from the molecular beam. The experimental set-up included a special transmission mechanism, which allowed two effusion cells

**Table 1**  
Chemical composition of samples of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system\*

No of sample	Concentration, mole fraction		
	TiO <sub>2</sub>	CaO	SiO <sub>2</sub>
1	0.105	0.505	0.388
2	0.105	0.462	0.431
3	0.106	0.400	0.489
4	0.157	0.475	0.365
5	0.166	0.429	0.401
6	0.165	0.370	0.458
7	0.160	0.311	0.520

\* Ti<sub>2</sub>O<sub>3</sub> content in all samples studied was less than 0.5 mole %

to be held in the same block simultaneously, and therefore, to study two samples under the same experimental conditions. The samples were vaporised in the molybdenum effusion cell with the ratio of the vaporization area to the effusion orifice of 100:1. The effusion cells were put into molybdenum block and heated by the electron beam. The temperature was measured by the optical pyrometer EOP-66 with the vanishing glow lamp filament /33/, produced by the Kharkov Optical Plant, Ukraine. The temperature gradient between the top and bottom of the effusion cells was  $10 \pm 5$  K at the temperature 2000 K.

The experimental installation was calibrated using gold (99.999 wt%) and anhydrous  $\text{CaF}_2$  (99.9 wt%).

## RESULTS AND DISCUSSION

### Calculation of partial pressures of species and thermodynamic properties

The vapour partial pressure ( $p_i$ ), sublimation enthalpy  $\Delta H_{s,T,i}$  and thermodynamic activity  $a_i$  of a component  $i$  at temperature  $T$  were calculated as follows /33-35/:

$$p_i = b \sum_j I_{ij} T / \sigma_i, \quad (1)$$

$$p_i = p_s I_{ij} T_i \sigma_s \gamma_s / I_s T_s \sigma_i \gamma_j, \quad (2)$$

$$p_i = (q_i / s t L) (2 \pi R T / M_i)^{-1/2}, \quad (3)$$

$$d \ln K_p / dT = H^\circ_{s,T,i} / R T^2, \quad (4)$$

$$a_i = p_i / p_i^\circ, \quad (5)$$

where  $I_{ij}$  is the value of the  $j$ -ion current resulting from ionization of the  $i$ -component of the gaseous phase;  $\sigma_i$  is the ionization cross-section of vaporizing molecule;  $b$  is the device sensitivity constant;  $q_i$  is the mass of the substance vaporized from the surface area  $s$  in time  $t$  through the effusion orifice of the Knudsen cell with the Clausing coefficient  $L$ ;  $M_i$  is the molecular mass of the  $i$ -component;  $R$  is the gas constant;  $K_p$  is the equilibrium constant for the reaction of vapourisation;  $H^\circ$  is the enthalpy of vaporization; index "o" corresponds to the component in the standard state. The ionization cross-section of molecules in Eqn.2, were calculated from the atomic ionization cross-sections

/36,37/. Partial pressures were obtained by the ion current comparison method (Eqn. 2), and by the complete isothermal vaporization method (Eqn. 3).

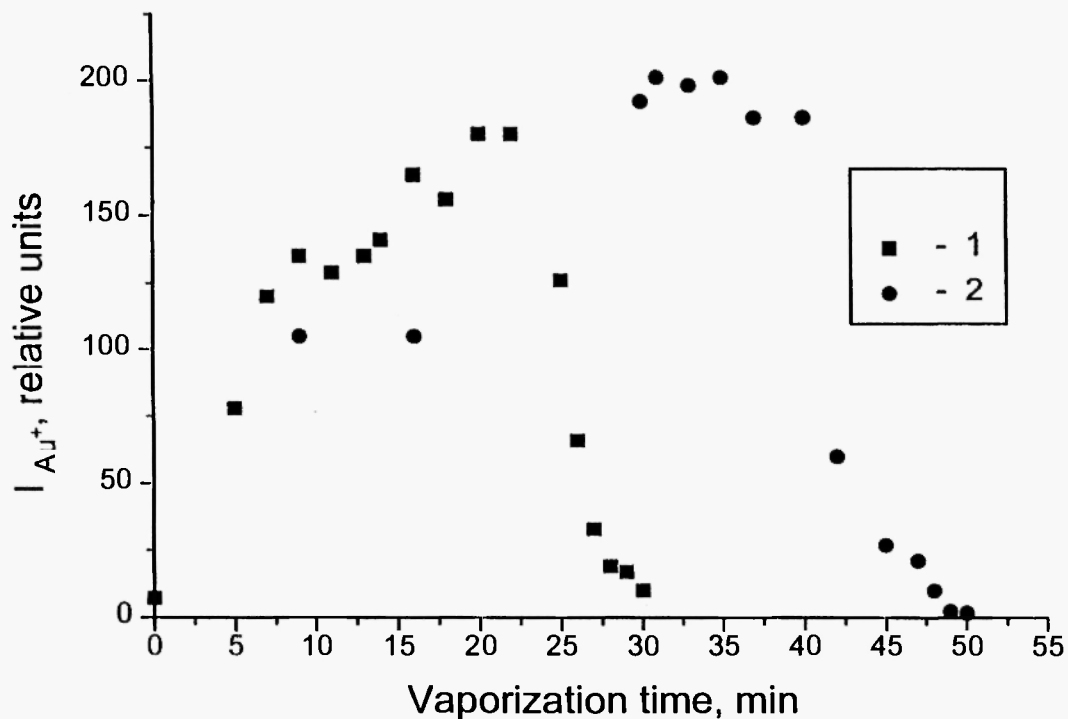
### Vaporization of standard samples of Au and $\text{CaF}_2$

Vaporization of gold (99.999 wt%) from molybdenum effusion cell was carried out in accordance with the recommendations of IUPAC (Division of Inorganic Chemistry Commission on High Temperatures and Refractory Materials) as a standard for measuring the vapour pressure and enthalpy of sublimation of materials /38/.

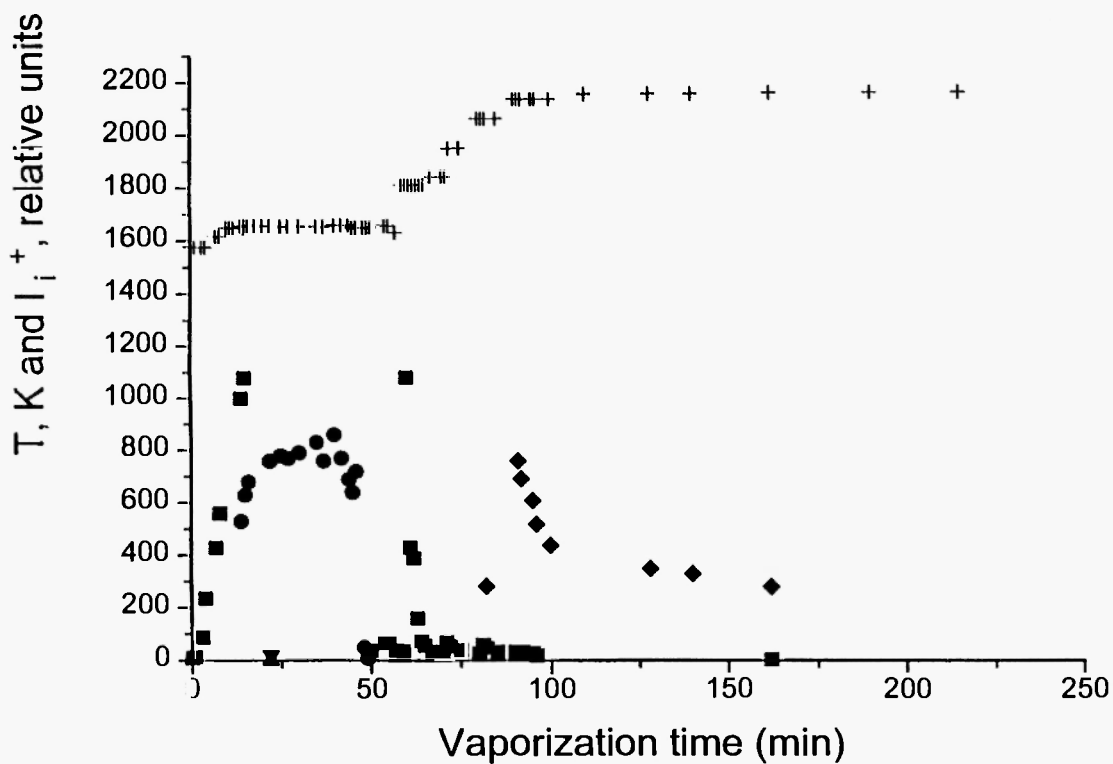
Data on the vapour pressure of gold were obtained by the complete isothermal vaporization method (Eqn. 3), from both molybdenum cells as a function of the complete vaporization time of gold at 1744 K (Fig. 1). The Au partial pressure was found to be  $(4.3 \pm 0.6) \cdot 10^{-5}$  atm, which is in a good agreement with data recommended by Paule and Mandel /38/.

A  $\text{CaF}_2$  sample was used as an additional standard. Data on thermodynamic properties of  $\text{CaF}_2$  are summarized in Refs. 39,40. Vaporization of  $\text{CaF}_2$  was studied by the Knudsen effusion mass spectrometric method in works /41,42/. The main ion species, identified in the vapour mass spectra of  $\text{CaF}_2$ , were  $\text{Ca}^+$ ,  $\text{CaF}^+$ , which are the fragment ions, and  $\text{CaF}_2^+$ , which is the molecular ion. The fragment ion  $\text{CaF}^+$  was the most intensive in the vapour mass spectra of  $\text{CaF}_2$ . At the ionization energy 25 eV and temperature 1613 K the ratio of intensities of  $\text{Ca}^+:\text{CaF}^+:\text{CaF}_2^+$  in the  $\text{CaF}_2$  vapour mass spectra was  $(0.0014 \pm 0.0003): 1: (0.031 \pm 0.002)$ . Vaporization isotherms at 1658 K used for the determination of  $\text{CaF}_2$  partial pressures (Eqn. 3) are shown in Fig.2. The further temperature increase is also indicated in Fig.2, which was done with the aim of vaporizing the substance completely from the cell. The calculated partial pressure of  $\text{CaF}_2$  as a function of temperature is plotted in Fig.3. It agrees well with reference data /40/ also presented in Fig. 3. The sublimation enthalpy of  $\text{CaF}_2$ , calculated using plot of  $\log (T I_{\text{CaF}_2^+})$  vs  $1/T$  and Eqns. 1 and 4, was found to be  $(389.3 \pm 5.4)$  kJ/mole. Thus,  $\text{CaF}_2$  partial pressure as a function of temperature is described by the following equation in the temperature range of 1600-1690 K :

$$\log p_{\text{CaF}_2} (\text{atm}) = -(20422 \pm 283) / T + (13.99 \pm 0.03) \quad (6)$$



**Fig. 1:**  $\text{Au}^+$  ion current from two molybdenum cells as a function of vaporization time of gold at 1744 K in calibration experiments: 1 - data obtained during vaporization from cell 1 and 2 - data from cell 2.



**Fig. 2:**  $\text{Ca}^+$  ( $\blacklozenge$  - cell 1,  $\nabla$  - cell 2),  $\text{CaF}^+$  ( $\blacksquare$  - cell 1,  $\bullet$  - cell 2),  $\text{CaF}_2^+$  ( $\Delta$  - cell 2) ion currents from two molybdenum cells as functions of the vaporization time of  $\text{CaF}_2$  at 1658 K in calibration experiments. + - temperature change.

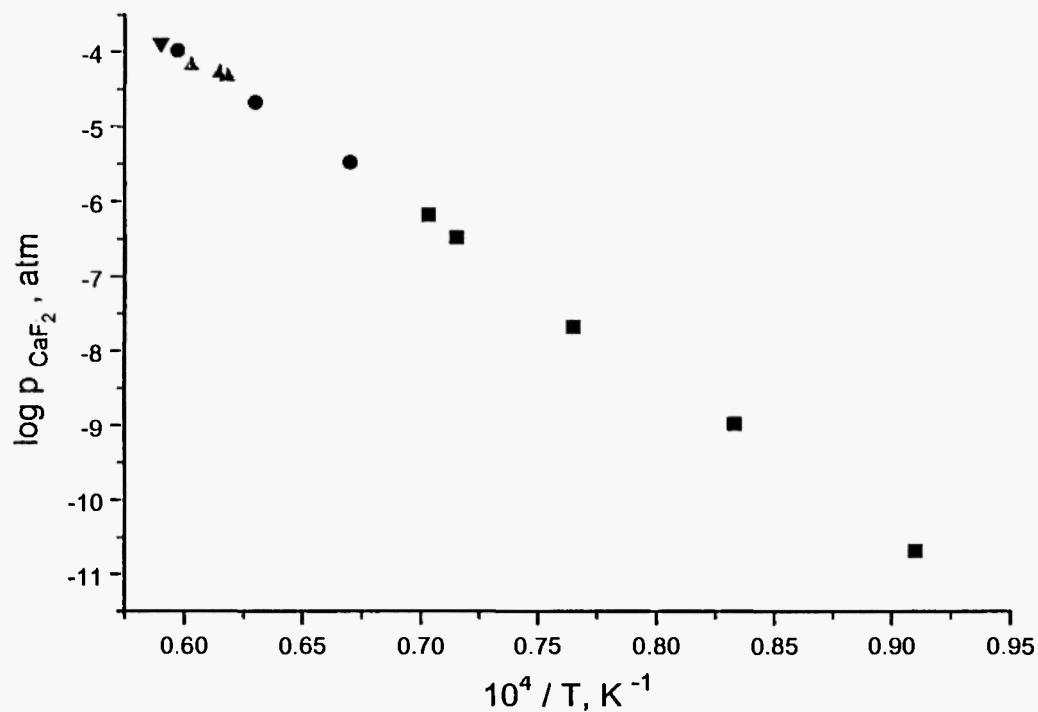


Fig. 3: Partial pressure of  $CaF_2$  over pure  $CaF_2$  as a function of temperature obtained in the present study ( $\Delta$ ) and recommended in Ref. 44:  $\blacksquare$  the temperature range is 1100-1424 K;  $\bullet$  -1424-1691 K;  $\nabla$  temperature is higher than 1691 K.

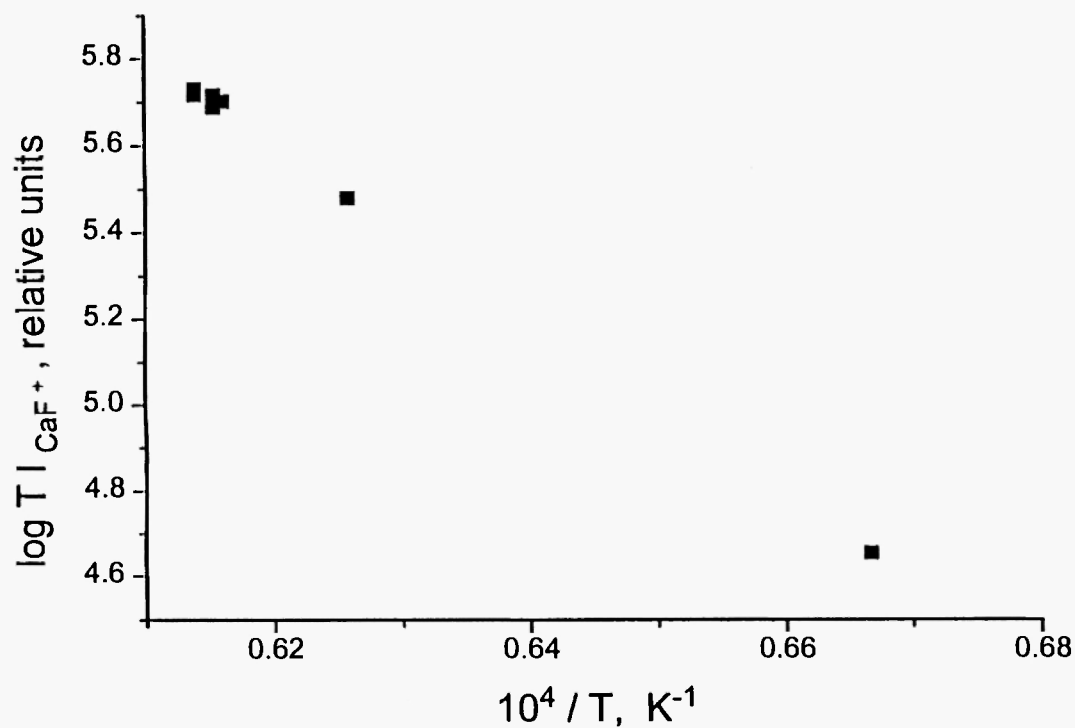


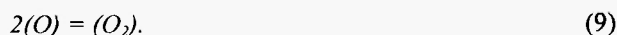
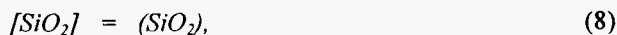
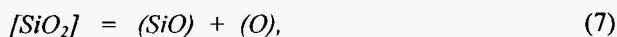
Fig. 4: Temperature dependence of the  $CaF^+$  ion current in mass spectra of vapour over  $CaF_2$  for the determination of vaporization enthalpy using Eqns. 1,4.

**Table 2**  
Partial pressures of vapour species over pure SiO<sub>2</sub> as a function of temperature  
(vaporization from Mo- cells).

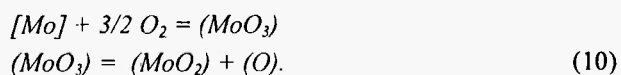
Vapour species	Temperature (K)	log p <sub>i</sub> = - A/ T + B (atm)		References
		A	B	
SiO	1805-1995	27525±833	10.53±0.43	Present study
	2000-2038	25680±2105	9.61±1.04	Present study
	1600-1800	23802±2110	6.58±1.73	Ref. 46
	1601-1754	18266±1617	3.66±0.96	Ref. 45
O	1805-1995	26972±971	5.62±0.43	Present study
	2000-2038	25822±1032	5.93±0.66	Present study
	1600-1800	25250±2140	6.55±1.03	Ref. 46
	1601-1754	29992±2539	9.63±1.51	Ref. 45
SiO <sub>2</sub>	1833-1995	30710±510	8.46±0.47	Present study
	2000-2038	28771±669	7.60±0.64	Present study
	1996-2300	28834± 35	8.27±0.02	Ref. 40

### Vaporization of pure silica

In the mass spectra of vapour over pure SiO<sub>2</sub>, the following ions were identified at the ionization energy 25 eV in the temperature range 1700-2350 K: SiO<sup>+</sup>, Si<sup>+</sup>, SiO<sub>2</sub><sup>+</sup>, MoO<sub>3</sub><sup>+</sup>, and MoO<sub>2</sub><sup>+</sup>. The ratio of currents of SiO<sup>+</sup>:Si<sup>+</sup> ions was 1:0.14. This vapour mass spectra is in agreement with our previous results [27,43,44] and the data from Ref. 32. The SiO<sup>+</sup> and SiO<sub>2</sub><sup>+</sup> ions are the parent ones and Si<sup>+</sup> is the result of fragmentation. Silica vaporizes according to the following reactions:



The MoO<sub>3</sub><sup>+</sup> and MoO<sub>2</sub><sup>+</sup> ions in the vapour phase were formed as a result of interaction of Mo-cell with oxygen:



(Throughout the paper, chemical formulae in round brackets correspond to components in the gas phase, while those in square brackets refer to the condensed phase.) Partial pressures of vapour species over SiO<sub>2</sub> were obtained by the complete isothermal vaporization method (equation 3), and by the ion current comparison method (equation 2) with the use of gold as a standard. The data on silica partial pressure as a function of temperature are presented in Table 2 in comparison with literature data [45, 46], obtained under similar conditions, when vaporization was carried out from Mo-cells. Partial pressures of SiO<sub>2</sub> obtained at the present study are in agreement with data recommended in Ref. 40.

### Vaporization of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system

Samples of 3-8 mg of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system were vaporized from the Knudsen cell. The standard sample of CaF<sub>2</sub> was vaporized simultaneously from the second effusion cell for determination of partial pressures of vapour species by the ion current comparison method (Eqn. 2) and to control the sensitivity of the mass spectrometer during experiments. At temperatures up to 1900 K the main vapour

**Table 3**

Partial pressures of SiO and O over samples of the CaO-SiO<sub>2</sub>-TiO<sub>2</sub> system at 1850 K, obtained by the complete isothermal vaporization method, Eqn. 3, (I) and by the ion current comparison method, Eqn. 2, (II).

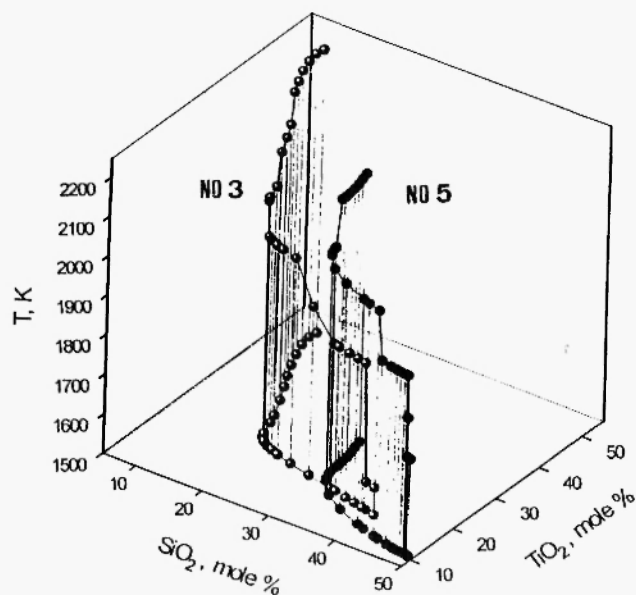
No of sample	Partial pressure, atm		
	SiO*10 <sup>5</sup>	SiO*10 <sup>5</sup>	O*10 <sup>9</sup>
	(I)	(II)	(II)
1	0.463	0.283	1.14
2	0.701	0.690	1.71
3	0.830	0.878	2.60
4	0.368	0.299	1.04
5	0.537	0.716	1.94
6	1.020	1.490	2.11
7	1.830	2.770	3.67

species in vapour were SiO, MoO<sub>3</sub> and MoO<sub>2</sub>. Table 3 presents partial pressures of SiO over samples No 1-7 at 1850 K, obtained by the complete isothermal vaporization method (Eqn. 3), and by the ion current comparison method (Eqn. 2). Partial pressures of oxygen, presented in this table, were calculated from a ratio of MoO<sub>3</sub><sup>+</sup> and MoO<sub>2</sub><sup>+</sup> ion currents ( $I_{\text{MoO}_3^+}/I_{\text{MoO}_2^+}$ ) using equilibrium (10):

$$p\text{O} = K_{o10} [I_{\text{MoO}_3^+} \cdot \sigma(\text{MoO}_2) / I_{\text{MoO}_2^+} \cdot \sigma(\text{MoO}_2)], \quad (11)$$

where  $K_{o10}$  is the equilibrium constant of reaction (10), calculated using data from Ref. 40. Partial pressures of Ca, TiO and TiO<sub>2</sub> evaluated by the ion current comparison method (Eqn. 2) at 1900 K were lower than 10<sup>-8</sup> atm.

As a result of significant difference in the volatility between SiO, Ca, TiO and TiO<sub>2</sub> at 1850 K, initial compositions of the system changed during isothermal vaporization of samples. For example, Fig. 5 illustrates changes in concentrations of TiO<sub>2</sub> and SiO<sub>2</sub> in samples No 3 and No 5 of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system relative to the initial concentrations as a result of the selective vaporization of components in the temperature range 1500-2200 K, which was calculated by the complete isothermal vaporization method [47,48].



**Fig. 5:** Changes in concentrations of TiO<sub>2</sub> and SiO<sub>2</sub> in samples No 3 and No 5 of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system as a result of the selective vaporization of components in the temperature range 1500-2200 K determined by the complete isothermal vaporization method.

Experimental data for samples of changing composition were treated by the following method. The change in the composition of a sample after vaporization of some quantity of SiO<sub>2</sub> in the form of SiO, O and SiO<sub>2</sub> species was calculated as:

$$q_1 = q_0 (1 - s_1/s_0), \quad (12)$$

where  $q_1$  is the fraction of the substance (SiO<sub>2</sub>) evaporated by time  $t_1$  from the initial sample;  $q_0$  is the quantity of this substance (SiO<sub>2</sub>) in the sample before vaporisation;  $s_0$  is the area under the plot of the SiO<sup>+</sup> ion current as a function of time for the time of complete evaporation  $t$ ;  $s_1$  is the area under the same plot for time  $t_1$ . It was assumed that vaporization of a sample (SiO<sub>2</sub>) eventually goes to completion as a result of gradually increasing temperature.

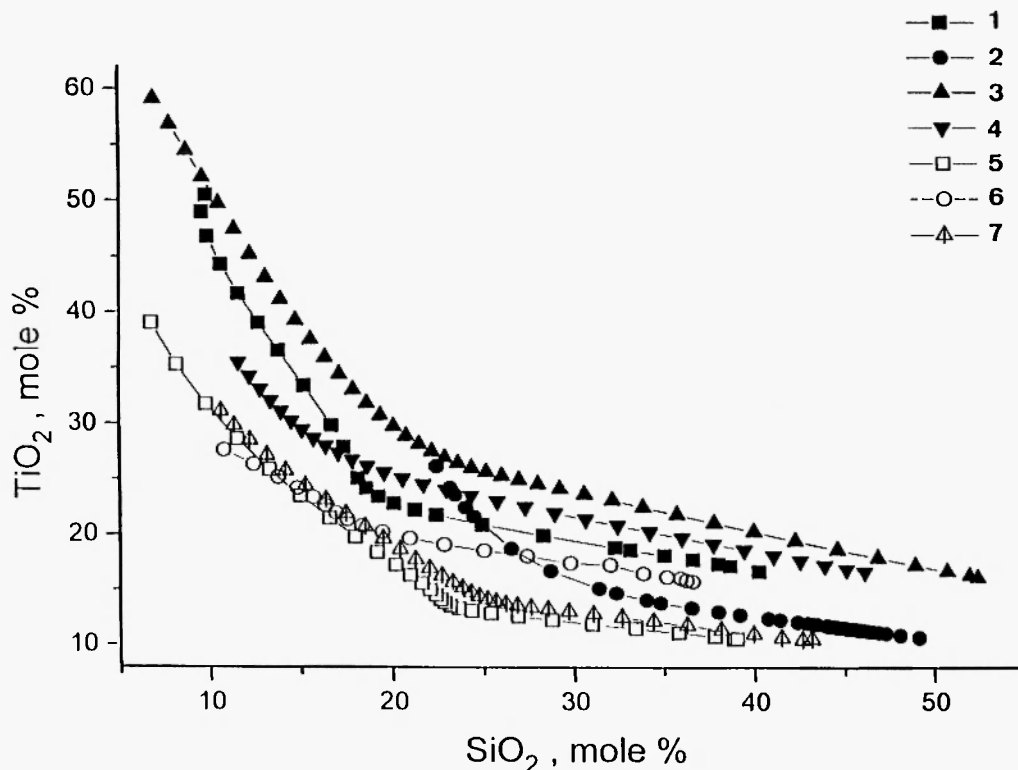
The quantity of remaining CaO and TiO<sub>2</sub> in the residual sample was calculated from the ratio of the areas  $s'_0$  and  $s'_1$  for CaO and TiO<sub>2</sub> from the ratio of the areas  $s''_0$  and  $s''_1$  by the same way. A sample for the experiment was taken of such a mass, to have the rate of change in the sample composition not greater than 0.01-0.04 mole fractions/hour,

as required for the complete isothermal vaporization method /47,48/.

The accuracy of such calculation of partial pressures of vapour species in oxide systems was found to be satisfactory in an earlier study /49,50/ of the B<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub> and B<sub>2</sub>O<sub>3</sub>-Al<sub>2</sub>O<sub>3</sub> systems.

Fig. 6 summarizes changes in concentrations of SiO<sub>2</sub> and TiO<sub>2</sub> in samples No 1-7 of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system during the selective vaporization of components, when the temperature of vaporization increased from 1550 to 2175 K. Enrichment of the condensed phase of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system with CaO and TiO<sub>2</sub> during vaporization is also presented on the phase equilibrium diagram /6/ for the all samples studied (Fig. 7). These data on the distillation of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system during vaporization can be useful in various high-temperature processes.

In the vapour mass spectra of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system (samples No 1-7, Table 1) the following ions were identified at the ionizing-electron energy of 25 eV in the temperature range of 1700-2350 K: Ca<sup>+</sup>, TiO<sup>+</sup>, TiO<sub>2</sub><sup>+</sup>, Ti<sup>+</sup>, SiO<sup>+</sup>, SiO<sub>2</sub><sup>+</sup>, Si<sup>+</sup>, MoO<sub>2</sub><sup>+</sup>, and MoO<sub>3</sub><sup>+</sup>. The ratio of SiO<sup>+</sup>/Si<sup>+</sup> ions was the same as in the mass spectra of vapour over pure SiO<sub>2</sub>. The



**Fig. 6:** Changes in concentration of SiO<sub>2</sub> and TiO<sub>2</sub> in samples No 1-7 of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system during the selective vaporization of components, when the temperature of vaporization increased from 1750 to 2175 K.



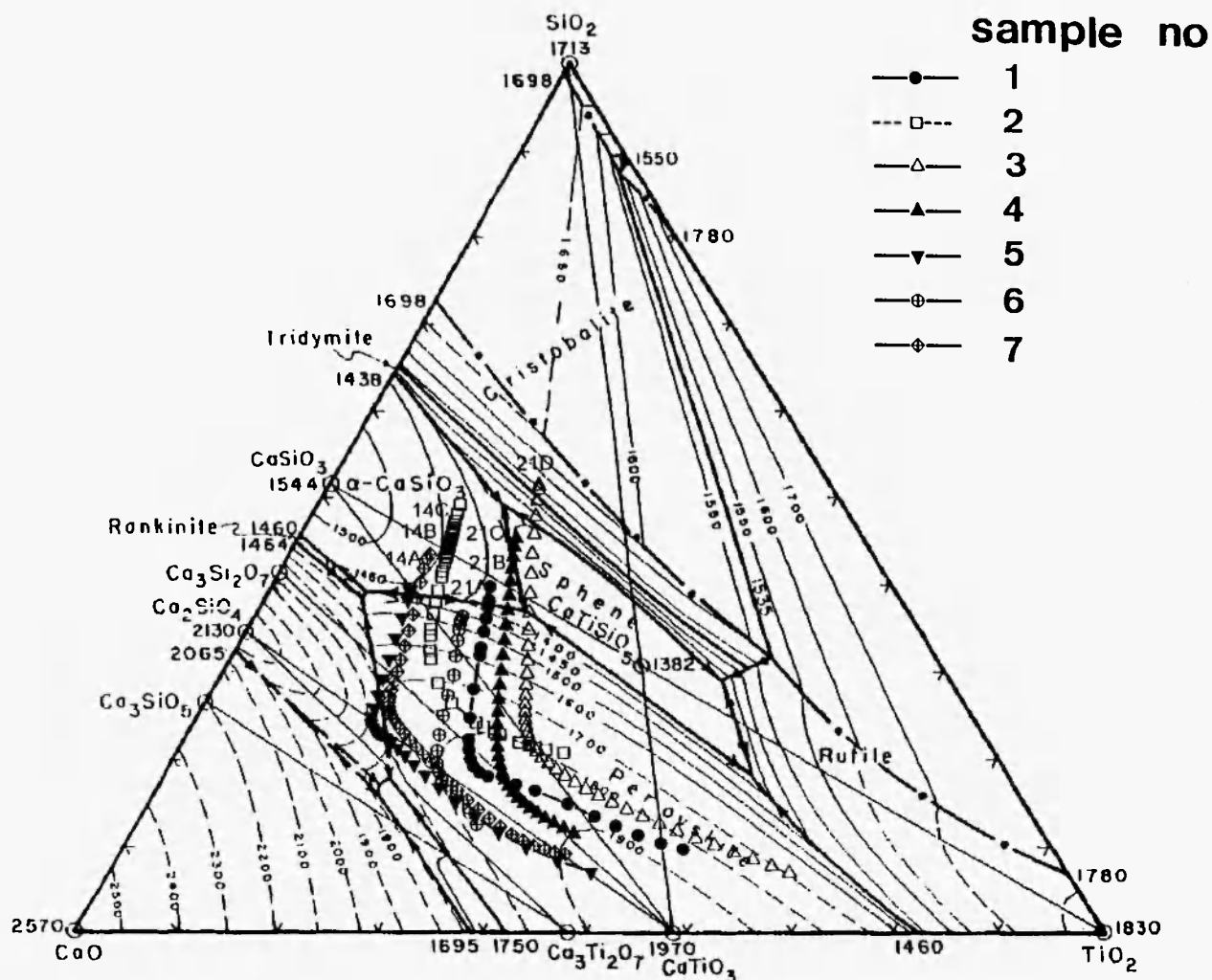
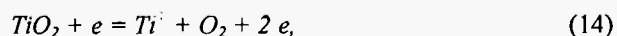
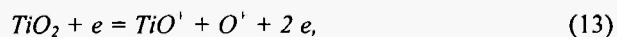


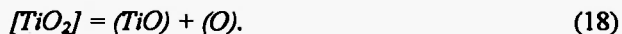
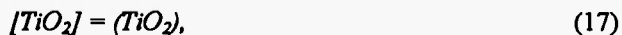
Fig. 7: Phase equilibrium diagram for the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system from Ref. 10 with an indication of the changes in composition of samples No 1-7 during selective vaporization of components, when the temperature of vaporization increased from 1750 to 2175 K.

intensity of the Ti<sup>+</sup> ion current was evaluated to be about 7 % of the value of the TiO<sup>+</sup> ion current. Further experiments were carried out at the ionizing-electron energy of 12 eV to be sure that ionization of TiO<sub>2</sub> and TiO species in accordance with reactions



could be neglected. The ionization processes in vapour over TiO<sub>2</sub> and TiO oxides were studied by Banon *et al.* /29/ who showed that at low ionizing electron energies (12.2 eV), the TiO<sup>+</sup> ion may be considered as the parent. (Ionization energy of Ti and TiO is (6.3±0.3)eV and (6.8±0.5)eV respectively /51/). The vapour mass spectra over the samples of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system of different compositions showed the same ion species as in the mass spectra of vapour over individual oxides /32/ CaO, TiO<sub>2</sub>, and SiO<sub>2</sub> and over the CaO-SiO<sub>2</sub> /27/, CaO-TiO<sub>2</sub>-Ti<sub>2</sub>O<sub>3</sub> /31/ systems. The main vaporization processes for the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system in the

temperature range 1700-2350 K were presented by reactions (7-10) and (16)-(18):



#### Thermodynamic activity of SiO<sub>2</sub> in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system

The SiO<sub>2</sub> activity in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system at 1850 K was calculated using experimental data on vapour pressure of SiO and O (reaction 7) measured by the values of corresponding ion currents, over the samples of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system and pure SiO<sub>2</sub>. It follows from Equation (5), that

$$a_{SiO_2} = (p_{SiO}/p^{\circ}_{SiO}) * (p^{\circ}_O/p_O), \quad (5a)$$

where  $p_i$  and  $p^{\circ}_i$  are the values of ion currents of vapour species over the samples of the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system and over SiO<sub>2</sub>, respectively. Calculated partial pressures of SiO and O are given in Table 3. In Table 4, the activity of SiO<sub>2</sub> in the ternary CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system is presented in comparison with data for the binary CaO-SiO<sub>2</sub> system also obtained by the Knudsen mass-spectrometric method.

SiO<sub>2</sub> activities and activity coefficients in the binary

CaO-SiO<sub>2</sub> system decrease with decreasing basicity ( $x_{CaO}/x_{SiO_2}$  ratio). The same trend is observed for the ternary CaO-SiO<sub>2</sub>-TiO<sub>2</sub> system. Replacement of CaO with TiO<sub>2</sub> in the slag of basicity  $x_{CaO}/x_{SiO_2}=1.07$  with about the same SiO<sub>2</sub> concentration ( $x_{SiO_2}=0.431-0.434$ ) results in sharp increase in SiO<sub>2</sub> activity and activity coefficient. This could be expected because TiO<sub>2</sub> is regarded as an acidic oxide, while CaO is a basic oxide. However, the effect of TiO<sub>2</sub> on the SiO<sub>2</sub> activity and activity coefficient in slags of low basicity is rather small, which could be an indication that in the acidic slag TiO<sub>2</sub> behaves as a basic oxide. To determine accurately the thermodynamic activities of CaO and TiO<sub>2</sub> in the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system, much higher temperatures are required, as vapour pressures of species in the vaporization of these oxides are much lower than of SiO. However, at high temperatures SiO<sub>2</sub> would be about completely vaporised. Because of this limitation, activities of CaO and TiO<sub>2</sub> were not determined.

#### CONCLUSIONS

The CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system, containing about 10.5 and 16.5 mole% TiO<sub>2</sub> was examined at temperatures of 1850-2200 K using the Knudsen effusion method. In the vapour mass spectra of this system the following ions were identified at the ionizing-electron energy of 25 eV in the temperature range of 1700-2350 K: Ca<sup>+</sup>, TiO<sup>+</sup>, TiO<sub>2</sub><sup>+</sup>, Ti<sup>+</sup>, SiO<sup>+</sup>, SiO<sub>2</sub><sup>+</sup>, Si<sup>+</sup>, MoO<sub>2</sub><sup>+</sup> and MoO<sub>3</sub><sup>+</sup>. As a result of

Table 4  
SiO<sub>2</sub> activity and activity coefficient in ternary CaO-SiO<sub>2</sub>-TiO<sub>2</sub> and binary CaO-SiO<sub>2</sub> systems

Ternary CaO-SiO <sub>2</sub> -TiO <sub>2</sub> system					Binary CaO-SiO <sub>2</sub> system		
$x_{TiO_2}^*$	$x_{CaO}$	$x_{SiO_2}$	$a_{SiO_2}$	$\gamma_{SiO_2}$	$x_{CaO}$	$x_{SiO_2}$	$\gamma_{SiO_2}$
0.105	0.505	0.388	0.07	0.189			
0.105	0.462	0.431	0.15	0.348	0.564	0.434	0.035
0.106	0.400	0.489	0.27	0.552	0.517	0.483	0.455
0.157	0.475	0.365	0.05	0.137			
0.166	0.429	0.401	0.13	0.324			
0.165	0.370	0.458	0.27	0.590			
0.160	0.311	0.520	0.84	1.615	0.342	0.658	1.490

\* The balance is Ti<sub>2</sub>O<sub>3</sub>

significant difference in the volatility between SiO, Ca, TiO and TiO<sub>2</sub> initial compositions of the system changed during isothermal vaporization of samples.

Using experimental data on vapour pressure of SiO and O over the CaO-TiO<sub>2</sub>-SiO<sub>2</sub> system, the thermodynamic activity of SiO<sub>2</sub> was calculated and compared with the SiO<sub>2</sub> activity in the binary CaO-SiO<sub>2</sub> system. Replacement of CaO with TiO<sub>2</sub> in the slag of basicity  $x_{\text{CaO}}/x_{\text{SiO}_2}=1.07$  with about the same SiO<sub>2</sub> concentration results in a sharp increase in SiO<sub>2</sub> activity. However, this has little effect on the SiO<sub>2</sub> activity in slags of low basicity. This indicates that TiO<sub>2</sub> could be regarded as an acidic oxide in a basic slag and as a basic oxide in the acidic slag.

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### REFERENCES

1. H. Schenck and M.G. Froberg, *Arch. Eisenhutt.Wes.*, **33**, 421 (1962).
2. M.G. Froberg and R. Weber, *Arch. Eisenhutt.Wes.*, **36**, 477 (1965).
3. K. Mori, *Tetsu-to-Hagane*, **46**, 134 (1960).
4. I.D. Sommerville and H.B. Bell, *Canadian Metall. Quarterly*, **21**, 145 (1982).
5. G. Tranell, O. Ostrovski and S. Jahanshahi, *Proceedings The 5th International Conference on Molten Slags, Fluxes and Salts' 97*. The Iron and Steel Society, Sydney, Australia, 1997, p. 501.
6. R.C. DeVries, R. Roy and E.F. Osborn, *J. Am. Ceram. Soc.*, **38**, 158 (1955).
7. H. Sakai and H. Suito, *ISIJ International*, **36**, 138 (1996).
8. H. Sakai and H. Suito, *ISIJ International* **36**, 143 (1996).
9. J. Tanabe and H. Suito, *Steel Research*, **63**, 515 (1992).
10. G.A. Semenov, *Izvestiya AN SSSR, Neorganicheskie Materialy* **5**, 67 (1969).
11. P.W. Gilles, K. D. Carlson, H.F. Franzen and P.G. Wahlbeck, *J. Chem Phys.*, **46**, 2461 (1967).
12. P.G. Wahlbeck and P.W. Gilles, *J. Chem Phys.*, **46**, 2465 (1967).
13. P.W. Gilles, *J. Chem Phys.*, **46**, 4987 (1967).
14. P.W. Gilles, H.F. Franzen, G.D. Stone and P.G. Wahlbeck, *J. Chem Phys.*, **48**, 1938 (1968).
15. P.W. Gilles, P.J. Hampson and P.G. Wahlbeck, *J. Chem Phys.*, **50**, 1048 (1969).
16. P.W. Gilles, *The Chemistry of Extended Defects in Non-Metallic Solids*. L.Eyring and M. O'Keefe, Eds., North-Holland Publish. Co., Amsterdam, 1970, p. 75.
17. P.J. Hampson and P.W. Gilles, *J. Chem Phys.*, **55**, 3712 (1971).
18. P.W. Gilles and P.J. Hampson, *Advances in Mass Spectrometry*, **5**, A. Quayle, Ed., Elsevier Publishing Co., New York, 1971, p. 374.
19. B.R. Conard, J.E. Bennett and P.W. Gilles, *J. Chem. Phys.*, **63**, 5502 (1975).
20. P.W. Gilles, G.H. Rinehart and R.I. Sheldon, *J. Chem. Phys.*, **66**, 2229 (1977).
21. Q.D. Wheatley, R.I. Sheldon and P.W. Gilles, *J. Chem. Phys.*, **66**, 3712 (1977).
22. P.W. Gilles and R.I. Sheldon, *Rev. Int. Hautes Temper. Refract.*, **15**, 315 (1978).
23. R.I. Sheldon and P.W. Gilles, *Proceedings of the Tenth Materials Research Symposium on Characterization of High-Temperature Vapors and Gases*, N.B.S., Gaithersburg, Maryland, N.B.S. Special Publication 561, 231 (1979).
24. S. A. Heideman, T.B. Reed and P.W. Gilles, *High Temperature Sci.*, **13**, 79 (1980).
25. B. Granier and P.W. Gilles, *Rev. Int. Hautes Temper. Refract.*, **18**, 227 (1981).
26. B. Granier and P.W. Gilles, *High Temperatures-High Pressures*, **14**, 383 (1982).
27. V.L. Stolyarova, S.I. Shornikov, G.G. Ivanov and M.M. Shultz, *J. Electrochem. Soc.*, **138**, 3710 (1991).
28. S. Banon, C. Chatillon and M. Allibert, *High Temperature Sci.*, **15**, 17 (1982).
29. S. Banon, C. Chatillon and M. Allibert, *High*

- Temperature Sci.*, **15**, 105 (1982).
30. S. Banon, C. Chatillon and M. Allibert, *High Temperature Sci.*, **15**, 129 (1982).
  31. S. Banon, C. Chatillon and M. Allibert, *Canadian Metallurg. Quart.*, **20**, 79 (1981).
  32. E.K. Kazenas and D.M. Chzhikov, *Davlenie i sostav para nad okislami khimicheskikh elementov*, Nauka, Moskva, 1976.
  33. G.A. Semenov, E.N. Nikolaev, K.E. Frantseva, *Primenenie mass spektrometrii v neorganicheskoi khimii*, Leningrad, Khimiya (1976).
  34. M.G. Ingram, J. Drowart, *Proc. Int. Symp. on High Temperature Technology*, New York, (1959), p. 219.
  35. V.L. Stolyarova and G.A. Semenov, *Mass Spectrometric Study of the vaporization oxide systems*, J.H. Beynon, Ed., Wiley, Chichester, 1994.
  36. J.B. Mann, *J. Phys. Chem.* **46**, 1646 (1967).
  37. J.W. Otvos and D.P. Stevenson, *J. Am. Chem. Soc.*, **78**, 546 (1956).
  38. C.P. Paule and J. Mandel, *Pure and Applied Chem.*, **31**, 395 (1972).
  39. I. Barin, *Thermochemical Data of Pure Substances*, VCH (1993).
  40. L. V. Gurvich, I.V. Veitz, V. A. Medvedev *et al.* *Thermodynamic properties of the individual substances*, Moskva, Nauka, Vol. 3, part 2 (1981).
  41. A.I. Zaitzev, N.V. Korolev and B.M. Mogutnov, *Teplofizika vysokikh temperatur*, **27**, 465 (1989).
  42. A.I. Zaitzev, N.V. Korolev and B.M. Mogutnov, *High Temperature Sci.*, **28**, 341 (1990).
  43. V.L. Stolyarova, S.I. Shornikov and M.M. Shultz, *High Temperature Mater. Sci.*, **36**, 1 (1996).
  44. M.M. Shultz, I.Yu. Archakov, G.G. Ivanov, V.L. Stolyarova V.L., *Glastechn. Ber. Technology*, **68**, 320 (1995).
  45. A.N. Nesmeyanov and L.P. Firsova, *Zh. Fiz. Khim.*, **34**, 1907 (1960).
  46. A.N. Nesmeyanov and L.P. Firsova, *Zh. Fiz. Khim.*, **34**, 2615 (1960).
  47. L.N. Sidorov, V.I. Belousov, *Zhurnal Fizicheskoi Khimii*, **44**, 1601 (1970).
  48. L.N. Sidorov, V.B. Shol'tz, *Intern. J. Mass Spectrometry and Ion Physics*, **8**, 437 (1972).
  49. M.M. Shultz, G.G. Ivanov, V.L. Stolyarova, B.A. Shakhmatkin, *Fizika i Khimiya Stekla*, **12**, 285 (1986).
  50. V.L. Stolyarova, A.L. Shilov, G.G. Ivanov, Shultz M.M. and Seetharaman S., *Rapid Communic. Mass Spectrom.* **9**, 1244 (1995).
  51. L.V. Gurvich, G.V. Karachentzev and V.N. Kondratyev, *Energii razryva khimicheskikh svyazei. Potentsialy ionizatsii i sredstvo k elektronu*, Nauka, Moskva, 1974.